# **Chapter 7 On Molecular Dynamics of the Diamond D5 Substructures**

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**Abstract** Diamond  $D_5$  is a hyperdiamond, with the rings being mostly pentagonal and built up on the frame of *mtn* structure, appearing in type II clathrate hydrates. The centrohexaquinane  $C_{17}$  was proposed as the seed of  $D_5$  (Diudea, Studia Univ Babes-Bolyai Chemia, 55(4):11–17, 2010a; Diudea, Nanomolecules and nanostructures – polynomials and indices. University of Kragujevac, Kragujevac, 2010b). In this chapter, we present some results on molecular dynamics (MD) of four structures based on  $C_{17}$  skeleton, as all-carbon or partly oxygenated derivatives. The results are discussed in terms of structural stability as given by DFT calculations as well as by the stable fluctuations of root-mean-square deviations (*RMSD*) and total, potential, and kinetic energies provided by MD calculations. Within  $D_5$ , several other substructures are discussed in this chapter. The structural stability of such intermediates/fragments appearing in the construction/destruction of  $D_5$  net is also discussed in terms of molecular dynamics simulation. The calculations herein discussed have been done using an empirical many-body potential energy function for hydrocarbons. It has been found that, at normal temperature, the hexagonal hyper-rings are more stable, while at higher temperature, the pentagonal ones are relatively stronger against the heat treatment.

### **7.1 Introduction**

In the nano-era, a period starting with the discovery of  $C_{60}$  in 1985, the carbon allotropes played a dominant role. Among the carbon structures, fullerenes (zero dimensional), nanotubes (one dimensional), graphene (two dimensional), diamond,

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**Fig. 7.1** Diamond  $D_6$  (*left*) and its repeating unit, adamantane (*right*)

<span id="page-1-0"></span>and spongy carbon (three dimensional) were the most studied (Diudea [2005,](#page-17-0) [2010a;](#page-17-1) Diudea and Nagy [2007\)](#page-17-2), both from theoretical reasons and applications perspective.

Diamond  $D_6$ , the beautiful classical diamond, with all-hexagonal rings of  $sp^3$ carbon atoms (Fig. [7.1\)](#page-1-0), crystallized in the face-centered cubic *fcc* network (space group *Fd-*3*m*), has kept its leading interest among the carbon allotropes, in spite of the "nano" varieties (Decarli and Jamieson [1961;](#page-17-3) Aleksenskil et al. [1997;](#page-16-0) Osawa [2007,](#page-18-0) [2008;](#page-18-1) Williams et al. [2007;](#page-18-2) Dubrovinskaia et al. [2006\)](#page-17-4). Its aesthetical appeal and mechanical characteristics are of great importance in jewelry and industry. Synthetic diamonds are currently produced by a variety of methods, including high pressure-high temperature (HPHT), chemical vapor deposition (CVD), and ultrasound cavitation (Khachatryan et al. [2008\)](#page-18-3).

However, the diamond  $D_6$  is not unique: a hexagonal network called lonsdaleite (space group *P*63/*mmc*) (Frondel and Marvin [1967\)](#page-17-5) was discovered in a meteorite in the Canyon Diablo, Arizona, in 1967. Several diamond-like networks have also been proposed (Diudea et al. [2010;](#page-17-6) Hyde et al. [2008\)](#page-17-7).

In a previous study, Diudea and Ilic  $(2011)$  $(2011)$  described some multi-tori (i.e., structures showing multiple hollows (Diudea and Petitjean [2008\)](#page-17-9)); one of them is illustrated in Fig. [7.2,](#page-2-0) left.

The reduced graph of this multi-torus provided the structure for the seed of diamond  $D_5$ :  $C_{17}$  (Fig. [7.2,](#page-2-0) right) consisting of a tetravalent atom surrounded by six pentagons, the maximum possible number of pentagons around an  $sp<sup>3</sup>$  carbon atom. According to the chemical nomenclature,  $C_{17}$  is a centrohexaquinane, a class of structures previously studied by Gund and Gund [\(1981\)](#page-17-10), Paquette and Vazeux [\(1981\)](#page-18-4), and more recently by Kuck [\(1984,](#page-18-5) [2006\)](#page-18-6) and Kuck et al. [\(1995\)](#page-18-7).

Diamond  $D_5$  is the name given by Diudea to diamondoids consisting mostly of pentagonal rings (Diudea  $2010a$ , [b;](#page-17-11) Diudea and Ilić  $2011$ ). D<sub>5</sub> is a hyperdiamond built up in the frame of the trinodal *mtn* structure, while its seed is eventually the centrohexaquinane  $C_{17}$ . However,  $D_5$  belongs to the family of Clathrates with the point symbol net  $\{5^{\circ}5.6\}12\{5^{\circ}6\}5$  and  $2[5^{12}] + [5^{12} \times 6^4]$  tiling and belongs to the



**Fig. 7.2** A multi-torus (*left*) and its reduced graph  $C_{17}$  (*right*), the seed of diamond  $D_5$ 

<span id="page-2-1"></span><span id="page-2-0"></span>**Fig. 7.3** A joint of two C17 units (*left*) to give a dimer C34 (*right*), the repeat unit (in crystallographic terms) of the diamond  $D_5$  network



space group *Fd*3*m* (Delgado-Friedrichs et al. [2005\)](#page-17-12). It is precisely type II clathrate, also called  $C_{34}$  (Blasé et al. [2010\)](#page-17-13), of which  $Si_{34}$ -analogue was already synthesized.

 $C_{17}$  can dimerize to  $2 \times C_{17} = C_{34}$  (Fig. [7.3\)](#page-2-1), the repeating unit, in crystallo-<br>which terms of the diamond D<sub>s</sub> network. Thus D<sub>s</sub> (Fig. 7.4) and fcc-Co<sub>1</sub> are graphic terms, of the diamond  $D_5$  network. Thus,  $D_5$  (Fig. [7.4\)](#page-3-0) and  $fcc-C_{34}$  are herein synonyms (Fig. [7.5\)](#page-3-1).

In a chemist's view, the building of  $D_5$  network may start with the seed  $C_{17}$  and continue with some intermediates, the adamantane- and diamantane-like ones included (Figs. [7.6,](#page-3-2) [7.7,](#page-4-0) and [7.8\)](#page-4-1). The structure ada 20 158 (Fig. 7.7, left) corresponds to adamantane (Fig.  $7.5$ , left) in the classical diamond  $D_6$ . In crystallochemical terms, an adamantane-like structure, as ada\_20\_158, is the



**Fig. 7.4** Diamond  $D_5_20_860$  net (*left*) and  $D_5_28_1022$  co-net (*right*)

<span id="page-3-0"></span>

**Fig. 7.5** Diamond D6: adamantane (*left*), diamantane (*right*)

<span id="page-3-1"></span>

<span id="page-3-2"></span>**Fig. 7.6** Intermediate structures originating in  $C_{34}$  unit:  $C_{51}$  (*left*) and  $3 \times C_{51}$  (*right*)

monomer which will probably condense to form the  $D_5$  network (Fig. [7.4\)](#page-3-0). The adalike structure, starting from  $C_{28}$  can be seen in Fig. [7.7,](#page-4-0) right. Diamantane-like units can also be modeled, as in Fig. [7.8](#page-4-1) (compare with the diamantane, Fig. [7.5,](#page-3-1) right). In fact, there is one and the same triple periodic  $D_5$  network, built up basically from  $C_{20}$  and having as hollows the fullerene  $C_{28}$ . The co-net  $D_5$  28 cannot be derived from  $C_{28}$  alone since the hollows of such a net consist of  $C_{57}$  units (a  $C_{20}$ -based



**Fig. 7.7** Adamantane-like structures: ada\_20\_158 (*left*) and ada\_28\_213 (*right*)

<span id="page-4-0"></span>

<span id="page-4-1"></span>**Fig. 7.8** Diamantane-like structures: dia\_20\_226 net (*left*) and dia\_28\_292 co-net (*right*)

structure, see above) or higher tetrahedral arrays of  $C_{20}$ , thus needing extra C atoms per ada-unit. It is worthy to note the stabilizing effect of the wings in case of  $C_{34}$ (Fig. [7.3,](#page-2-1) right) in comparison to  $C_{20}$ .

Remark the efforts made by a series of bright scientists (Prinzbach et al. [2006;](#page-18-8) Paquette et al. [1981;](#page-18-9) Saito and Miyamoto [2001;](#page-18-10) Eaton [1979\)](#page-17-14) to reach the dodecahedral cage  $C_{20}$ , either as fullerene or hydrogenated species. Also remark the endeavor to synthesize the centrohexaquinane  $C_{17}$ , both as oxygen-containing heterocycle (Simmons and Maggio [1981;](#page-18-11) Paquette and Vazeux [1981\)](#page-18-4) or all-carbon structure (Gestmann et al. [2006;](#page-17-15) Kuck [2006\)](#page-18-6). Thus, the hyperdiamond  $D_5$  20/28 mainly consists of  $sp<sup>3</sup>$  carbon atoms building ada-type repeating units (including  $C_{28}$  as hollows). The ratio C-sp<sup>3</sup>/C-total trends to one in a large enough network. As the content of pentagons  $R[5]$  (Aleksenskii et al. [1997;](#page-16-0) Williams et al. [2007\)](#page-18-2) per total rings trend to 90 %, this network was named the diamond  $D_5$  (Diudea [2010a,](#page-17-1) [b\)](#page-17-11).

In the above symbols, "20" refers to  $C_{20}$  and "28" refers to  $C_{28}$ , while the last number counts the carbon atoms in structures.

## **7.2 Method**

In a study of structural stability, performed by Szefler and Diudea [\(2012\)](#page-18-12), ab initio calculations and molecular dynamics have been used. The four structures (Figs. [7.9](#page-5-0) and [7.12\)](#page-11-0) based on  $C_{17}$  skeleton, as all-carbon or partly oxygenated derivatives, were optimized at the Hartree-Fock (HF) (HF/6-31G\*\*) and DFT (B3LYP/6-  $311+\text{G}^{**}$ ) levels of theory and submitted to molecular dynamics (MD) procedure. All calculations were performed in gas phase by Gaussian 09 (Gaussian 09 software package [2009\)](#page-17-16) while MD calculations were done in vacuum, using Amber 10.0 software (Case et al. [2005\)](#page-17-17). The single-point energy minima obtained for the investigated structures are shown in Table [7.1.](#page-5-1) Before MD, the atomic charges were calculated according to Merz-Kollmann scheme via the RESP (Wang et al. [2000\)](#page-18-13) procedure, at HF/6-31G\*\* level. The AMBER force field (Wang et al. [2004\)](#page-18-14) was used for dynamic trajectory generation. There were several steps of molecular dynamics. After stabilization of energies and RMSD values during run, the actual molecular dynamics were performed, in a cascade way. Each tested system was heated by 20 ps while MD simulations were 100 ns long. The visualizations were prepared in the GaussView program. After MD run, the values of RMSD and energies of analyzed structures were recorded: total energy  $(E_{\text{tot}})$ , kinetic energy  $(E_{kin})$ , and potential energy  $(E_{pot})$ . In the analysis, averaged values of all generated points of energies and values of RMSD in every 1 ps of MD were used.

The stability of 12 other substructures was investigated by Kyani and Diudea [\(2012\)](#page-18-15) by performing a molecular dynamics (MD) computer simulation, using an empirical many-body adaptive intermolecular reactive empirical bond-order (AIREBO) potential energy function. All the diamond  $D_5$  substructures are fully



**Fig. 7.9** C<sub>17</sub> hexaquinane trioxo derivatives: Paquette P<sub>1</sub> (*left*) and Diudea, D<sub>1</sub> (*middle*) and D<sub>2</sub> (*right*)

<span id="page-5-1"></span><span id="page-5-0"></span>**Table 7.1** The single-point energies of the optimized structures at DFT  $(B3LYP/6-311+G**)$  level of theory

| B3LYP           | $C_{17}$   | Dı         | D٠         | Ρ,         |
|-----------------|------------|------------|------------|------------|
| B3LYP(a.u)      | $-655.058$ | $-766.491$ | $-766.480$ | $-766.479$ |
| $B3LYP_Gap(eV)$ | 5.868      | 6.461      | 6.264      | 6.274      |

hydrogenated ones. The studied structures were optimized at the semiempirical PM3 level of theory and then submitted to the MD simulation procedure. Canonical ensemble molecular dynamics was used for this simulation. Within this ensemble, the number of atoms  $N$ , the volume  $V$ , and the temperature  $T$  are considered constants while velocities are scaled with respect to *T*, ensuring that the total kinetic energy, and hence the temperature, is constant (isokinetic MD). The initial velocities follow the Maxwell distribution. The AIREBO potential energy function (PEF) developed for hydrocarbons (Stuart et al. [2000\)](#page-18-16), as provided by LAMMPS software (Plimpton [1995\)](#page-18-17), was used to investigate the stability of nanostructures at increasing temperatures. This parameterized potential adds Lennard-Jones and torsional contributions to the many-body REBO potential (Brenner [2000;](#page-17-18) Brenner et al. [2000\)](#page-17-19). It is similar to a pairwise dispersion-repulsion potential, while adding a bond-order function modulates the dispersion term and incorporates the influence of the local atomic environment. Through this interaction, individual atoms are not constrained to remain attached to specific neighbors, or to maintain a particular hybridization state or coordination number. Thus, at every stage of the simulation, the forming and breaking of the bonds is possible. This potential is derived from ab initio calculations, and therefore, it is well adapted to classical molecular simulations of systems containing a large number of atoms such as carbon nanostructures. The equations of particle motion were solved using the Verlet algorithm (Verlet [1967,](#page-18-18) [1968\)](#page-18-19), and the temperature was gradually increased by 100 K at each run. One time step was taken to be 10–16 s and at each run a relaxation with 5,000 time steps was performed. The root-mean-square deviation (RMSD) of the atoms was used as criterion for examining the stability of the simulated structures.

#### **7.3 Results and Discussion**

Stability evaluation was performed on four hypothetical seeds of  $D_5$ , the all-carbon structure  $C_{17}$  (Fig. [7.2,](#page-2-0) right) and three trioxa derivatives of  $C_{17}$ . The isomer in Fig. [7.9,](#page-5-0) left, was synthesized by Paquette and Vazeux [\(1981\)](#page-18-4) and is hereafter denoted as  $P_1$ . Other two structures, denoted as  $D_1$  and  $D_2$  (Fig. [7.9,](#page-5-0) middle and right), were proposed (Szefler and Diudea [2012\)](#page-18-12), as possibly appearing in rearrangements of the Paquette's  $P_1$  structure. The last two structures would be the appropriate ones in the next step of dimerization to  $C_{34}$ , in fact the repeating unit of D5 (Blasé et al. [2010\)](#page-17-13).

The stability of molecules was evaluated both in static and dynamic temperature conditions. The isomer  $D_1$  seems the most stable among all studied structures, as given by optimization in gas phase at DFT level (Table [7.1\)](#page-5-1). In a decreasing order of stability, it follows  $P_1$  and  $D_2$ . However, at MD treatment, the all-carbon  $C_{17}$ appears the most stable, even at DFT level is the last one. This is probably because the C–C bond is more stable at temperature variations (see Fig. [7.12,](#page-11-0) below).

In MD,  $C_{17}$  keeps its structure up to about 1,800 K, while its destruction starts at 2,000 K (Tables [7.2](#page-7-0) and [7.4,](#page-10-0) Figs. [7.10](#page-7-1) and [7.11\)](#page-8-0). Kuck has reported a

| $C_{17}$                    |         |         |         |         |         |         |         |
|-----------------------------|---------|---------|---------|---------|---------|---------|---------|
| TEMP(K)                     | 300     | 600     | 900     | 1,200   | 1,400   | 1,600   | 1,800   |
| $E_{\text{tot}}$ (kcal/mol) | 127.405 | 180.379 | 233.358 | 286.460 | 322.227 | 357.665 | 393.123 |
| $\delta$                    | 5.789   | 11.585  | 17.373  | 23.121  | 27.070  | 30.932  | 34.938  |
| $D_1$                       |         |         |         |         |         |         |         |
| TEMP(K)                     | 300     | 600     | 900     | 1,200   | 1,400   | 1,600   |         |
| $E_{\text{tot}}$ (kcal/mol) | 144.996 | 197.907 | 250.689 | 303.114 | 337.947 | 373.136 |         |
| $\delta$                    | 5.782   | 11.593  | 17.273  | 23.106  | 26.739  | 30.745  |         |
| D <sub>2</sub>              |         |         |         |         |         |         |         |
| TEMP(K)                     | 300     | 600     | 900     |         |         |         |         |
| $E_{\text{tot}}$ (kcal/mol) | 116.45  | 169.266 | 222.049 |         |         |         |         |
| $\delta$                    | 5.783   | 11.542  | 17.363  |         |         |         |         |
| P <sub>1</sub>              |         |         |         |         |         |         |         |
| TEMP(K)                     | 300     | 600     | 900     | 1,200   | 1,400   | 1,600   |         |
| $E_{\text{tot}}$ (kcal/mol) | 142.574 | 195.448 | 247.889 | 300.329 | 335.524 | 370.602 |         |
| $\delta$                    | 5.798   | 11.535  | 17.300  | 23.076  | 26.931  | 30.678  |         |
|                             |         |         |         |         |         |         |         |

<span id="page-7-0"></span>**Table 7.2** The average total energy  $(E_{\text{tot}})$  values estimated, by MD, on geometries in the gas phase

The averaged values were calculated on all the generated points of energies in every 1 ps of molecular dynamics

Symbol  $\delta$  means the standard deviation



<span id="page-7-1"></span>**Fig. 7.10** The plot of total energy  $(E_{\text{tot}})$  versus temperature (TEMP)

centrohexaindane as the most symmetric structure in this series but also a benzocentrohexaquinane (Kuck et al. [1995;](#page-18-7) Kuck [2006\)](#page-18-6) as the last step structure in the synthesis of a nonplanar 3D structure, designed according to mathematical rules. However, in the synthesis of centrohexaquinane derivatives,  $C_{17}$  remained yet elusive.



<span id="page-8-0"></span>**Fig. 7.11** The plot of RMSD versus temperature (TEMP)

Very close to  $C_{17}$  behaves the oxygen-containing isomer  $D_1$ , as expected from its highest stability at DFT level.

Despite, in molecular dynamics, a very long time (100ns) was leaded, it is believed that prolonged annealing at 1,800 K for both  $P_1$  and  $D_1$  isomers finally resulted in the destruction of these molecules. Thus,  $P_1$  and  $D_1$  isomers behave similarly in MD conditions. The isomer  $D_2$  was the least stable one, as the largest RMSD values were recorded for this isomer.

According to molecular dynamics, it is clear that increasing the temperature resulted in higher values of energy and RMSD of all the analyzed structures, with high values of correlation. The plots of  $E_{\text{tot}}$  versus temperature for all tested systems are given in Fig. [7.10,](#page-7-1) while for RMSD, the plots are given in Fig. [7.11.](#page-8-0) As expected, the correlations in the RMSD plot are a little lower than those for  $E_{\text{tot}}$ . The MD calculations, listed in Tables [7.2,](#page-7-0) [7.3,](#page-9-0) and [7.4,](#page-10-0) show the following.

As can be seen from Tables [7.2](#page-7-0) and [7.3,](#page-9-0) the values of standard deviations of the averaged values of  $E_{\text{tot}}$  are closely correlated with the values of temperature, in the range the molecular dynamics simulations were done. The values of these standard deviations at a given temperature are similar for all four studied structures, due to their structural relatedness. The smallest values of the RMS deviation are observed for  $C_{17}$ , with the lowest values of standard deviation ( $\delta$ ) at all the studied values of temperature (Table [7.4\)](#page-10-0).

In the case of  $P_1$ , one can see a similar behavior but somewhat with larger values of RMSD, compared to the all-carbon structure  $C_{17}$  (Table [7.4\)](#page-10-0). It confirms the structural stability of the above structures. The largest values of the RMS deviation were recorded for  $D_2$  isomer (Table [7.4](#page-10-0) and Fig. [7.11\)](#page-8-0). Visualization of the structural changes (first step destruction, the right column) is presented in Fig. [7.12.](#page-11-0)



<span id="page-9-0"></span>**Table 7.3** The values of standard deviations of  $E_{\text{tot}}$  at a given temperature (see the center of each slide) for the four investigated structures

When discussing about diamonds, we consider structures consisting mostly of  $sp<sup>3</sup>$  hybridized carbon atom. The molecular dynamic with formulation and parametrization intended for carbon system is REBO (Tersoff [1988a,](#page-18-20) [b;](#page-18-21) Abell [1985\)](#page-16-1). The Tersoff's and Brenner's (Brenner [1990,](#page-17-20) [1992\)](#page-17-21) models could describe single-, double-, and triple-bond energies in carbon structures such as hydrocarbons and diamonds, where extended Tersoff's potential function is extended to radical and conjugated hydrocarbon bonds by introducing two additional terms into the bond-order function. Compared to the classical first-principle and semiempirical approaches, the REBO model is less time-consuming. In recent years, the REBO model has been widely used in studies concerning mechanical and thermal properties of carbon nanotubes (Ruoff et al. [2003;](#page-18-22) Rafii-Tabar [2004\)](#page-18-23).

Molecular dynamic MD calculations using the REBO model were performed by Kyani and Diudea [\(2012\)](#page-18-15) on the structures listed in Figs. [7.13](#page-12-0) and [7.14.](#page-13-0) The last number in the symbol of structures refers to the number of carbon atoms.

| $C_{17}$       |       |       |       |       |       |       |       |
|----------------|-------|-------|-------|-------|-------|-------|-------|
| TEMP(K)        | 300   | 600   | 900   | 1,200 | 1,400 | 1,600 | 1,800 |
| <b>RMSD</b>    | 0.151 | 0.224 | 0.305 | 0.366 | 0.358 | 0.437 | 0.438 |
| δ              | 0.024 | 0.032 | 0.045 | 0.056 | 0.054 | 0.054 | 0.062 |
| $D_1$          |       |       |       |       |       |       |       |
| TEMP(K)        | 300   | 600   | 900   | 1,200 | 1,400 | 1,600 |       |
| <b>RMSD</b>    | 0.324 | 0.328 | 0.448 | 0.465 | 0.532 | 0.574 |       |
| $\delta$       | 0.134 | 0.088 | 0.079 | 0.086 | 0.091 | 0.118 |       |
| D <sub>2</sub> |       |       |       |       |       |       |       |
| TEMP(K)        | 300   | 600   | 900   |       |       |       |       |
| <b>RMSD</b>    | 0.432 | 0.500 | 0.542 |       |       |       |       |
| $\delta$       | 0.257 | 0.245 | 0.208 |       |       |       |       |
| $P_1$          |       |       |       |       |       |       |       |
| TEMP(K)        | 300   | 600   | 900   | 1,200 | 1,400 | 1,600 |       |
| <b>RMSD</b>    | 0.220 | 0.283 | 0.365 | 0.392 | 0.440 | 0.431 |       |
| $\delta$       | 0.062 | 0.069 | 0.092 | 0.070 | 0.093 | 0.070 |       |

<span id="page-10-0"></span>**Table 7.4** The averaged RMSD values estimated by molecular dynamics (MD) on the geometries in the gas phase

The averaged values were calculated on all the generated points of RMSD in every 1 ps of MD

Symbol  $\delta$  represents the standard deviation

The main reason for doing calculations on hydrogenated species, although the fragments can appear as non-hydrogenated ones, is the  $sp<sup>3</sup>$  hybridization of carbon atoms in the diamond structures. Thus, the four-valence state is preserved.

PM3 calculations show the hexagonal hyper-rings more stable than the pentagonal ones, either with a hollow or filled one as in case of lens-like structures (Table [7.5,](#page-14-0) entries 5, 6 and 10). There is one exception; the empty hexagon of  $C_{28}$ fullerenes (entry 9) is less stable than the corresponding empty pentagon (entry 7). The hexagonal filled ring structures are more stable than the empty ones, except the empty hexagon 20<sup>6</sup> H<sub>60</sub> (entry 5), made from C<sub>20</sub>, which seems to be the most stable, as isolated structure, herein discussed. Similarly,  $C_{20}H_{20}$  (entry 1) is the stabilized form of the most reactive/unstable fullerene  $C_{20}$ . The stabilizing effect of ring filling is a reminiscence of the infinite crystal lattice, whose substructures are  $20^6$   $28^2$ H<sub>68</sub> (diamond D<sub>5</sub> net) and  $28^6$   $20^2$ H<sub>92</sub> (lonsdaleite L<sub>5</sub> net).

As in studies of the stability of hypothetical seeds of the diamond  $D_5$  by using Amber 10.0 for MD, also here we could see that with increasing temperature, the energy of the systems increases, what would be expected. The total energy versus temperature for three of the considered systems is illustrated in (Fig. [7.15\)](#page-14-1).

MD simulations on increasing temperature evolution show that the studied structures are stable up to 2,000 K, for the  $C_{20}$ -based structures (Fig. [7.13\)](#page-12-0), and up to 1,500–2,000 K, for the  $C_{28}$ -based structures (Fig. [7.14\)](#page-13-0). The three numbers at the figure bottom represent temperatures, in K degree, for: geometry modifications, topology changes, and major destruction of the structure. Where there are only two data, the topological changes were not observed.



<span id="page-11-0"></span>Fig. 7.12 The structure of the tested hypothetical seeds of the diamond D<sub>5</sub> during molecular dynamics



 $20^5$  75 2000, 2200, 2500 K



 $20^5$   $20^2$   $85$  top 2000, 2700, 3000 K



206 90 2000, 2300 K



 $20^6$ \_28<sup>2</sup>\_110 top 2000, 2500 K



<span id="page-12-0"></span>**Fig. 7.13** C<sub>20</sub>-based structures;  $20^6 \text{--} 28^2 \text{--} 110$  a substructure of D<sub>5</sub> $\text{--}20/28$  (Reproduced from Central European Journal of Chemistry 2012, 10(4), 1028–1033)



 $28^{5}$ \_110 2000, 2100, 2500 K



 $28^5$   $20^2$  120 top 1800, 2000, 3000 K



 $28^6$  132 100,1500, 2500 K



 $28^6\_20\_134$  top 1500, 1700, 2000 K



<span id="page-13-0"></span>Fig. 7.14 C<sub>28</sub>-based structures; 286\_20\_134 a substructure of L5\_28/20. The *red* lines show the core substructure, that is  $C_{20}$  the smallest fullerene (Reproduced from Central European Journal of Chemistry 2012, 10(4), 1028–1033)

<span id="page-14-0"></span>**Table 7.5** PM3 energies calculated on fully hydrogenated species

|    | Structure                     | Total energy (a.u.) | Gap (eV) |
|----|-------------------------------|---------------------|----------|
|    | $C_{20}H_{20}$                | $-0.0620$           | 13.905   |
| 2  | $C_{28}H_{28}$                | $-0.0035$           | 13.769   |
| 3  | $20^5$ H <sub>50</sub>        | 0.0045              | 12.953   |
| 4  | $20^5 - 20^2$ H <sub>50</sub> | 0.0432              | 12.898   |
| 5  | $20^6$ _H <sub>60</sub>       | $-0.1215$           | 13.198   |
| 6  | $20^6$ $28^2$ H <sub>68</sub> | $-0.0167$           | 13.061   |
| 7  | $28^5$ _H <sub>80</sub>       | 0.1564              | 13.034   |
| 8  | $28^5 - 20^2$ H <sub>80</sub> | 0.1648              | 13.061   |
| 9  | $28^6$ _H <sub>60</sub>       | 0.3432              | 12.490   |
| 10 | $28^6 - 20^2$ H <sub>92</sub> | 0.1353              | 13.034   |

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<span id="page-14-1"></span>**Fig. 7.15** Total energy versus temperature of some of the diamond  $D_5$  substructures:  $20^5 \text{--} 20^2 \text{H}_{50}$ (*solid line*, Table [7.5,](#page-14-0) entry 4),  $20^6$ <sup>-282</sup>H<sub>68</sub> (*simple line*, Table 7.5, entry 6), and  $28^5$ <sup>-202</sup>H<sub>80</sub> (*dotted line*, Table [7.5,](#page-14-0) entry 8) (Reproduced from Central European Journal of Chemistry 2012, 10(4), 1028–1033)

The structures in Fig. [7.16](#page-15-0) show elongated bonds (i.e., broken bonds, marked by orange color) at temperatures above 2,000 K. Question about structure preserving must be addressed when more than one broken bond will appear (see the bottom row, Fig. [7.16\)](#page-15-0). Above 2,500 K, the complete destruction is expected for all the three structures shown in Fig. [7.16;](#page-15-0) for two structures (those with only two temperatures on their bottom) the topological changes were not observed. In case of the basic fullerenes, the data are:  $C_{20}H_{20}$ , 2,700 and 3,000 K, and  $C_{28}H_{28}$ , 2,500, 2,600,



<span id="page-15-0"></span>**Fig. 7.16** Relaxed structures of  $20^5$   $20^2$  85,  $20^6$   $28^2$  85 and  $28^5$   $20^2$  85 at various temperatures. The *orange* colors mark the broken/formed bonds (Reproduced from Central European Journal of Chemistry 2012, 10(4), 1028–1033)

and 3,000 K. Among the frequent topological changes, the most important is the expansion of two pentagons sharing an edge to octagon and also the apparition of trigons, squares, or larger rings before the structure dramatically decomposes. It is obvious that the structural changes will affect the energetics of the system.

The MD data agree with the PM3 data; in fact, the  $C_{20}$ -based structures are more stable than the  $C_{28}$ -ones, which corresponds to the higher stability of diamond  $D_5$ compared to lonsdaleite  $L_5$  (Aste and Weaire [2008\)](#page-17-22). At higher temperature, the fivefold hyper-rings seem to be more stable, at least in the isolated fragments (see Figs. 8.13 and 8.14). Remind that such fragments can appear either in synthesis of  $D_5$ 

or in its destruction, their knowledge thus being of real interest. It is important to know the limit temperature, e.g., in the annealing process of repeating unit  $C_{34}$  in the possible synthesis of  $D_5$ . Conversely, in the analysis of such diamondoids, the possible fragments appearing in the destruction of their lattice must be known.

#### **7.4 Conclusions**

In this chapter, structural stability of four seeds of the diamond  $D<sub>5</sub>$  and several substructures/fragments related to the  $D_5$  diamond were investigated. In the first case, it was evaluated both in static and dynamic temperature conditions by molecular dynamics (MD). During MD, the all-carbon  $C_{17}$  appeared the most resistant to changes in temperature. Structural and energetic stability of the other three seeds of  $D_5$  vary both with the values of temperature and evolution time in molecular dynamics and the arrangement of oxygen atoms in the molecules. Among all the studied structures, the  $D<sub>2</sub>$  isomer is the most sensitive to changes in temperature. After optimization by B3LYP,  $D_1$  isomer seemed to be the most stable one. The structure stability of  $D_1$  and  $P_1$  isomers in MD are similar. These two isomers are only slightly more sensitive to temperature as compared with the all-carbon  $C_{17}$ .

Other substructures/fragments related to the  $D_5$  diamond (and its relative L5 lonsdaleite) were constructed and investigated for stability, by both static PM3 calculation and molecular dynamics simulation procedures as well. The results show a good stability of several hyper-rings made from the small fullerenes  $C_{20}$  and  $C_{28}$ , modulated function of the central hollow, and the type consisting of small cages. At normal temperature, the hexagonal hyper-rings are more stable (as given by PM3 data), while at higher temperature, the pentagonal ones appear more stable, at least as isolated fragments. The substructures belonging to  $D_5$  and  $L_5$  showed a pertinent stability, possibly increased in the infinite corresponding lattice. The actual study employed the molecular dynamics simulation in finding the temperature limits for the most important events in a molecule: changes in topology and next its destruction.

These results could be useful in guiding further reactions, e.g., the dimerization to  $C_{34}$  and condensation to adamantane-like structures, finally leading to the diamond  $D<sub>5</sub>$  or, in general, in the design and synthesis of new strong structures, with possible applications in nanotechnology.

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