

# Achieving Low Resistance Ohmic Contacts to Transition Metal Dichalcogenides (TMDCs)



Anirudh Gajula, Wes Wen Jun Lee, Calvin Pei Yu Wong,  
and Kuan Eng Johnson Goh

**Abstract** Transition metal dichalcogenides (TMDCs) harbor great potential for use in high performance electronic devices. However, their practical usage has been hindered by a lack of suitable low resistance ohmic contacts, resulting in high contact resistances and low electron mobilities. Our study aims to investigate the performances of alternative contacts strategies such as exfoliated graphite contacts, bottom-up gold (Au) contacts and evaporated gold-capped indium (In-Au) contacts to exfoliated tungsten disulfide ( $WS_2$ ) by first fabricating field-effect transistors (FET) and later, conducting transfer line measurements (TLM). Our results show that evaporated gold-capped indium/ $WS_2$  contacts achieved the best ohmic performance out of the three contact strategies with a significantly higher field effect electron mobility of  $114 \text{ cm}^2\text{V}^{-1} \text{ s}^{-1}$  and a lower contact resistance of  $462 \text{ k } \Omega \mu\text{m}$  to few layer  $WS_2$  and a mobility of  $5.45 \text{ cm}^2\text{V}^{-1} \text{ s}^{-1}$  and contact resistance of  $169 \text{ M } \Omega \mu\text{m}$  to monolayer  $WS_2$  at room temperature, while graphite/ $WS_2$  contacts and bottom up Au/ $WS_2$  contacts yielded poor non-ohmic characteristics with a field effect electron mobility of 0.0409 and  $0.00542 \text{ cm}^2\text{V}^{-1} \text{ s}^{-1}$  respectively. Our results also show that low resistance ohmic contacts for  $WS_2$  can be achieved through the direct evaporation of gold-capped indium (In-Au) contacts. This is of current relevance and importance as  $WS_2$  has been found to have a plethora of applications from high mobility field-effect transistors to quantum information processing and the formation of the low resistance ohmic contact is a fundamental step towards achieving these goals.

---

A. Gajula (✉) · W. W. J. Lee  
School of Mathematics and Science, NUS High, Singapore, Singapore  
e-mail: [h1510040@nushigh.edu.sg](mailto:h1510040@nushigh.edu.sg)

W. W. J. Lee  
e-mail: [h1510150@nushigh.edu.sg](mailto:h1510150@nushigh.edu.sg)

C. P. Y. Wong · K. E. J. Goh  
Institute of Materials and Research Engineering (IMRE), Agency for Science, Technology and Research (A\*STAR), Singapore, Singapore  
e-mail: [calvin\\_wong@imre.a-star.edu.sg](mailto:calvin_wong@imre.a-star.edu.sg)

K. E. J. Goh  
e-mail: [gohj@imre.a-star.edu.sg](mailto:gohj@imre.a-star.edu.sg)

**Keywords** Electrical properties · Van der waals contacts · Nanomaterials · 2D materials · Transition metal dichalcogenides · Tungsten disulfide

## 1 Introduction

Tungsten disulfide ( $\text{WS}_2$ ), a transition metal dichalcogenide (TMDC), has recently shown great promise in the pioneering field of ultrathin 2D semiconductors. Bulk  $\text{WS}_2$  has an appreciable band gap of 1.32 eV, which increases to 1.8 eV for monolayer  $\text{WS}_2$  [1] and contributes significantly to its distinct electronic properties and potential to act as ultrathin transistors in various digital circuits. For example,  $\text{WS}_2$  has been shown to work as a high mobility field effect transistor [2, 3] and also shown promise in quantum information processing, with the tunable valley polarization that provides an opportunity to control the valley pseudospin [4].

Despite the great promise brought about by the advent of TMDCs, the field remains plagued by significant challenges in the largely unexplored field of solid-state physics. The contacts between 2-dimensional monolayer TMDCs and 3-dimensional metal contacts are known to have high contact resistance, which hinders their potential for practical usage in high-performance devices and microelectronics [5, 6]. Thus, some studies have attempted alternative methods, such as chemical doping methods using lithium fluoride or F4TCNQ to successfully reduce the contact resistance by an order of magnitude [7, 8]. However, it is not clear if these dopants are stable in the long term upon device fabrication.

Recently, it has been shown that weakly interacting van der Waals contacts to TMDC have low contact resistances [5, 9]. Hence, the objective of this study is to investigate van der Waals contacts to  $\text{WS}_2$  devices fabricated using 3 different strategies (1) bottom-up graphite contacts, (2) bottom-up gold (Au) contacts, and (3) top-down (evaporated) gold-capped indium (In–Au) contacts. We analyzed these fabrication methods using electron mobilities extracted from field effect transistor measurements and contact resistance extracted from the transfer line measurements. We found that the top-down gold-capped indium (In–Au) contacts resulted in the best device performance with an electron mobility of  $114 \text{ cm}^2\text{V}^{-1} \text{ s}^{-1}$  and a contact resistance of  $462 \text{ k } \Omega \mu\text{m}$  for few layer  $\text{WS}_2$ , and an electron mobility of  $5.45 \text{ cm}^2\text{V}^{-1} \text{ s}^{-1}$  and a contact resistance of  $169 \text{ M } \Omega \mu\text{m}$  for monolayer  $\text{WS}_2$ .

## 2 Experimental Design

The first method to achieve van der Waals contacts to  $\text{WS}_2$  involves the positioning of the exfoliated  $\text{WS}_2$  on top of the two graphite contacts, which are contacted using chromium palladium gold (Cr–Pd–Au) contacts. We expect to have a low contact resistance van der Waals contact between the  $\text{WS}_2$  monolayer and the graphite contact. We exfoliated  $\text{WS}_2$  monolayers onto Polydimethylsiloxane (PDMS) and transferred

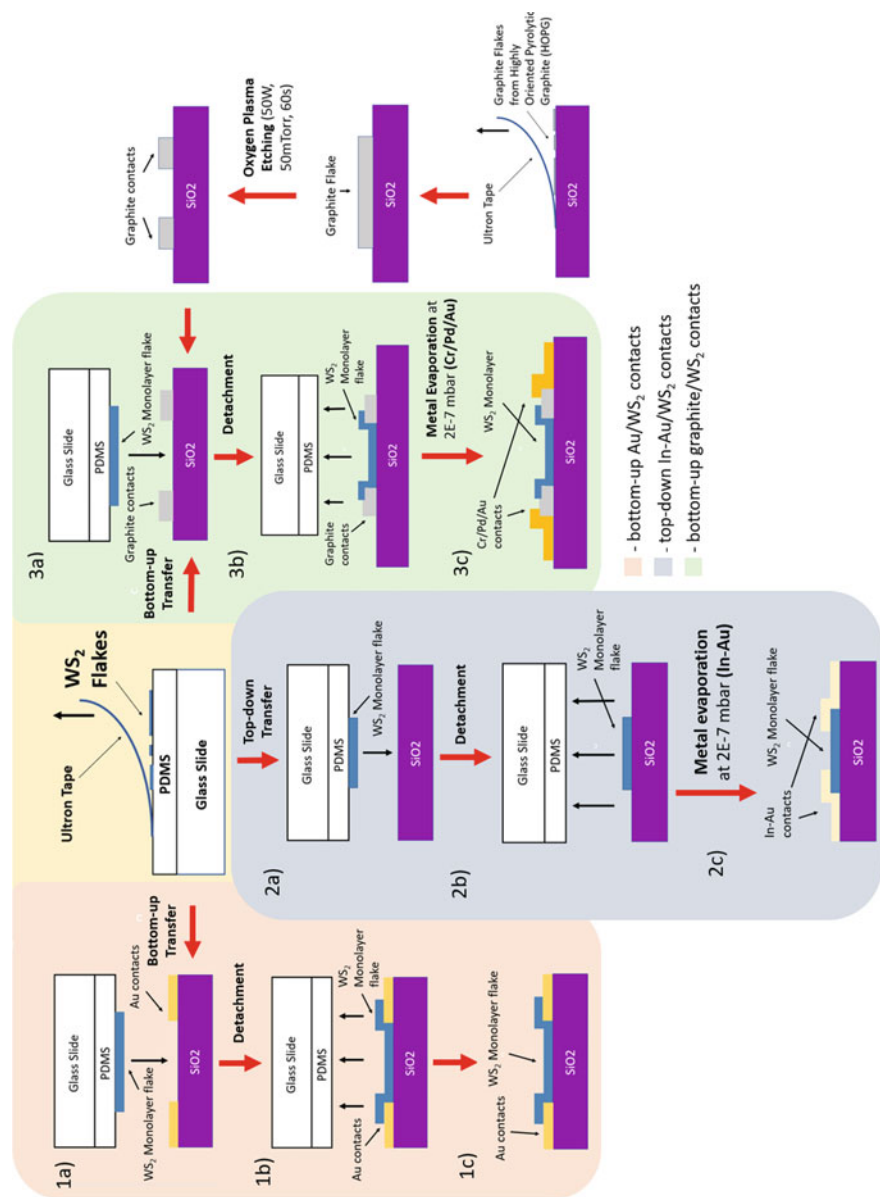
them onto the graphite contacts using a microscope transfer station. The second method involves the direct contact of the monolayer  $\text{WS}_2$  on the gold contacts, also using the microscope transfer station. The third method involves the direct metallization of gold-capped indium (In–Au) contacts onto the wafer, using In as a soft non-invasive contacting metal, which has been shown to have a lower contact resistance than directly evaporated titanium gold (Ti–Au) contacts [5]. The direct metallization of gold-capped indium (In–Au) contacts allows for a simple fabrication method for multiple contacts of different channel lengths for the transfer length measurements, which are difficult to achieve using bottom up contacts.

### 3 Methodology

Figure 1 above shows the device fabrication process for the three types of devices. The exfoliation of  $\text{WS}_2$  monolayer flakes was carried out using an adhesive tape (Ultron) onto a polymer layer of polydimethylsiloxane (PDMS, Gelpak X4) from a  $\text{WS}_2$  crystal (2D semiconductors, Inc). The identification of the  $\text{WS}_2$  monolayers was carried out with an optical microscope (Olympus BX60) using the optical contrast method. Graphite flakes were exfoliated from highly oriented pyrolytic graphite (SPI) onto a 285 nm  $\text{SiO}_2$  coated silicon wafer and similarly identified through an optical microscope (Olympus BX60).

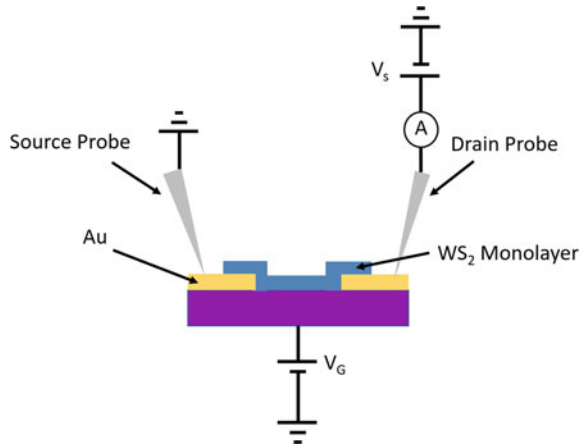
Contacts to the graphite flakes were fabricated using electron beam lithography (EBL). First, Polymethyl methacrylate (PMMA) A5 was spin coated onto the exfoliated flakes at 4000 rpm for 90 s. Next, the PMMA was patterned by electron beam lithography (EBL), using an ELS-7000 Elionix at 100 kV, and developed in a 1:3 ratio of methyl isobutyl ketone (MIBK) and isopropyl alcohol (IPA) for 70 s. Then, Cr–Pd–Au contacts were deposited using a Denton Explorer 14 e-beam evaporator at a base pressure of  $2\text{E}-7$  mbar. Lift off was carried out in acetone for 2 h. The graphite break junction of 5  $\mu\text{m}$  for contacting  $\text{WS}_2$  was fabricated by etching the exfoliated graphite flake. The exposed areas of the graphite flakes through the PMMA mask were etched off via oxygen-plasma etching at 50 W and 50 mTorr for 60 s. This created graphite contacts that were separated by a 5-micron gap.

The transfer of the  $\text{WS}_2$  flakes from the PDMS polymer to the silicon wafer or the graphite break junction was done through a PDMS dry transfer method [10], which involved the mounting of the glass slide containing the  $\text{WS}_2$  flakes onto a micromanipulator. The receiving substrate was then heated to a temperature of 80  $^\circ\text{C}$ , and the glass slide was carefully lowered down onto the silicon wafer to initiate the transfer process. Lastly, electrical measurements were carried out in a Janis vacuum probe station (pressure 5E-5 mbar), using two Keithley 2450 for the source-drain-gate measurements, as illustrated in Fig. 2.



**Fig. 1** A schematic diagram representation of the device fabrication process. Process 1, denoted by **a**, **b** and **c**, depicts the fabrication process of the bottom-up gold/WS<sub>2</sub> contacts. Process 2, denoted by **a**, **b** and **c**, depicts the fabrication process of the top-down gold-capped indium/WS<sub>2</sub> contacts. Process 3, denoted by **a**, **b** and **c**, depicts the fabrication process of the bottom-up graphite/WS<sub>2</sub> contacts

**Fig. 2** A schematic diagram representation of the electrical measurement process. Conventional current flows from the source probe to the drain probe



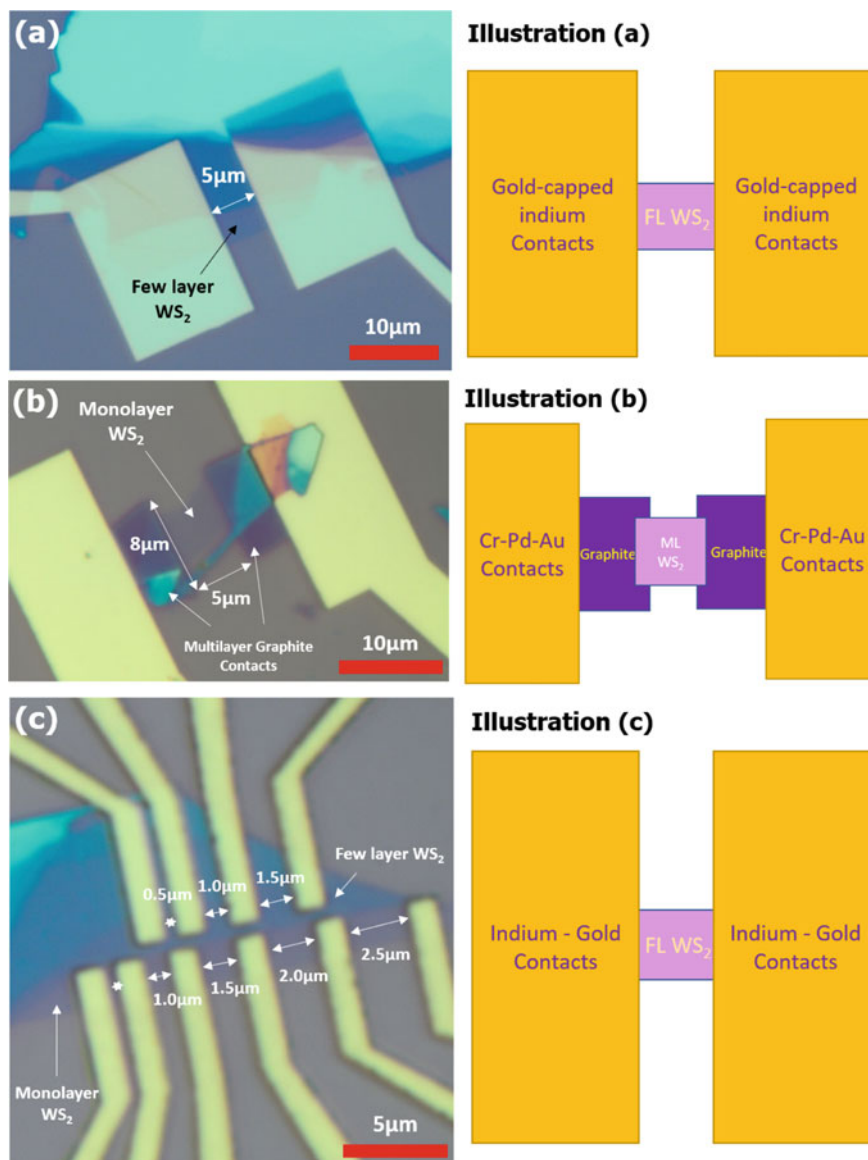
## 4 Results and Discussion

### 4.1 Optical Images

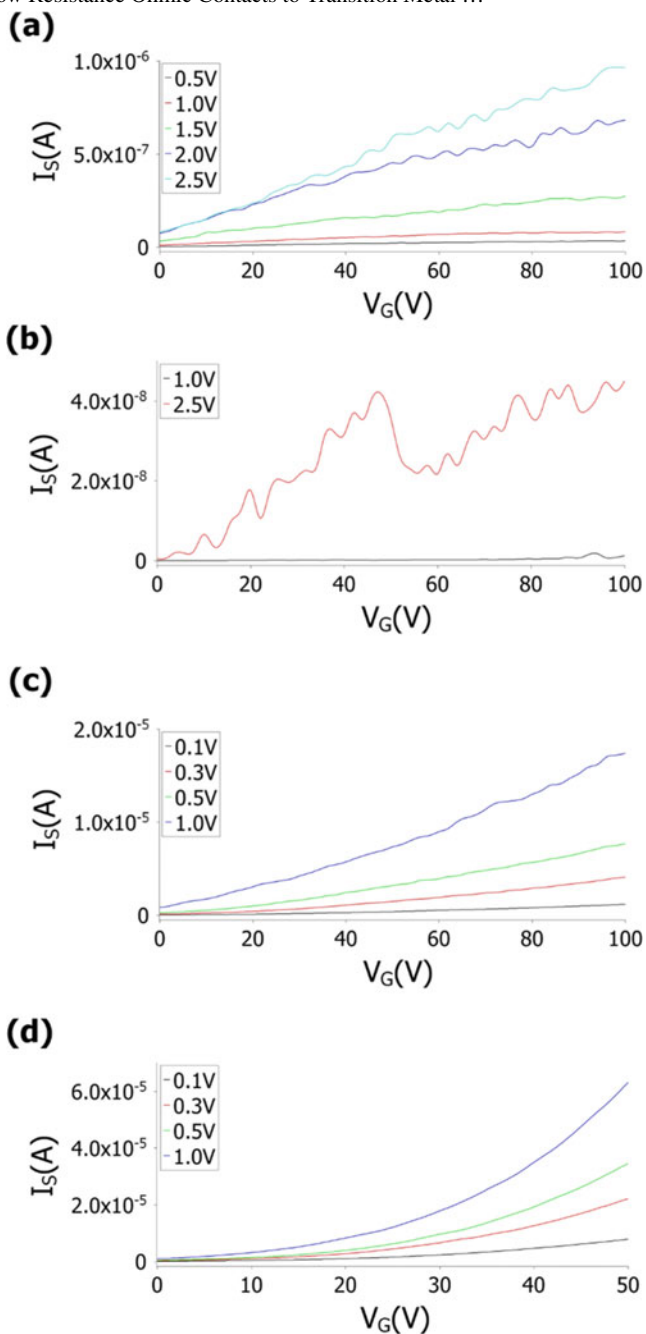
Figure 3 shows the optical micrographs of the fabricated devices. For the bottom-up gold (Au) contacts (Fig. 3a), the few layer exfoliated WS<sub>2</sub> was placed onto the contacts across the 5 μm gap, achieving a 2-terminal contact. The thickest regions of the WS<sub>2</sub> were not contacted, so current is flowing through the few layer regions in the device in Fig. 3a. For the bottom-up graphite contacts (Fig. 3b), the exfoliated monolayer WS<sub>2</sub> was placed across the 5 μm gap between the 2 graphite contacts, similarly achieving a 2-terminal contact. The top-down gold-capped indium (In-Au) contacts (Fig. 3c) are placed across multiple contacts exfoliated WS<sub>2</sub> to obtain the results for the transfer line measurement analysis.

### 4.2 Carrier Mobilities

Figure 4 shows the transfer curves ( $I_s-V_g$ ) of the fabricated devices, from which the field effect mobility is extracted from. The field effect mobility is extracted from (Eq. 1), where  $\mu_{FE}$  refers to the field-effect carrier mobility,  $L_{CH}$  refers to the channel length,  $g_m$  refers to the intrinsic transconductance,  $W_{CH}$  refers to the channel width,  $C_G$  refers to the gate capacitance per unit area, and  $V_s$  refers to the source voltage.  $C_G$  can be further calculated from  $C_G = \epsilon_0 \epsilon_r d_{ox}$ , where  $\epsilon_0$  is the permittivity of free space,  $\epsilon_r$  represents the relative dielectric constant of 3.8 for SiO<sub>2</sub>, and the thickness of the oxide layer,  $d_{ox} = 285$  nm.



**Fig. 3** Optical microscopy images of the devices. **a** Optical image of bottom-up gold contacts to WS<sub>2</sub> (gold/WS<sub>2</sub>). The WS<sub>2</sub> monolayer was 6µm by 12µm. **b** Optical image of bottom-up chromium palladium gold (Cr-Pd-Au) contacts to WS<sub>2</sub> (graphite/WS<sub>2</sub>). The WS<sub>2</sub> monolayer was 5 by 8 µm. **c** Optical image of evaporated gold-capped indium contacts to few layer & monolayer WS<sub>2</sub> (gold-capped indium/few layer WS<sub>2</sub> and gold-capped indium/monolayer WS<sub>2</sub>) for transfer line measurements. The lengths ranged from 0.5 to 2.5 µm for the WS<sub>2</sub> monolayer; the lengths ranged from 0.5 to 1.5 µm for the WS<sub>2</sub> few layer



**Fig. 4** Graphs of source current ( $I_S$ ) against gate voltage ( $V_G$ ) used to determine the field-effect mobilities for each of the devices. **a** Graph of  $I_S$  against  $V_G$  for gold/WS<sub>2</sub>. The extracted electron mobility was  $0.00542 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ . **b** Graph of  $I_S$  against  $V_G$  for graphite/WS<sub>2</sub>. The extracted electron mobility was  $0.0409 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ . **c** Graph of  $I_S$  against  $V_G$  for gold-capped indium/monolayer WS<sub>2</sub>. The extracted electron mobility was  $5.45 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ . **d** Graph of  $I_S$  against  $V_G$  for gold-capped indium/few layer WS<sub>2</sub>. The extracted electron mobility was  $114 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$

$$\mu_{FE} = \frac{L_{CH} g_m}{W_{CH} C_G V_S}. \quad (1)$$

The gate voltage induced carrier density,  $n_{2D}$ , was extracted using the parallel-plate capacitor model from (Eq. 2), where  $C_G$  refers to the gate capacitance per unit area,  $V_G$  refers to the gate voltage,  $V_{G,th}$  refers to the threshold voltage, and  $e$  refers to the unit charge (Fig. 4).

$$n_{2D} = \frac{C_G (V_G - V_{G,th})}{e}. \quad (2)$$

The electron mobilities were computed and compared at carrier densities of approximately  $2.2 \times 10^{12} \text{ cm}^{-2}$ . Thus, the derivative of the  $I_S$  against  $V_G$  curve ( $g_m$ ) was taken at  $V_G$  values which corresponded to the carrier density of  $2.2 \times 10^{12} \text{ cm}^{-2}$  for each of the devices. This is to conduct a valid comparison between the electron mobilities that were obtained for each of the devices.

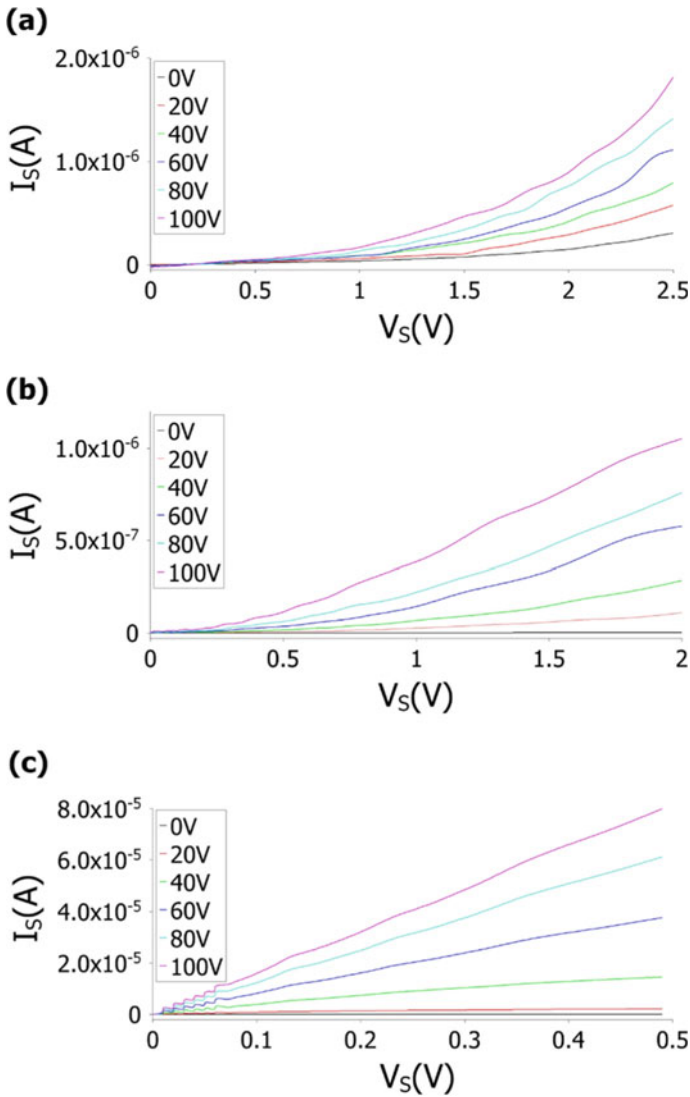
The gold-capped indium (In–Au) contacts to few layer  $\text{WS}_2$  yielded the highest electron mobility of  $114 \text{ cm}^2 \text{V}^{-1} \text{ s}^{-1}$ , 5 orders of magnitude larger than the electron mobility of the gold/ $\text{WS}_2$  contacts. Other studies have also found that evaporated gold-capped indium (In–Au) contacts offer higher electron mobilities than pure gold (Au) or gold-capped titanium (Ti–Au) contacts for TMDCs [5]. However, the graphite contacts to  $\text{WS}_2$  yielded lower than expected electron mobilities of  $0.0409 \text{ cm}^2 \text{V}^{-1} \text{ s}^{-1}$ , even though they were shown to form low resistance contacts to TMDCs [11]. Nonetheless, the electron mobility of  $\text{WS}_2$  with the graphite contact was nearly 10 times that of bottom-up gold (Au) contact even though the graphite/ $\text{WS}_2$  device is a monolayer device which has typically shown lower mobilities than the few layer  $\text{WS}_2$  devices. Since only one graphite contact was functional, of the 11 graphite contacts produced, fabrication of more graphite contacts to  $\text{WS}_2$  is required to confirm if graphite contacts indeed offer, on average, higher electron mobilities than gold (Au) contacts.

### 4.3 Contact Resistance Determination

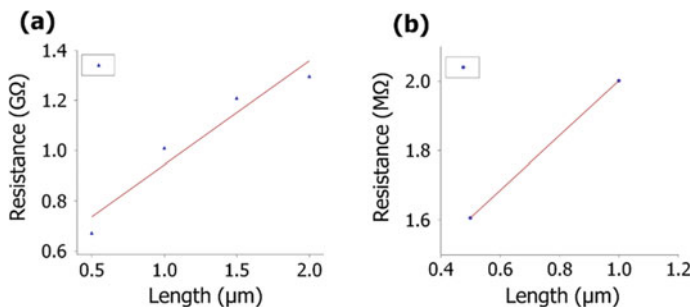
Figure 5 shows the output curves ( $I_S$ – $V_S$ ) of the bottom-up gold/ $\text{WS}_2$  contacts, gold-capped indium (In–Au) contacts to monolayer  $\text{WS}_2$ , and gold-capped indium (In–Au) contacts to few layer  $\text{WS}_2$ . The output curves of the gold/ $\text{WS}_2$  contacts (Fig. 5a) are not linear, from which we can conclude that the contact is not ohmic, and a large Schottky barrier exists at the interface. The output curves of (Fig. 5b) is approximately linear, implying that a small Schottky barrier exists for the gold-capped indium/monolayer  $\text{WS}_2$  contact. The linear output curves of gold-capped indium/few layer  $\text{WS}_2$  (Fig. 5c) shows that the contact is ohmic.

Using the transfer line measurement readings, the contact resistance and sheet resistance for the gold-capped indium (In–Au) contacts to  $\text{WS}_2$  were determined by





**Fig. 5** Graphs of source current ( $I_S$ ) against source voltage ( $V_S$ ) for devices at constant gate voltage ( $V_G$ ). **a** Graph of  $I_S$  against  $V_S$  for gold/WS<sub>2</sub>. It is evident that the gold/WS<sub>2</sub> contact does not display ohmic behavior, and a large Schottky barrier exists at the interface. **b** Graph of  $I_S$  against  $V_S$  for gold-capped indium/monolayer WS<sub>2</sub>. It is evident that the gold-capped indium/monolayer WS<sub>2</sub> contact displays slight Schottky behavior. **c** Graph of  $I_S$  against  $V_S$  for gold-capped indium/few layer WS<sub>2</sub>. The linear behavior shows that the gold-capped indium/few layer WS<sub>2</sub> contact displays extensive ohmic behavior



**Fig. 6** Graphs of resistance against channel length (obtained from transfer line measurements). **a** Graph of resistance against channel length for gold-capped indium/monolayer WS<sub>2</sub>. The extracted contact resistance was 169 M Ω μm. **b** Graph of resistance against length for gold-capped indium/few layer WS<sub>2</sub>. The extracted contact resistance was 462 k Ω μm

taking the limit of the trend lines in the graphs of Fig. 6a and Fig. 6b to a zero-length resistor. We obtain the contact resistances for the gold-capped indium/monolayer WS<sub>2</sub> and the gold-capped indium/few layer WS<sub>2</sub> as 169 M Ω μm and 462 k Ω μm respectively.

The contact resistance of 462 k Ω μm for the gold-capped indium/few layer WS<sub>2</sub> is much higher than those obtained in the latest studies, where Yan Wang et al. reported low contact resistances of 2.4 k Ω μm [5]. They used Ar/H<sub>2</sub> gas annealing to reduce the Schottky barrier and interface contamination. We did not subject our samples to Ar/H<sub>2</sub> annealing due to the lack of facilities. Hence the higher contact resistances could arise either from the current having to tunnel through a wider Schottky barrier or scattering by the contaminants.

Only 2 terminal measurements were taken for gold/WS<sub>2</sub> and graphite/WS<sub>2</sub> contacts as the WS<sub>2</sub> flakes obtained from exfoliation are typically very small, ranging from 10 to 20 μm in length. Since it is much more difficult to align flakes of such small structures under the microscope, this method was not attempted for gold/WS<sub>2</sub> and graphite/WS<sub>2</sub> contacts. Nonetheless, as we used WS<sub>2</sub> flakes that were exfoliated from the same bulk crystal, it can be inferred from the lower field-effect mobilities for gold/WS<sub>2</sub> and graphite/WS<sub>2</sub> contacts that gold/WS<sub>2</sub> and graphite/WS<sub>2</sub> have significantly higher contact resistances than gold-capped indium/WS<sub>2</sub>. Our 2D transfer yield for the bottom up graphite and gold (Au) contacts was poor (only 1 out of 10 devices work), showing the difficulty of this fabrication method. Although an experienced and well-trained practitioner can have higher fabrication yield, this method is not suitable for scaling up and the top down (evaporation) method is preferred.

## 5 Summary of Values

See Table 1.

**Table 1** Summary of values obtained for each device

Device	Electrical properties			
	Electron mobility ( $\text{cm}^2\text{V}^{-1}\text{s}^{-1}$ )	Contact resistance ( $\text{M}\ \Omega\ \mu\text{m}$ )	Sheet resistance ( $\text{M}\ \Omega$ )	Contact resistivity ( $\Omega\ \text{cm}^2$ )
Gold/ $\text{WS}_2$	0.00542	–	–	–
Graphite/ $\text{WS}_2$	0.0409	–	–	–
Gold-capped Indium/ $\text{WS}_2$ monolayer (ML)	5.45	169	580	2.36
Gold-capped Indium/ $\text{WS}_2$ few layer (FL)	114	0.462	1.11	0.00647

## 6 Future Work

To further improve the reliability of the results, the monolayer  $\text{WS}_2$  and few layer  $\text{WS}_2$  should be etched into rectangular strips, so that the widths and lengths are well defined. This would also enable the isolation of the monolayer regions from the few later  $\text{WS}_2$  and ensure that current is passing through the monolayer  $\text{WS}_2$  or few layer  $\text{WS}_2$  only. Next, more data points should be collected for the transfer line measurements of gold-capped indium/few layer  $\text{WS}_2$ . In addition, temperature dependent measurements can be carried out to determine the Schottky barrier height, which would further validate our claim that In-Au contacts offer the lowest contact resistance and highest mobilities of the 3 methods proposed. Lastly, contact resistances and carrier mobilities may be improved by conducting a hydrogen annealing step together with h-BN capping [5, 12].

## 7 Conclusion and Applications

We have shown that low resistance ohmic van der Waals contacts for the TMDC  $\text{WS}_2$  can be achieved using the direct metallization of gold-capped indium (In-Au) contacts onto  $\text{WS}_2$ . In contrast, the transfer of the few layer  $\text{WS}_2$  on pure gold (Au) contacts or graphite contacts produced poorer contacts, which we attribute mainly to gold (Au) having a higher work function than Indium (In) by 1.38 eV, or by the presence of more contamination on the gold (Au) or graphite surface. Our findings are of current relevance and importance as electrical metal contacts formed are known to strongly affect device performance and the successful development of a low contact resistance van der Waals contact is imperative for the next-generation development of energy-efficient electronics and opto-electronics applications of  $\text{WS}_2$ .

**Acknowledgements** A.G. and W.W.J.L. thank the Institute of Materials Research and Engineering and the Agency for Science, Technology and Research (A\*STAR) for providing the facilities for the

successful completion of this project. This research was supported by A\*STAR under its A\*STAR 2D PHAROS Grant No. 1527000016 and A\*STAR QTE Grant No. A1685b0005.

## References

1. A. Kuc, N. Zibouche, and T. Heine. "Influence of quantum confinement on the electronic structure of the transition metal sulfide  $TS_2$ ." *Phys. Rev. B* 83, 245213, (2011).
2. D. Ovchinnikov, A. Allain, Y. S. Huang, D. Dumcenco, & A. Kis. "Electrical Transport Properties of Single-Layer  $WS_2$ ." *ACS Nano*, 8(8), 8174–8181, (2014).
3. X. Liu, J. Hu, C. Yue, N. Della Fera, Y. Ling, Z. Mao, & J. Wei. "High Performance Field-Effect Transistor Based on Multilayer Tungsten Disulfide." *ACS Nano*, 8(10), 10396–10402, (2014).
4. T. Norden, C. Zhao, P. Zhang, R. Sabirianov, A. Petrou, & H. Zeng. "Giant valley splitting in monolayer  $WS_2$  by magnetic proximity effect." *Nature Communications*, 10(1), (2019).
5. Y. Wang, J. C. Kim, R.J. Wu, J. Martinez, X. Song, J. Yang, F. Zhao, A. Mkhoyan, H. Y. Jeong, M. Chhowalla. "Van der Waals contacts between three-dimensional metals and two-dimensional semiconductors." *Nature*, 568(7750), 70–74, (2019).
6. A. Allain, J. Kang, K. Banerjee, & A. Kis. "Electrical contacts to two-dimensional semiconductors." *Nature Materials*, 14(12), 1195–1205, (2015).
7. D. Kiefer, R. Kroon, A. I. Hofmann, H. Sun, X. Liu, A. Giovannitti, D. Stegerer, A. Cano, J. Hynynen, L. Yu, Y. Zhang, D. Nai, T. F. Harrelson, M. Sommer, A. J. Moulé, M. Kemerink, S. R. Marder, I. McCulloch, M. Fahlman, S. Fabiano, C. Müller. "Double doping of conjugated polymers with monomer molecular dopants." *Nature Materials*, 18(2), 149–155, (2019).
8. H. M. W. Khalil, M. F. Khan, J. Eom, H. Nom. "Highly Stable and Tunable Chemical Doping of Multilayer  $WS_2$  Field Effect Transistor: Reduction in Contact Resistance." *ACS Applied Materials & Interfaces*, 7(42), 23589–23596, (2015).
9. Y. Liu, J. Guo, E. Zhu, L. Liao, S. J. Lee, M. Ding, X. Duan. "Approaching the Schottky–Mott limit in van der Waals metal–semiconductor junctions." *Nature*, 557(7707), 696–700, (2018).
10. A. C. Gomez, M. Buscema, R. Molenaar, V. Singh, L. Janssen, H. S. J van der Zant, G. A. Steele. "Deterministic transfer of two-dimensional materials by all-dry viscoelastic stamping." *2D Materials*, 1(1), 011002, (2014).
11. Y. Liu, H. Wu, H. C. Cheng, S. Yang, E. Zhu, Q. He, M. Ding, D. Li, J. Guo, N. O. Weiss, Y. Huang, and X. Duan. "Toward barrier free contact to molybdenum disulfide using graphene electrodes." *Nano Lett.* 15, 3030–3034 (2015).
12. M. W. Iqbal, M. Z. Iqbal, M. F. Khan, M. A. Shehzad, Y. Seo, J. H. Park, C. Hwang, J. Eom. "High-mobility and air-stable single-layer  $WS_2$  field-effect transistors sandwiched between chemical vapor deposition-grown hexagonal BN films." *Scientific Reports*, 5(1), (2015).