# **Mixed-Potential-Type NO<sub>x</sub> Sensor Based on YSZ and Zinc Oxide Sensing Electrode**

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**Abstract.** Electrochemical sensors using tubular yttria-stabilized zirconia (YSZ) and oxide sensing electrode (SE) were fabricated and examined for NO<sub>y</sub> detection at high temperatures. The mixed-potential-type  $NO<sub>x</sub>$  sensor using ZnO-SE gave the highest sensitivity to  $NO<sub>x</sub>$  among other single-type oxides tested as SEs in the temperature range of  $600-700$  °C. The response of the ZnOattached device was a linear for the logarithm of  $NO<sub>2</sub>$  (NO) concentrations from 40 to 450 ppm. The sensing mechanism of the sensor was discussed on the basis of the gas adsorption-desorption behavior, the catalytic activity data, and electrochemical behavior for oxides examined.

### **1. Introduction**

One of the most desirable applications for *"in-situ"* solid electrolyte  $NO<sub>x</sub>$  sensors is monitoring and controlling  $NO<sub>x</sub>$ concentration in car exhausts. However, it is necessary to develop compact reliable  $NO<sub>x</sub>$  sensors capable of operation at temperatures above ca.  $600^{\circ}$ C to meet the requirements of automotive industry for such instruments. In addition, recent legislation in EU, Japan and USA has significantly increased demand for solid-state gas sensors. For example, the EU emission limits for passenger cars and heavy-duty diesel vehicles will be reconsidered soon again [1]. So far, the most reliable sensors in the very harsh hightemperature environment of car exhausts are YSZbased oxygen sensors ( $\lambda$ -sensors). Their attractive features, such as low cost, fast response, and compactness, have led to their extensive use for on-line emission control, and their production for the world-wide automotive industry has been rising rapidly and already reached one hundred million pieces per year.

The automotive industry has released new ultra leanburn (or direct-injection type) engine systems to improve

fuel efficiency as well as to reduce  $CO<sub>2</sub>$  emissions from the engine system. In this ultra-lean-burn engine system, a new  $NO<sub>x</sub>$ -storage catalyst is needed for compensation of the low  $NO<sub>x</sub>$ -removal ability of the conventional threeway catalyst under lean-burn (air rich) condition. The  $NO<sub>x</sub>$ concentration in the exhaust gas coming from the  $NO<sub>x</sub>$ storage catalyst increases gradually with time due to the saturation of  $NO<sub>x</sub>$ -storage capacity of the catalyst as shown in Fig. 1 [2]. Fuel-rich gas containing high concentrations of hydrocarbons is allowed to flow through the catalyst to regenerate the storage ability. As a result,  $NO<sub>x</sub>$ concentration in the gas emitted from this new catalyst decreases rapidly down to zero level and then gradually increases again. It is, therefore, important to install *"insitu*" NO<sub>x</sub> sensors to optimize the catalyst performance. One of the major requirements of the *on-board* NO<sub>x</sub> sensor is determination of  $NO<sub>x</sub>$  concentration in several-tensppm level in the downstream of new catalyst so that the timing for regenerating the catalyst can be adjusted.

Solid-state gas sensors based on mixed-potential have gained a great deal of interest in the scientific and

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Fig. 1. Regeneration pattern of new  $NO<sub>x</sub>$ -storage catalyst for a direct-injection-type engine.

technological sensor communities [3]. The special attention has been paid lately to the YSZ-based sensors using single-type and spinel-type oxide SEs due to their high sensing performances to various gaseous pollutants in oxygen containing atmospheres at high temperatures. Table 1 provides typical examples of characteristics of the mixed-potential type YSZ-based gas-sensing-devices using oxide SE reported by the authors to date. For instance, we have extensively investigated several oxides as SEs for  $NO<sub>x</sub>$  sensing, such as CdMn<sub>2</sub>O<sub>4</sub> [4], CdCr<sub>2</sub>O<sub>4</sub> [5], WO<sub>3</sub> [6], NiCr<sub>2</sub>O<sub>4</sub> [7], ZnFe<sub>2</sub>O<sub>4</sub> [8] and ZnCr<sub>2</sub>O<sub>4</sub> [9,10], since our first report in 1996 [3] suggested that the use of oxide SE in this type of  $NO<sub>x</sub>$  sensor is very effective for sensitive detection of  $NO<sub>2</sub>$  at high temperatures. Several papers [14-18] dealing with oxides as a sensing electrode of mixed-potential-type  $NO<sub>x</sub>$  sensor have also been published by several research groups after our first report in 1996. Specifically, we investigated the zinc-family oxides as SEs for  $NO_x$  sensors [8-10]. Consequently, we have found that, among the single-type oxides examined, the ZnO-SE is most stable and gives the highest sensitivity to  $NO_x$  at temperature as high as 700 °C. Therefore, in this study the main sensing properties of the YSZ-based mixed-potential NO<sub>x</sub> sensor using ZnO-SE have been investigated.

#### **2. Experimental Description**

Commercial half-opened YSZ tubes  $(8 \text{ mol.}\% \text{ Y}_2\text{O}_3)$ doped, NKT Co., Ltd.) were used for fabrication of  $NO<sub>x</sub>$ sensors. The tubes were 300 mm in length and 5 and 8 mm in inner and outer diameter, respectively. Commercially available oxide powders were applied on the outer surface of the YSZ tube and then sintered at  $1200$  °C for 2 h to form the SE. The Pt paste was applied on the inner surface of the YSZ tube and then sintered at  $1200$  °C for 2 h to form the counter electrode (CE). The sensing performances were evaluated in a conventional gas-flow apparatus in the temperature range of 600-700  $^{\circ}$ C from 40 ppm up to 450 ppm of  $NO<sub>x</sub>$  concentration. The ppm refers to the volume concentration of  $NO<sub>x</sub>$  in the gaseous mixture. The flow rate of the sample gas (or base air) was fixed at 100 cm3/min. The difference in potential (EMF) between SE and CE was measured by means of a digital electrometer (Advantest, R8240) as a sensing signal of the mixed-potential-type sensor when SE was exposed to the base air or to the sample gas with different  $NO<sub>x</sub>$  concentrations. The CE was always exposed to the atmospheric air during the experiments.

Gas	Oxide sensing electrode	Operating temperature	Measuring concentrations	Year of publication	Referenc e number
		$(C^{\circ}C)$	(ppm)		
	CdMn <sub>2</sub> O <sub>4</sub>	$500 - 600$	$5 \sim 4000$	1996	[3,4]
	CdCr, O <sub>4</sub>	$500 - 600$	$20 \sim 600$	1997	$\lceil 5 \rceil$
NO,	WO.	$500 - 700$	$5 \sim 200$	2000	[6]
	Nicr, O <sub>a</sub>	$550 - 650$	$15 \sim 500$	2001	[7]
	ZnFe <sub>2</sub> O <sub>4</sub>	$550 - 700$	$20 \sim 500$	2001	[8, 9]
	$ZnCr_2O_4$	$550 - 650$	$20 \sim 500$	2002	[10]
$\rm CO$	$CdO+SnO,$	600	$20 \sim 4000$	1997	$[11]$
$H_2S$	$Au - WO2$	400	$0.6 - 50$	1994	[12]
$C_3H_6$	CdO	600	$30 \sim 800$	2000	[13]

Table 1. Typical examples of sensing characteristics of the mixed-potential YSZ-based devices using oxide-SE reported by us.



Fig. 2. SEM images of the surface of YSZ substrate (a), Pt electrode layer (b), ZnO layer sintered at 800 °C (c) and ZnO layer sintered at  $1200$  °C (d).

The crystal structure and the surface state of various sensor materials used here were investigated by means of  $XRD$  analysis (with  $CuK\alpha$  radiation) and SEM observation. XRD measurements were carried out with a RIGAKU X-ray diffractometer (RINT 2100VLR/PC). SEM observation was performed by using JEOL electron microscope (JSM-6340F) operating at 3.0 kV or 15.0 kV. The adsorption-desorption behavior of  $NO<sub>2</sub>$  and  $O<sub>2</sub>$  were examined using a temperature-programmed desorption (TPD) apparatus (Bell Japan Inc., TPD-1-AT). The oxide powder (0.3 g) was set in the quartz-glass cell of the TPD apparatus. The sample was heated up to 800  $^{\circ}$ C in an He gas stream and was exposed to pure  $O_2$  at this temperature, followed by cooling down to 50  $^{\circ}$ C in the O<sub>2</sub> stream. Then, the sample was treated with  $1000$  ppm  $NO<sub>2</sub>$  diluted with He at 50  $\degree$ C for 5 min for the NO<sub>2</sub> adsorption. The desorbed  $NO<sub>x</sub>$  and  $O<sub>2</sub>$  from the sample were detected with a chemiluminescence  $NO<sub>x</sub>$  analyzer (Yanako, ECL-88A) and

a QP-mass spectrometer, respectively, when the sample was heated up to 800  $\degree$ C at heating rate of 10  $\degree$ C/min in a He stream. The catalytic activity of each oxide (0.01 g) for the gas-phase decomposition of  $NO<sub>2</sub>$  was evaluated by using the TPD apparatus as well as the  $NO<sub>x</sub>$  analyzer in the temperature range of 200-700  $^{\circ}$ C.

# **3. Results and Discussion**

*3.1. Crystal Structure and Surface State of Sensor Materials.* All peaks in the XRD spectrum for ZnO annealed at  $1200$  °C could be assigned to ZnO (JCPDS 22-1010). Figure 2 shows SEM images of the surfaces of YSZ substrate *(a),* Pt electrode *(b),* ZnO layer sintered at 800 °C  $(c)$  and ZnO layer sintered at 1200 °C  $(d)$ . The surface of YSZ substrate has no open porosity and a grain is sized from 1  $\mu$ m to 3  $\mu$ m. The Pt electrode was porous with an average particle size of about  $1 \mu m$ . The ZnO layer sintered at 800  $^{\circ}$ C was very porous with the average



Fig. 3. EMF values at 600  $^{\circ}$ C to both NO and NO<sub>2</sub> (200 ppm each) in air for the tubular YSZ-based sensors using each of several single-oxide SEs tested.

grain size in the  $0.2$ -0.6  $\mu$ m range. However, sintering of the ZnO layer at 1200  $^{\circ}$ C provided larger grain sizes from 0.5  $\mu$ m to 1  $\mu$ m in average. This result suggests that the thermal treatment brings about the structural stability and the low surface-to-volume ratio for ZnO, which is very important for good sensing characteristics of the sensor.

3.2. Sensing Performances of the NO<sub>x</sub> Sensor. The EMF in response to both NO and  $NO<sub>2</sub>$  concentrations (200 ppm each) in air for the tubular YSZ-based sensors using



Fig. 4. NO<sub>x</sub> sensing characteristics for the YSZ-based sensor using ZnO-SE in the tested temperature range of 600-700  $^{\circ}$ C.



Fig. 5. Response transients to  $NO<sub>2</sub>$  (a) and  $NO<sub>1</sub>$  (b) for the YSZ-based sensor using ZnO-SE at 700 °C.

several single-oxide SEs tested at 600 °C are shown in Fig. 3. The EMF value was close to zero in the carrier gas (dry synthetic air), so the measured EMF values were considered as the sensitivities to  $NO$  and  $NO<sub>2</sub>$ . As clearly shown in this figure, ZnO-SE gave the highest sensitivity to both  $NO$  and  $NO<sub>2</sub>$  among other single-oxides tested as SE. Figure 4 shows that the EMF response of the YSZbased sensor using ZnO-SE depends linearly on the logarithm of  $NO<sub>x</sub>$  concentration from 50 ppm up to 450 ppm in the tested temperature range of 600-700  $^{\circ}$ C. It is also evident that among the examined and previously published oxide SEs [3-10], ZnO gave one of the highest sensitivity to  $NO<sub>2</sub>$  in the above-mentioned temperature range: the EMF value of the present sensor to 50 ppm



Fig. 6. TPD profiles of NO<sub>2</sub> from ZnO,  $In_2O_3$  and  $Co_3O_4$ powder samples.

 $NO<sub>2</sub>$  was as high as ca. 40 mV even at 700 °C. In addition, the evaluation of sensing performances of the NOx sensors using  $ZnO-SE$  showed that the NO<sub>x</sub> sensitivities gave little degradation even at  $700 \degree C$  during the test period of about one month. The response transients to different concentration of both  $NO<sub>2</sub>$  (a) and to  $NO<sub>2</sub>$ (b) for the YSZ-based sensor using ZnO-SE at 700  $^{\circ}$ C are shown in Fig. 5. The 90% response time was within 60 s at 700  $\rm{^{\circ}C}$  for all NO<sub>x</sub> concentrations tested.

*3.3. Examination of Sensing Mechanism.* To understand the mechanism by which the ZnO-SE provides such a high  $NO<sub>2</sub>$  sensitivity at high temperatures, various properties of the representative oxides (ZnO,  $In_2O_3$ , Co<sub>3</sub>O<sub>4</sub>), such as gas adsorption-desorption behavior, oxygen



Fig. 7. Temperature dependence of NO<sub>2</sub> conversion to NO for the gas-phase decomposition reaction on the various oxides tested.



Fig. 8. Correlation between EMF value (NO<sub>2</sub> sensitivity) and NO<sub>2</sub> conversion for the three oxides tested.

sensing characteristics, and the catalytic activity for the gas-phase decomposition reaction of  $NO<sub>2</sub>$ , have been examined.

The TPD profiles of  $NO_2$  from ZnO,  $In_2O_3$  and  $Co_3O_4$ samples, shown in Fig. 6, reveal that the amount of  $NO<sub>2</sub>$ desorption from  $Co<sub>3</sub>O<sub>4</sub>$  is large and comprable with that from In<sub>2</sub>O<sub>3</sub>. Moreover, the desorption peaks for Co<sub>3</sub>O<sub>4</sub> occur in the relatively low temperature range of 100-300  $^{\circ}$ C and In<sub>2</sub>O<sub>3</sub> also has a peak at 100-300  $^{\circ}$ C. In the case of ZnO and  $In_2O_3$ , the NO<sub>2</sub> desorption peaks appear at higher temperature of about 450  $^{\circ}$ C. This suggests that the NO<sub>2</sub> gas adsorbed at the YSZ/ZnO-SE and at the YSZ/In<sub>2</sub>O<sub>3</sub>-SE interfaces may promote the rate of the following cathodic reaction (1) of  $NO<sub>2</sub>$  at high temperatures:



Fig. 9. Graphical explanation of the influence of catalytic activity of oxide SE on the actual concentration of  $NO<sub>2</sub>$  at the interface.



Fig. 10. TPD profiles of  $O_2$  from ZnO,  $In_2O_3$  and  $Co_3O_4$  pow-der samples.

$$
NO2 + 2e^- \rightarrow NO + O2-
$$
 (1)

In spite of high desorption rate at relatively high temperature for  $In_2O_3$ , the NO<sub>x</sub> sensitivity is still low for this oxide at  $600 \degree C$ , as already gives in Fig. 3. Therefore, we examined the catalytic activity of these oxides tested for  $NO<sub>2</sub>$  decomposition reaction. Figure 7 shows the  $NO<sub>2</sub>$ conversion to NO due to the following non-electrochemical gas-phase reaction.

$$
NO_2 \rightarrow NO + 1/2O_2 \tag{2}
$$

In the gas mixture of 100 ppm  $NO_2$ , 21 vol.%  $O_2$  and He balance, the  $NO<sub>2</sub>$  conversion on  $In<sub>2</sub>O<sub>3</sub>$  is higher than that on ZnO at higher temperatures. Since NO, thermodynamically, dominates in the equilibrated  $NO<sub>x</sub>$  gas mixture at temperatures above 500 °C [4], the conversion of  $NO<sub>2</sub>$  to NO is usually high when the effective catalysts are used. This indicates that the catalytic activity of ZnO is rather low, compared to those of other two oxides. Figure 8 shows that the catalytic activity for the oxides is roughly correlated to the  $NO<sub>2</sub>$  sensitivity obtained; the lower the catalytic activity of oxide, the higher the  $NO<sub>2</sub>$  sensitivity of the sensor. Figure 9 shows schematically the influence of catalytic activity of oxide on the actual concentration of  $NO<sub>2</sub>$  at the oxide/YSZ interface. If the catalytic activity of SE is reasonably high, most of  $NO<sub>2</sub>$  can be easily converted to NO accordingly to the gas-phase reaction (2) on the surface or in the bulk of oxide SE layer and thus it is rather difficult for  $NO<sub>2</sub>$  to arrive at the YSZ/SE interface



Fig. 11. Dependence of EMF on the logarithm of  $O<sub>2</sub>$  concentration for the YSZ devices using each of SEs at 700 °C.

(in the case of  $Co<sub>3</sub>O<sub>4</sub>$ ). As a result, the NO<sub>2</sub> sensitivity is low for the device using such an SE. Conversely, if the catalytic activity of the SE is rather low,  $NO<sub>2</sub>$  can diffuse through the SE layer without serious decomposition to NO, and then almost all the  $NO<sub>2</sub>$  can reach the YSZ/SE interface, resulting in high  $NO<sub>2</sub>$  sensitivity (in the case of ZnO). When the catalytic activity is moderate, moderate NO<sub>2</sub> sensitivity is obtained (in the case of  $In_2O_3$ ).

The investigation on  $O_2$  desorption has revealed that the oxygen adsorbed on the oxide SE also plays a significant role in the sensing mechanism involving mixed potential. Figure 10 shows the TPD profiles for oxygen desorption from the oxides tested. The desorption peaks



Fig. 12. Schematic polarization curves for the cathodic reaction (1) of  $NO<sub>2</sub>$  and anodic reaction (4) of  $O<sub>2</sub>$ .

for both  $Co<sub>3</sub>O<sub>4</sub>$  and  $In<sub>2</sub>O<sub>3</sub>$  are significantly higher than that for ZnO. This suggests that the catalytic activity of the ZnO-SE for the electrochemical reaction involving  $O_2$  is low.

To confirm this assumption, the oxygen sensing properties of the devices using  $ZnO$  and  $In<sub>2</sub>O<sub>3</sub>$  were evaluated. For this purpose, the EMF values of the YSZbased devices were measured when the oxagen concentration in the gas mixture  $(N_2+O_2)$  was changed from 1 to 100 vol.% at 700 °C. Figure 11 shows the Nernstian plots at 700  $^{\circ}$ C for the devices. As a comparison, the data for the sensor using the Pt-paste electrode were also given in this figure. The ZnO-SE provided smaller Nernstian slope (36 mV/decade) than the theoretical value (48 mV/decade,  $n = 4.0$ ). On the other hand,  $In_2O_3$ -SE and Pt-SE gave the theoretical Nernstian slope. This indicates that the ZnO-SE is working as an irreversible oxygen electrode and then the catalytic activity of the ZnO-SE for the electrochemical reaction (3) of oxygen is low as expected. Thus, such a low electrochemical catalytic activity may also contribute to the high sensitivity to  $NO<sub>2</sub>$ of the present sensor at high temperatures.

$$
O_2 + 4e^- = 2O^{2-}
$$
 (3)

Figure 12 shows the influence of electrochemical catalytic activity for both cathodic (1) and anodic (4) reactions on polarization curves, which schematically explains how the mixed potential works. The mixed po-



Fig. 13. Modified polarization curves of the YSZ sensor using ZnO-SE.

Table 2. Comparison between estimated mixed-potential values and observed EMF values for the YSZ-based device using ZnO-SE at 700 °C.

	$200$ ppm $NO$	$200$ ppm $NO2$		
$-$ mi	EMF	∸m2	<b>EMF</b>	
$-15$ mV	$-15$ mV	$+68$ mV	$+65$ mV	

$$
20^{2-} \to O_2 + 4e^-
$$
 (4)

tential  $(E_m)$  can be defined at the intersection of cathodic and anodic polarization curves, where the absolute values of cathodic and anodic currents are equal to each other. Both electrochemical reactions (1) and (4) proceed simultaneously at same reaction rate (same current density) at this potential  $E_m$ . If the catalytic activity for cathodic reaction (1) of  $NO<sub>2</sub>$  for oxide tested is high, the polarization curve for NO<sub>2</sub> shifts upward. Consequently, it changes the mixed potential to the direction of positive potential  $E_{m1}$  and increases the NO<sub>2</sub> sensitivity ( $\Delta E_1$ ). Alternatively, if the catalytic activity for anodic reaction of  $O<sub>2</sub>$  (4) is low, the anodic polarization curve for oxygen shifts downwards. The mixed potential, therefore, also changes to the direction of positive potential  $(E_{m2})$  and gives the increase in  $NO_2$  sensitivity ( $\Delta E_2$ ). For further verification of the mixed-potential model for YSZ-based  $NO<sub>x</sub>$  sensor using the ZnO-SE, the polarization curves were measured for the ZnO-SE under exposure to both NO



Fig. 14. EMF response (gas sensitivity) to various gases (400 ppm each) at 700  $^{\circ}$ C for the YSZ-based device attached with ZnO-SE.



Fig. 15. Cross-sectional view of the laminated YSZ-based mixed-potential NO, sensor.

and  $NO<sub>2</sub>$  (200 ppm each). The estimated mixed-potential values and the observed EMF values were in good agreement with each other (see Fig. 13 and Table 2), which confirms the mixed-potential model previously reported [5,6].

From the practical point of view, NO should be electrochemically oxidized forcibly to  $NO<sub>2</sub>$ , since the response direction to NO of the present mixed-potential sensor is opposite to that to  $NO<sub>2</sub>$ . As previously proposed [19], installation of an NO conversion electrode would allow the total  $NO<sub>x</sub>$  content to be measured by the present  $NO<sub>x</sub>$  sensing system using the ZnO-SE for detecting  $NO<sub>2</sub>$ . In addition, the change in  $O<sub>2</sub>$  concentration in the sample gas can be compensated for by means of an  $O_2$  sensing electrode (Pt) which is installed near the oxide SE. Furthermore, as shown in Fig. 14, the present YSZ-based  $NO<sub>x</sub>$  sensor attached with the ZnO-SE have cross-senitivities to combustible gases which usually exist in real car-exhausts. Thus, the combustible gases should be oxidized to  $CO_2$  and  $H_2O$  before reaching the ZnO-SE, by using an oxidation catalyst. These functioning devices can be combined into a planar laminated-type  $NO<sub>x</sub>$  device, as reported previously [19,20]. Therefore, in the laminatedtype sensor, combustible gases can be first oxidized by the

oxidation catalyst before they reach the SE and NO can be oxidized to  $NO<sub>2</sub>$  by the conversion electrode. Then, the change in  $O_2$  concentration around the SE can be compensated for by the oxygen sensor. Consequently, the total  $NO<sub>x</sub>$  content can be monitored without the interference of combustible gases as well as the variation in  $O<sub>2</sub>$  concentration.

#### 4. **Conclusions**

The mixed-potential-type  $NO<sub>x</sub>$  sensors using each of several single-oxide SEs were fabricated and their sensing properties were investigated in the temperature range of 600-700 °C. The YSZ-based device using the ZnO-SE gave the highest sensitivity to  $NO<sub>x</sub>$  even at 700 °C among other single-oxide SEs tested here and reported to date. This suggests that ZnO is one of the promising candidate for SE of the practical  $NO<sub>x</sub>$  sensor working at high temperature. Gas adsorption-desorption behavior, chemical and electrochemical catalytic activities, and other results suggest that the  $NO<sub>2</sub>$  sensitivity can be influenced by these factors in a complex manner. For the practical application, however, a laminated-type multi-component structure for the mixed-potential YSZ-based  $NO<sub>x</sub>$  sensor should be used. The main advantage of this design

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is that the total  $NO<sub>x</sub>$  content can be measured and the NOx sensitivity can be protected from the influence of the co-existing combustible gases as well as from the deviation of oxygen concentration in exhaust gas.

# **5. Acknowledgements**

This work was partially supported by a Grant from the Ministry of Education, Science, Sports and Culture of Japan.

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*Manuscript rec. Feb. 23, 2004; acc. Apr. 15, 2004.*