



Crystallographic, Morphological Analysis on Al Doped ZnO Nanoparticles

A. Bendoumou^(✉), A. Belahmar, M. Fahoume, and A. Chouiyakh

Department of Physics, Laboratory of Condensed Matter Physics,
Faculty of Science, Ibn Tofail University, B.P.133, 14000 Kenitra, Morocco
bendoumouabdallah@gmail.com

Abstract. Undoped and Al-doped ZnO (AZO) nanoparticles (NPs) have been successfully synthesized by the simple sol-gel method. The NPs have been characterized by x-ray diffraction (XRD) and scanning electron microscopy (SEM) coupled with energy dispersive X-ray spectroscopy (EDX). XRD patterns show that pure and AZO samples have a hexagonal wurtzite structure. X-ray peak broadening analysis was used to evaluate the crystallite sizes, lattice strain and Young's modulus. The average crystallite size decreases with increase in Al concentration. Further appropriate physical parameters such as strain, stress, and energy density values were also calculated using Williamson-Hall (W-H) methods with different models. The surface morphology of the pellets samples was examined by (SEM) approve the nanostructure nature of the AZO powders. EDX confirms the presence of Al in the AZO nanoparticles.

Keywords: Sol-gel growth · $Zn_{1-x}Al_xO$ nanoparticles · XRD
MEB

1 Introduction

Zinc oxide (ZnO) is a semiconductor with a wide band-gap in the near UV at 3.36 eV and large exciton binding energy. These characteristics make this material interesting for many applications, such as solar cells [1], optical coating [2], and gas sensors [3]. The role of the particle size, doping, impurities, and morphology is very important to these applications, which has driven researchers to focus on the synthesis of doped and pure nanocrystalline ZnO in recent years. Doping semiconductor nanostructures in order to control their physical properties is an active field of research related to the development of nanotechnology applications. However, ZnO can be doped by many elements, such as S [4], N [5], Cu [6], Al [7], Ag [8], Eu [9] and Mg [10], to meet the demands of different applications. Among various dopants, Al can be easily doped in the lattice of ZnO for its similar radius and electronic shell to Zn atom.

Various methods have been used for the preparation of ZnO nanoparticles, such precipitation method [11], sol gel processing [12] and mechanical milling [13]. The sol-gel has some advantages such low temperature synthesis, high purity, homogeneity, repeatability and particularly its cheapness and simplicity. In this work, a simple sol gel method was used to prepare ZnO nanoparticles. The effects of Al content on structural and morphological properties of the obtained nanopowders were investigated.

2 Experimental Methods

ZnO particles were prepared by sol-gel method using zinc acetate as the precursor. A 4.4g zinc acetate ($\text{Zn}(\text{CH}_3\text{COO})_2 \cdot 2\text{H}_2\text{O}$) in 100 ml (70 ml of distilled water+30 ml of ethanol), this solution will be noted (A). 2 g sodium hydroxide was dissolved in 100 ml (80 ml of distilled water+20 ml ethanol) (solution B). After heating the solution A to 50 °C, solution (B) was added drop wise with constant warm stirring. This would result in the appearance of turbidity in the obtained solution leading to the formation of ZnO in the form of gel. After the complete addition, the resulting solution was stirred overnight which was then filtered, washed with water and dried at 100 °C for 2 h, manually ground and calcined at 200 °C for 3 h in order to obtain pure ZnO. In order to obtain $\text{Zn}_{1-x}\text{Al}_x\text{O}_x$ nanoparticles, an adequate amount of $((\text{Al}_2\text{SO}_4)_3, 18\text{H}_2\text{O})$ with Al/Zn proportions of (0.00, 0.01, 0.02 and 0.05) are dissolved in required amount of ethanol, were added to the solution A along with solution B simultaneously. After stirring the solution overnight (ZnO-doped) was formed, which was filtered, washed with distilled water to remove the excessions and insoluble materials remaining in the product and dried at 100 °C for 2 h. The powdered material obtained was manually ground and calcined at 200 °C for 3 h.

The structure and crystallite size of pure ZnO and (AZO) were determined by X-ray diffraction (XRD) using Cu-K α radiations ($\lambda = 0.15406 \text{ nm}$) in 2θ range from 10° to 80°. SEM technique coupled with EDX was employed to determine the size and the chemical composition of the samples.

3 Results and Discussions

3.1 Structural Parameters

Figure 1 shows the XRD patterns of $\text{Zn}_{1-x}\text{Al}_x\text{O}$ ($x\% = 0, 1, 2, 3, 5$), in the range of $2\theta = 10^\circ - 80^\circ$. They indicate the single phase formation with a hexagonal wurtzite structure for all samples. The three pronounced diffraction peaks at about $2\theta = 31.8^\circ, 34.5^\circ$ and 36.3° are identified, respectively, as the (100), (002) and (101) peaks of ZnO [14, 15]. The hexagonal structure is not changed during the doping, but it can be seen from this Figure that the incorporation of Al^{3+} ions affects the intensity of all peak, so it enhances the crystalline quality. For evaluating the preferred orientation of the samples, a texture coefficient will be used.

The quantitative information concerning the preferential crystal orientation can be obtained from the texture coefficient, TC, which can be defined as [16]:

$$TC_{(hkl)} = \left(\frac{I_{(hkl)}}{I_{0(hkl)}} \right) / \left[\frac{1}{N} \sum^n \frac{I_{(hkl)}}{I_{0(hkl)}} \right] \quad (1)$$

Where $TC_{(hkl)}$ is the texture coefficient, $I_{(hkl)}$ is the intensity of the XRD of the sample and n is the number of diffraction peaks considered. $I_{0(hkl)}$ is the intensity of the XRD reference of the randomly oriented grains. If $TC_{(hkl)} \approx 1$ for all the (hkl) planes are considered, then the nanoparticles are with a randomly oriented crystallite similar to

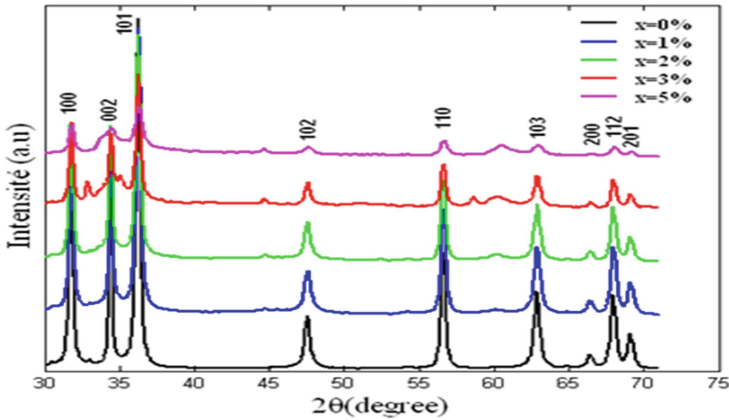


Fig. 1. XRD spectra of the undoped and Al doped ZnO nanoparticles for different doping contents

the JCPDS reference, while values higher than 1 indicate the abundance of grains in a given (hkl) direction. Values $0 < TC_{(hkl)} < 1$ indicate the lack of grains oriented in that direction. As $TC_{(hkl)}$ increases, the preferential growth of the crystallites in the direction perpendicular to the $\langle hkl \rangle$ plane is greater. In this analysis, three reflections are considered. From the Table 1, the (002) orientation is the preferred growth direction for all Al content. This direction will be considered for the calculation of all physical parameters.

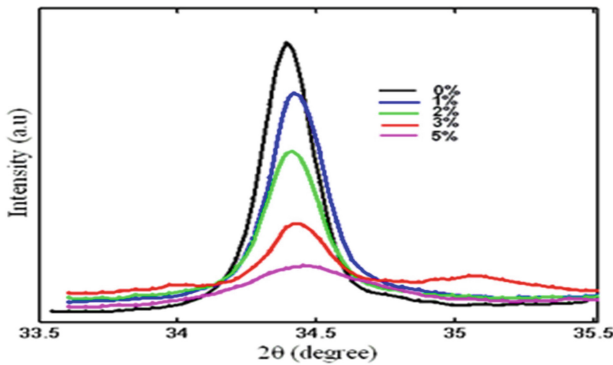


Fig. 2. Evolution of the preferred orientation peak (002) with various Al contents.

Indeed, the intensity of the preferred orientation (002) decreases gradually with increasing of Al content from 0% to 5%, as it is shown in Fig. 2, the width of this peak is also affected. Here, the Al^{3+} ions (ion radius 0.53 \AA) is smaller than Zn^{2+} ions (ion radius 0.74 \AA), so the increasing Al concentration will reduce the lattice constant of

Table 1. Texture coefficient $TC_{(hkl)}$ of as deposited $Zn_{1-x}Al_xO$ samples

Compound	2θ value($^\circ$)	$\langle hkl \rangle$	$TC_{(hkl)}$
ZnO	31.76	(100)	1.06
	34.39	(002)	1.39
	36.24	(101)	0.94
$Zn_{0.99}Al_{0.01}O$	31.79	(100)	1.02
	34.43	(002)	1.36
	36.27	(101)	0.93
$Zn_{0.98}Al_{0.02}O$	31.77	(100)	1.01
	34.41	(002)	1.33
	36.25	(101)	1.92
$Zn_{0.97}Al_{0.03}O$	31.78	(100)	1.11
	34.43	(002)	1.12
	36.27	(101)	0.96
$Zn_{0.95}Al_{0.05}O$	31.82	(100)	1.08
	34.32	(002)	0.84
	36.30	(101)	1.00

samples by substitution of Zn^{2+} ions with Al^{3+} . The average crystallite size was estimated using **Scherer's** formula.

$$D = K\lambda/\beta_{hkl}\cos\theta \quad (2)$$

Where D is the average crystallite size, K is a constant or shape factor and equal to 0.9, λ is the wavelength of the radiation (1.54056 Å with $CuK\alpha$ radiation), β_{hkl} is the peak width at half maximum intensity and θ is the peak position. The average crystallite size was in the range of 15–37 nm. It decreases with increasing Al content in $Zn_{1-x}Al_xO$ nanoparticles. It may be due to the small grain growth of (AZO) nanoparticles as compared with pure ZnO nanoparticles. The values of average crystallite size are listed in Table 3.

Owing to crystal imperfection and distortion, the strain induced in the samples was calculated by the formula [17]:

$$\varepsilon = \frac{\beta}{4\tan\theta} \quad (4)$$

The calculated values are listed in Table 3 and plotted in Fig. 4. It clearly shows that the un-doped sample shows low lattice strain values than as the doped samples. Figure 4 clearly explains the influence of Al doping on lattice strain with respect to the crystal orientation of un-doped and Al doped samples. We note that the increase of strain leads to the reduction in the nanoparticles size.

To understand the effect of Al doping on lattice parameters of ZnO, the lattice parameters should be calculated for undoped and Al doped ZnO nanoparticles. Since

Table 2. Structural parameters of Zn_{1-x}Al_xO

Compound	2θ value(°)	Particle size		Particle size	Strain (ε10 ⁻³)
		a(Å)	c(Å)	D _{hkl} (nm)	
ZnO	34.40	3.25	5.21	37.69	3.11
Zn _{0.99} Al _{0.01} O	34.43	3.25	5.21	35.31	3.32
Zn _{0.98} Al _{0.02} O	34.41	3.25	5.21	30.88	3.79
Zn _{0.97} Al _{0.03} O	34.43	3.25	5.20	30.63	3.82
Zn _{0.95} Al _{0.05} O	34.32	3.24	5.22	15.68	7.46

ZnO lattice is wurzite structure with hexagonal shape, there are two lattice parameters *a* and *c* are exist and can be calculated from the following relations [18]:

$$d_{(hkl)} = \left[\frac{4}{3} \left(\frac{h^2 + kl + k^2}{a^2} \right) + \frac{l^2}{c^2} \right]^{-1/2} \tag{5}$$

The lattice constant *a* for <100> lane is calculated by [18]:

$$a = \lambda / \sqrt{3} \sin \theta \tag{6}$$

For the <002> plane, the lattice constant *c* is calculated by [18]:

$$c = \lambda / \sin \theta \tag{7}$$

Where λ is the X-ray wavelength and $d_{(hkl)}$ is the spacing between planes of given Miller indices *h*, *k* and *l*. Using the relation given in Eq. (4) and (5), the lattice parameters of un-doped and Al doped ZnO nanoparticles were calculated and are summarized in Table 2. Thus, Al doping does not cause any remarkable change in the wurzite structure of ZnO. This may be due to the comparable ionic radii of Zn²⁺ and Al³⁺ which made the substitution of Zn²⁺ easy with Al³⁺ [19, 20] (Fig. 3).

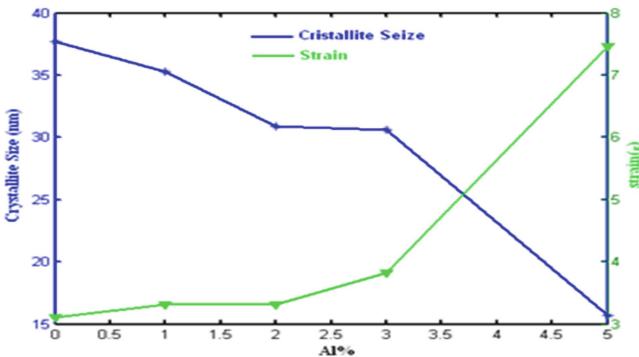


Fig. 3. Variation of crystallite (D) and strain (ε) of Zn_{1-x}Al_xO with different content

Residual stress is possible at the time of crystal growth. Moreover, the doping process may influence the residual stress of bare materials. So it has to be addressed for clear understanding the influence of doping on the change in residual stress of ZnO in this study. In this paragraph, only the effect of the doping is studied. $\epsilon Y = \sigma$

Where Y is the Young’s modulus. For a hexagonal crystal, Young’s modulus is given by the following relation [21]:

$$Y = \frac{\left[h^2 + \frac{(h+2k)^2}{3} + \left(\frac{al}{c} \right)^2 \right]^2}{s_{11} \left(h^2 + \frac{(h+2k)^2}{3} \right)^2 + s_{33} \left(\frac{al}{c} \right)^4 + (2s_{13} + s_{44}) \left(h^2 + \frac{(h+2k)^2}{3} \right) \left(\frac{al}{c} \right)^2} \quad (8)$$

Where $S_{11} = 7.858.10^{-12}$, $S_{13} = -2.206.10^{-12}$, $S_{33} = 6.940.10^{-12}$,

$S_{44} = 23.57.10^{-12}m^2/N$ respectively [22] are the elastic compliances of ZnO with values. Young’s modulus for $Zn_{1-x}Al_xO$ was calculated and reported in Table 3.

Table 3. Crystallite size, strain, dislocation density and stress of as deposited $Zn_{1-x}Mg_xO$ samples

Samples	Scherer’s method D (nm)	Strain ($\epsilon \times 10^{-3}$)	Dislocation Density $\delta.10^{15}m^{-2}$	Young’s Modulu $Y(GPa)$	Stress σ $10^6N.m^{-2}$
ZnO	37.69	3.11	0.7038	127.26	395.82
$Zn_{0.99}Al_{0.01}O$	35.31	3.32	0.8020	127.26	422.12
$Zn_{0.98}Al_{0.02}O$	30.88	3.79	1.0484	127.26	482.88
$Zn_{0.97}Al_{0.03}O$	30.63	3.82	1.0656	127.26	486.49
$Zn_{0.95}Al_{0.05}O$	15.68	7.46	4.0699	127.26	949.78

In the W-H method, the peak width from crystallite size does not follow a $1/\cos \theta$ dependency as in the Scherer’s equation but instead varies with $\tan \theta$. This fundamental difference allows for a separation of reflection broadening when both micro structural causes – small crystallite size and micro strain- occur together. The different approaches presented in the following assume that size and strain broadening are additive components of the total integral breadth of a Bragg peak [23]. Addition of the Scherer equation and $\epsilon = \beta/4 \tan \theta$ results in following equations:

$$\beta \cos \theta = \left(\frac{k\lambda}{D} \right) + 4\epsilon \sin \theta \quad (10)$$

Equation (10) represents the uniform deformation model (UDM), where the strain was assumed to be uniform in all crystallographic directions, thus considering the isotropic nature of the crystal, where the material properties are independent of the direction along which they are measured.

In the Uniform Stress Deformation Model, USDM, the generalized Hook's law referred to the strain, keeping only the linear proportionality between the stress and strain, i.e., $\sigma = Y \cdot \varepsilon$. Here, the stress is proportional to the strain, with the constant of proportionality being Young's modulus, denoted by Y . In this approach, the Williamson-Hall equation is modified by substituting the value of ε in Eq. (11), we get σ

$$\beta \cos \theta = \left(\frac{k\lambda}{D} \right) + \left(\frac{4\sigma \sin \theta}{Y} \right) \quad (11)$$

The strain can be calculated in the same way as ε .

For an elastic system that follows Hooke's law, the energy density ($u = (\varepsilon^2 Y)/2$) can be calculated using another model called the Uniform Deformation Energy Density Model, UDEDM:

$$\beta \cos \theta = \left(\frac{k\lambda}{D} \right) + \left(4 \sin \theta \left(\frac{2u}{Y} \right)^{1/2} \right) \quad (12)$$

For UDM model, the plot of $\beta \cos \theta$ along the y-axis, $4 \sin \theta$ along the x-axis and with a linear fit to the data, the crystalline size D was estimated from the y-intercept, and the strain ε , from the slope of the fit. In order to determine σ and u , the same method will be used for the UDSM AND UDEM models by keeping $\beta \cos \theta$ on the y-axis and $(4 \sin \theta / Y)$, $4 \sin \theta (2/Y)^{1/2}$ along the x-axis respectively. The corresponding plots of the three models are shown in Fig. 4:

Table 4 Summarizes the geometric parameters of ZnO nanoparticles obtained from Scherer's formula, various modified forms of W-H analysis. By comparing the values of average crystallite size obtained from UDM, UDSM, and UDEDM, it was found that the values are similar, implying that the inclusion of strain in various forms has a very small effect on the average crystallite size of ZnO nanoparticles. However, the average crystallite size obtained from Scherer's formula and W-H analysis shows a small variation; this is because of the difference in averaging the particle size distribution. The values of strain from each model are calculated by considering Young's modulus Y is equal 127 GPa.

3.2 Quantitative Results of Al Doped ZnO from EDX Analysis

The SEM photographs of the samples reported in Fig. 5 revealed that the crystallites are of the nanometer size.

The chemical compositions were analyzed by energy dispersive spectroscopy (EDX). Figure 6 shows the EDX spectrum obtained from pure ZnO powders, where only Zn and O are detected, while the presence of Al in the doped ZnO. From Table 5, the Atom% for Al increases with the increase of Al doping.

SEM results showed that the particle size was decreased with the doping of Aluminum.

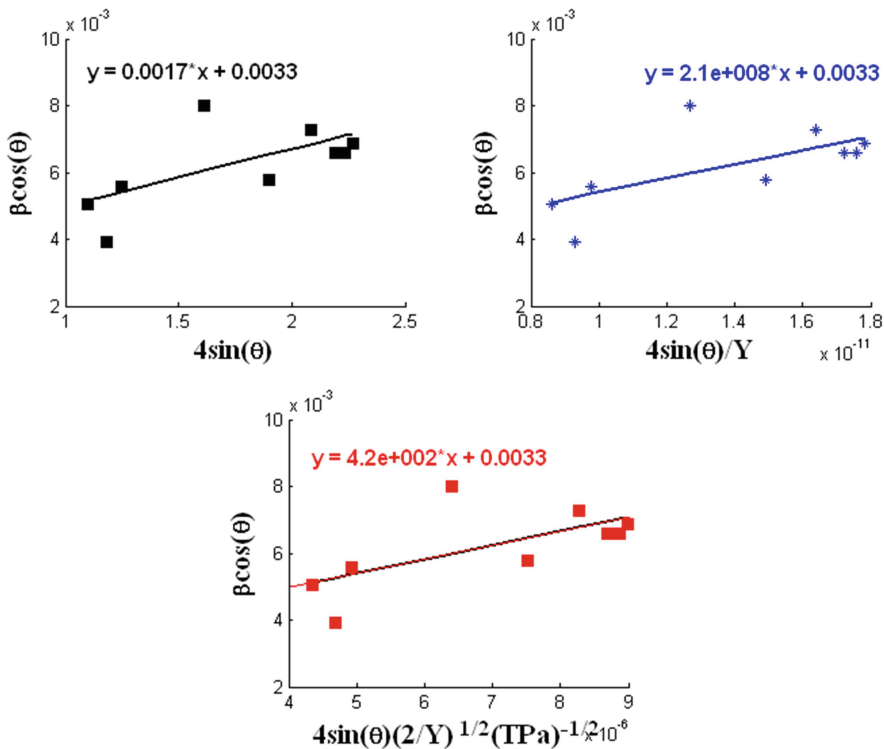


Fig. 4. UDM, USDM and UDEDM plots of $Zn_{0.99}Al_{0.01}O$ sample

Table 4. The geometric parameters of $Zn_{1-x}Al_xO$ samples

Samples	UDM		USDM		
	D(nm)	$\varepsilon \times 10^{-3}$	D(nm)	$\sigma(MPa)$	$\varepsilon \times 10^{-3}$
ZnO	50.21	1.8	50.21	230	1.80
$Zn_{0.99}Al_{0.01}O$	39.56	1.7	39.56	210	1.65
$Zn_{0.98}Al_{0.02}O$	40.79	1.6	40.79	210	1.65
$Zn_{0.97}Al_{0.03}O$	40.79	1.3	40.79	160	1.25
$Zn_{0.95}Al_{0.05}O$	22.12	1.5	22.12	190	1.49
Samples	UEDM				
	D(nm)	$\mu(kJm^{-1})$	$\sigma(MPa)$	$\varepsilon \times 10^{-3}$	
ZnO	50.21	202.5	227	1.8	
$Zn_{0.99}Al_{0.01}O$	39.56	176.4	211.9	1.7	
$Zn_{0.98}Al_{0.02}O$	40.79	168.1	206.8	1.6	
$Zn_{0.97}Al_{0.03}O$	40.79	102.4	161.4	1.3	
$Zn_{0.95}Al_{0.05}O$	22.12	136.9	186.7	1.5	

Table 5. Quantitative results of Al doped ZnO from EDX analysis

Sample compositions	pure		1%	2%	3%		5%			
Element	W%	A%	W%	A%	W%	A%	W%	A%	W%	A%
O	29.42	63.01	32.82	65.87	30.02	62.45	36.39	68.00	9.17	28.24
Al	0.0	0.0	1.63	1.94	2.65	3.27	4.46	4.94	3.02	5.52
Zn	70.58	36.99	65.55	32.20	67.33	34.28	59.15	27.06	87.82	66.23

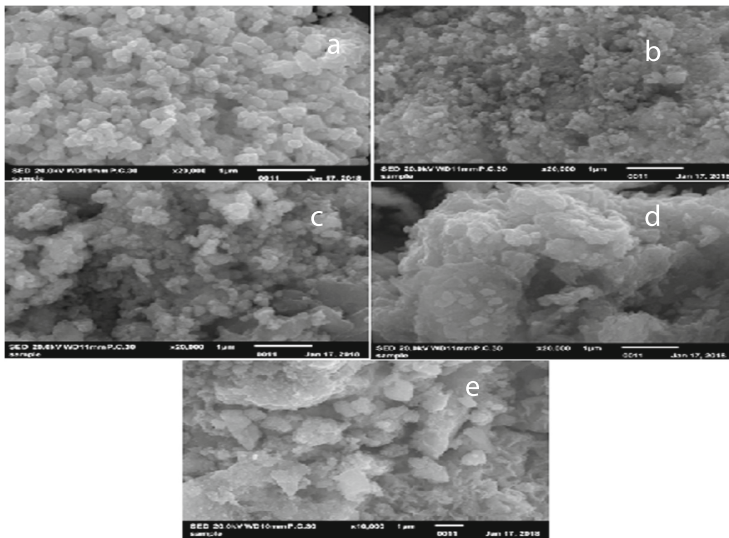


Fig. 5. SEM images of Al-doped ZnO powders with different doping amount. (a) $x = 0$, (b) $x = 0.01$, (c) $x = 0.02$ (d) $x = 0.03$ (e) $x = 0.05$

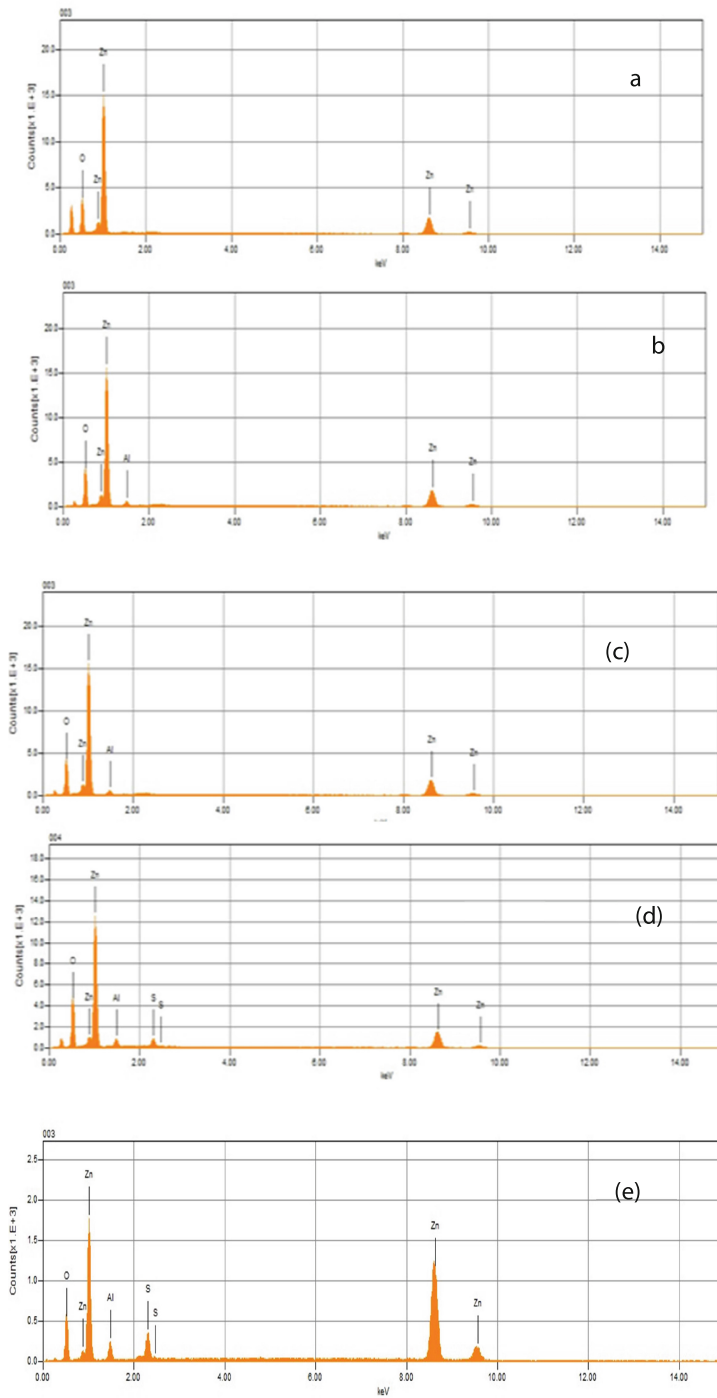


Fig. 6. EDX Spectra of Zn_{1-x}Al_xO samples (a) x = 0, (b) x = 0.01, (c) x = 0.02 (d) x = 0.03 (e) x = 0.05

4 Conclusion

Un-doped and Al doped ZnO nanoparticles were synthesized by a simple sol-gel method. The results of the XRD showed that the average particle size varies with the Al contents. The average values of crystallite size obtained from UDM, UDSM, and UDEDM are almost similar, which indicate the inclusion of strain in various forms of W–H analysis has a very small effect on the average crystallite size. The SEM confirms the nanostructure of the powders synthesized by sol gel, the EDX measurement indicates the incorporation of Al in the doped samples.

References

1. Lu L, Li R, Fan K, Peng T (2010) Effects of annealing conditions on the photo electrochemical properties of dye-sensitized solar cells made with ZnO nanoparticles. *Sol Energy* 84:844–853. <https://doi.org/10.1016/j.solener.2010.02.010>
2. Prepelita P, Medianu R, Sbarcea B, Garoi F, Filipescu M (2010) The influence of using different substrates on the structural and optical characteristics of ZnO thin films. *Appl Surf Sci* 256:1807–1811. <https://doi.org/10.1016/j.apsusc.2009.10.011>
3. Lee JH (2009) Gas sensors using hierarchical and hollow oxide nanostructures: overview. *Sens Actuators, B* 140:319–336. <https://doi.org/10.1016/j.snb.2009.04.026>
4. Xie XY, Zhan P, Li LY, Zhou DJ, Guo DY, Meng JX, Bai Y, Zheng WJ (2015) Synthesis of S-doped ZnO by the interaction of sulfur with zinc salt in PEG200. *J Alloys Compd* 644:383–389. <https://doi.org/10.1016/j.jallcom.2015.04.214>
5. Bai HW, Liu ZY, Sun DD (2012) Hierarchical nitrogen-doped flower like ZnO nanostructure and its multifunctional environmental applications. *Asian J Chem* 7: 1772–1780. <https://doi.org/10.1002/asia.201200260>
6. Xing GZ, Yi JB, Tao JG, Liu T, Wong LM, Zhang Z, Li GP, Wang SJ, Ding J, Sum TC, Huan CHA, Wu T (2008) Comparative study of room temperature ferromagnetism in Cu-doped ZnO nanowires enhanced by structural inhomogeneity. *Adv Mater* 20:3521–3527. <https://doi.org/10.1002/adma.200703149>
7. Baka O, Azizi A, Velumani S, Schmerber G, Dinia A (2014) Effect of Al concentrations on the electrode position and properties of transparent Al-doped ZnO thin films. *J Mater Sci Mat Electron* 25:1761–1769. <https://doi.org/10.1007/s10854-014-1796-3>
8. Yin XT, Que WX, Shen FY (2011) ZnO nanorods arrays with Ag nanoparticles on the (002) plane derived by liquid pitaxy growth and electrode position process. *Thin Solid Films* 520:186–192. <https://doi.org/10.1016/j.tsf.2011.07.016>
9. Lupan O, Pauporté T, Viana B, Aschehoug P, Ahmadi M, Cuenya BR, Rudzevich Y, Lin Y, Chow L (2013) Eu-doped ZnO nanowire arrays grown by electro deposition. *Appl Surf Sci* 282:782–788. <https://doi.org/10.1016/j.apsusc.2013.06.053>
10. Peng SY, Xu ZN, Chen QS, Wang ZQ, Lv DM, Sun J, Chen YM, Guo GC (2015) Enhanced stability of Pd/ZnO catalyst for CO oxidative coupling to dimethyl oxalate: effect of Mg²⁺ doping. *ACS Catal* 5:4410–4417. <https://doi.org/10.1021/acscatal.5b00365>
11. Chen CC, Liu P, Lu CH (2008) Synthesis and characterization of nanosized ZnO powders by direct precipitation, method. *Chem Eng J* 144:509–513. <https://doi.org/10.1016/j.cej.2008.07.047>

12. Ristiac M, Musiac S, Ivanda M, Popovia S (2007) Sol gel synthesis and characterization of nanocrystalline ZnO powders. *J Alloys Compd* 397:L1–L4. <https://doi.org/10.1016/j.jallcom.2005.01.045>
13. Suwanboon S, Amornpitoksuk P (2012) Preparation of Mg-doped ZnO nanoparticles by mechanical milling and their optical properties. *Procedia Eng* 32:821–826. <https://doi.org/10.1016/j.proeng.2012.02.018>
14. Moghaddam FM, Saeisian H, Moghaddam FM, Saeisian H (2007) Controlled microwave-assisted synthesis of ZnO nanopowder and its catalytic activity for O-acylation of alcohol and phenol. *Mater Sci Eng B* 265:139. <https://doi.org/10.1016/j.mseb.2007.03.002>
15. Tabib A, Sdiri N, Elhouichet H, Férid M (2015) Investigations on electrical conductivity and dielectric properties of Na doped ZnO synthesized from sol gel method. *J Alloys Compd* 622:687. <https://doi.org/10.1016/j.jallcom.2014.10.092>
16. Barret CS, Massalski TB (1980) *Structure of metals: crystallographic methods, principles and data*. Pergamon Press, Oxford, p 204
17. Klug HP, Alexander LE (1974) *X-ray diffraction procedures for polycrystalline and amorphous materials*. Wiley. [https://doi.org/10.1016/s0003-2670\(01\)95199-2](https://doi.org/10.1016/s0003-2670(01)95199-2)
18. Cullity BD, Stock SR (2001) *Elements of X-ray diffraction*, 3rd edn. Prentice Hall, New Jersey
19. Zhuang H, Wang J, Liu H, Li J, Xu P (2011) Structural and optical properties of ZnO nanowires doped with magnesium. *Acta Phys Polonica A* 119:819–823. <https://doi.org/10.12693/aphyspola.119.819>
20. Dai HQ, Xu H, Zhou YN et al (2012) Electrochemical characteristics of Al₂O₃-doped ZnO films by magnetron sputtering. *J Phys Chem C* 116:1519–1525. <https://doi.org/10.1186/s11671-016-162>
21. Warren BE, Averbach BL (1950) The effect of cold-work distortion on X-ray patterns. *J Appl Phys* 21:595–599. <https://doi.org/10.1063/1.1699713>
22. Nye JF (1985) *Physical properties of crystals their representation by tensor and matrices*. Oxford, New York, p 329
23. Mazhdi M, HosseinKhani P (2012) Structural characterization of ZnO and ZnO: Mn nanoparticles prepared by reverse micelle method. *Int J Nano Dimens* 2:233–240