**ENGINEERING FORMULAE FOR FATIGUE** STRENGTH REDUCTION DUE TO CRACK-LIKE **NOTCHES** 

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Among fatigue engineers, it is well known that the fatigue strength reduction factor  $K_f$  is lower than the elastic stress concentration factor  $K_{\perp}$ . This discrepancy means that the highest stress alone is no longer appropriate for characterizing the microprocess of fatigue occurring in the microstructure at the notch tip. To rectify this microstructural size effect, Neuber [1-3] has hypothesized that the controlling fracture parameter is the mean stress over the structural size ahead of the notch tip. On the other hand, Ishibashi [4] and Peterson [5] postulated that the controlling factor is the stress at the distance of the structural size ahead of the notch tip. Both Neuber [i] and Peterson [5] started with the stress distribution for deep notches and derived the following approximate formulae of the  $K_{\mathcal{L}}-K_{\mathcal{L}}$  relationship which were claimed to be applicable to various notches:

$$
K_{f} = 1 + (K_{f} - 1)/(1 + \sqrt{\rho_{*}/\rho})
$$
 (1)

$$
K_{\rm f} = 1 + (K_{\rm f} - 1)/(1 + r_{\rm x}/\rho)
$$
 (2)

where  $\rho$  is the notch-tip radius, and  $\rho_{\star}$  and  $r_{\star}$  are the material constant. Both  $\rho_{*}$  and  $r_{*}$  have been correlated experimentally to the ultimate tensile strength  $[5,6]$  or the yield strength  $[3]$ . McEvily and Groeger [7] proposed to use Neuber's formula (I) to explain the growth threshold of small fatigue cracks. Since  $K_t = 1 + 2\sqrt{a/\rho}$  for an elliptic notch of length 2a in an infinite plate under tension, (1) reduces to

 $K_f = 1 + 2\sqrt{a/\rho}_*$  (3)

Peterson's formula is not applicable because it yields  $K_r = 1$  for  $\rho$  = 0. Different formulae can be obtained if both Neuber's and Ishibashi-Peterson's hypotheses are applied to the crack stress field.

The distribution of stress  $\sigma_{\rm u}$  on the crack plane (y=0) for a crack subjected to remote tensile stress  $\sigma$  as shown in Fig. 1 is

$$
\sigma_{y} = \sigma |x| / \sqrt{(x^2 - a^2)}
$$
 (4)

By taking local coordinate,  $x^{\dagger}$  and  $y^{\dagger}$ , at the crack tip, the mean stress  $\sigma_{\rm o}$  over the structural size  $\ell_{\rm o}$  is determined by

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$$
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$$

$$
\overline{\sigma}_o = \frac{1}{\ell_o} \int_a^{a+\ell_o} \sigma_y dx = \frac{1}{\ell_o} \int_o^{\ell_o} \sigma_y dx
$$
 (5)

and the stress  $\sigma_0$  at a distance  $\ell_0$  from the notch tip is

$$
\sigma_{0} = \sigma(\mathbf{x}^{1}) \mid \mathbf{x}^{1} = \mathbf{R}_{0}
$$
 (6)

Since  $\sigma$   $\,$  or  $\sigma$  is interpreted as the fatigue limit of smooth specimens  $\sigma_{_{\rm WO}}$  K<sub>f</sub> is obtained by

$$
K_{f} = \overline{\sigma_{0}}/\sigma \quad \text{or} \quad \sigma_{0}/\sigma \tag{7}
$$

depending on the hypothesis adopted. The stress intensity factor at the fatigue threshold is given by

 $K = \sigma \sqrt{\pi} a$  (8)

As shown below,  $\sigma$  approaches to the fatigue limit of smooth specimens as a crack becomes small; K becomes constant for long cracks. The constant value is the threshold Stress intensity factor K obtained by ordinary fracture mechanics tests. Smith  $[8]$  introduced the concept of the intrinsic crack length a defined by

$$
a_o = (K_o / \sigma_{wO})^2 / \pi
$$
 (9)

The equations for K<sub>f</sub> and K/K<sub>s</sub> obtained from (4) by using Neuber's form-<br>ula (1), Neuber and Ishibashi-Peterson hypotheses are summarized in Table 1, together with the a value as a function of <sub>P, c</sub>or  $\ell$  . For a<br>crack subjected to antiplane shear (mode III), a similar calculation can be done with shear stress  $\tau_{\scriptscriptstyle{1.5}}$  on the crack plane. The stress distribution is given from (4) by changing  $\sigma$  and  $\sigma$  to  $\tau$  - and  $\tau$ , respectively. K for an elliptical notch is  $1+\forall a/\rho$ . The final expressions of  $K_{\varepsilon}$  and K/K  $\,$  versus a  $\,$  are the same as in tensile (mode I) loading except  $a_0 = \rho_*$  in (10) and (11), where  $a_0$  is defined by

$$
a_0 = (K_{\infty}/\tau_{\text{WO}})^2/\pi
$$
 (10)

(K : the threshold stress intensity factor for mode II:  $_{\tau}$  : the fatigue limit under antiplane shear or torsion).

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Figure 2 shows the relation between  $1/K_f = \sigma/\sigma$  and  $a/a$ . The curves of (13) and (15) are very close, while that of (11) is far below the two curves. The dashed lines are limiting lines corresponding to constant stress and stress intensity factor. Eqn. (13) was first derived by Hadded et al. [9] from a different consideration and has been confirmed as an accurate relation to express the data of various metals [9,10]. They derived (13) by equating the stress intensity factor for the small crack of a fictitious length of a plus a to K for long cracks at the threshold. Eqn. (11) is far too conservative. Some other analyses [10-12] based on the micromechanisms of the fatigue process yielded the relation very close to (13). Because of simplicity, (13) is recommended in engineering applications. The material microstructure comes into the relation through a which is determined from the data on K and <sub>σ</sub> by (9). When the data of a is not available,<br>its estimation can be made from <sub>p.</sub> where data are relatively abundant [3,6].

The same engineering hypotheses will be applied to deep notches with small tip radii. The distribution of stress  $\sigma_{\perp}$  near the tip of  $\,$ deep notches under tension (mode 1) as shown in Fig. 3 is obtained by Creager [12] as

$$
\sigma_{\mathbf{y}} = \frac{2\mathbf{K}}{\sqrt{\pi}} \frac{\rho + \mathbf{x'}(\rho + 2\mathbf{x'})^3 / 2} \tag{17}
$$

 $(y=0, x'-0)$ , where  $K_0$  is the stress intensity factor for a crack with identical dimensions to a notch except  $\rho$ . The maximum stress at the notch tip is

$$
(\sigma_{\rm y})_{\rm max} = 2K_{\rm \rho}/\sqrt{(\pi_{\rm \rho})}
$$
 (18)

By substituting (17) into (5) and (6),  $K_{\rho}$  can be determined as a function of the fatigue limit  $\sigma_{\text{w0}}(\sigma, \sigma)$  and  $\chi$ . As shown below,  $K\rho$  be-<br>comes a certain value at  $\rho=0$ , denoted by  $K^0$ , and  $(\sigma y)_{\text{max}}$  approaches<br> $\sigma_{\text{w0}}$  as  $\rho$  becomes large.  $\rho$  at the intersection of

$$
\rho_0 = (2K_0 / \sigma_{\text{WO}})^2 / \pi \tag{19}
$$

The relations between K<sub>p</sub>/K and p/p obtained by (1), (5), and (6) are<br>summarized in Table 2, together with <sub>p</sub> . For the case of antiplane shear (mode III) loading, the distribution of stress  $\tau_{yz}$  on y=0 is  $\tau_{yz}$ given by Creager [12] as

$$
\tau_{yz} = \frac{K}{\sqrt{\pi}} \frac{1}{(\rho + 2x^{\prime})}
$$
 (20)

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The maximum stress at the notch tip is

$$
\left(\tau_{yz}\right)_{\text{max}} = \frac{1}{2} \left(\sqrt{(\pi \rho)}\right) \tag{21}
$$

The same procedure is employed to calculate the relation between  $K_p/K$ and  $\rho/\rho_{\rm o}$ , where  $\rho_{\rm o}$  is determined by

$$
\rho_0 = (K_0 / \tau_{wo})^{2/\pi} \tag{22}
$$

 $\mathbf{o}$ 

The relations obtained are summarized in Table 2 with  $\rho$ .

In Fig. 4, K<sub>D</sub>/K is plotted against  $\sqrt{(p/\rho)}$ . The curve of (20) obtained from Neuber's formula is above all other lines. The curves of (21), (22), (26) are close. Experimental data of the threshold of fatigue crack initiation in polymer [14] and steels [15,16] indicated that K<sub>p</sub> increased proportionally to  $\sqrt{\rho}$  when  $\rho$  was large. As  $\rho$  becomes small,  $K<sub>0</sub>$  deviates from the proportional relation approaching to a constant value [13,15]. This transitional behavior was found to be best approximated by (21) in the experiment of low carbon steel [16]. (20) gives a dangerous estimate while (22) is conservative. A micromechanical model for fatigue crack initiation by Tanaka and Mura [12] also gives the relation close to (21). Several other proposed formulae [5, 16-18] can be proved to yield (21) for the cases of deep notches. Among (21), (22), and (26), (21) is recommended for simplicity. This equation has a simple interpretation. If the real notch is assumed to have a fictitious notch tip radius equal to  $\rho$  plus  $\rho_0$ , the fatigue limit is determined by the condition that the maximum stress at the fictitious notch tip equals the smooth specimen fatigue limit. The material parameter  $\rho_{\alpha}$  is obtained from (19) or (25) and can be estimated by  $\rho$ , which appears in Neuber's formula (l). If one assumes that K is identical to K , it is obtained from (9), (10), (19), (25)<br>that  $\int_{0}^{0}$  is equal to 4a for tensile (mode I) loading to a for torsional (mo $\overline{d}$ e IV) loading. <sup>O</sup>Experimental works are underway to check those relations and also to examine the applicability of the fictitious notch tip concept to the fracture toughness and stress corrosion tests. *REFERENCES* 

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Table I. Effect of crack length on fatigue strength reduction		
		Neuber's Ishibashi-Peterson's
Neuber's formula	hypothesis	hypothesis
		$K_{\epsilon}$ 1+ $\sqrt{a/a}$ (11) $(1+a/a_0)^{\frac{1}{2}}$ (13) $(1+2a/a_0)/(1+4a/a_0)^{\frac{1}{2}}$ (15)
		K/K $[1+\sqrt{a}]{a}]^{-1}$ (12) $(1+a)(a)^{-\frac{1}{2}}$ (14) $(4+a)(a)^{\frac{1}{2}}/(2+a)(a)$ (16)
$a_{\alpha}$ $\rho_{\star}/4$ (mode I)		
$\int_{*}^{\rho}$ (mode III)	$\frac{20}{3}$ 22.	

Table 1. Effect of crack length on fatigue strength reduction

	Neuber's formula	Neuber's hypothesis	Ishibashi-Peterson's hypothesis
Mode I			$K_{\rho}/K_{\rho}$ 1+ $\sqrt{\rho/\rho_{0}}$ (20) $(1+\rho/\rho_{0})^{\frac{1}{2}}$ (21) $(1+4\rho/\rho_{0})^{3/2}/(1+8\rho/\rho_{0})$ (22)
$P_{\rm o}$	$\rho_{*}$	$2\ell_{\rm o}$	$8\ell_{\rm o}$
Mode III	Ko/K <sub>o</sub> $1+\sqrt{\rho/\rho_o}$ . (20) $(1+\rho/4\rho_o)^{\frac{1}{2}}$ + ( $\rho/4\rho_o$ ) <sup>2</sup> (26)		$(1+\rho/\rho_0)^{2/3}$ $\ldots$ (21)
$P_{\rm o}$	$\rho_{\star}$	$\frac{2}{2}$ /2	$2\ell_{\Omega}$

Table 2. Effect of notch-tip radius on fatigue strength reduction (deep notches)



Figure 1. Crack in an infinite plate.



Figure 2. Effect of crack length on fatigue strength reduction.



Figure 3. Deep notch tip.



Figure 4. Effect of notch-tip radius on critical stress intensity.