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Abstract

This chapter provides an overview of the materials used for manufacturing IPMC actuators and sensors. Recently, considerable effort has been put into investigating various electrode materials and ionic polymer membranes to increase the actuation performance of IPMC and overcome some of the shortcomings to improve their reliability and stability. Various metallic and nonmetallic electrode materials with notable electrochemical and electromechanical properties have

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been considered for IPMC electrodes. Herein, some of the more commonly used noble metal-based electrodes and recently introduced nonmetallic conductive material (such as transition metal oxide and various carbon derivatives)-based electrode designs along with specific manufacturing approaches are introduced, highlighting their key aspects and design challenges. Also, several representative ionic polymer membranes used for IPMC fabrication such as sulfonated aromatic hydrocarbon, block copolymers, biopolymers, and nanocomposites capable of providing higher electro-chemo-mechanical properties have been investigated. Herein, more recently developed membrane materials including self-assembled sulfonated polyimide block copolymers, functional cellulose-based biopolymers, and graphene-reinforced nanocomposites are introduced, considering their main advantages, facile synthesis process (such as freeze drying method, all-solution process, and electrospinning technique) and actuation performance.

Keywords

Carbide-derived carbons (CDCs) • Degree-of-sulfonation (DOS) • Direct assembly method • Electrode materials • Metallic electrode materials • Nonmetallic electrode materials • Freeze-dried bacterial cellulose (FDBC) • Graphene-Nafion polymer actuator • Impregnation-reduction method • Ionic polymer-metal composites (IPMCs) • Biopolymer membrane materials • Electromechanical properties of • Metallic electrode materials • Nanocarbon-composite membrane materials • Nonmetallic electrode materials • Sulfonated block copolymer membrane materials • Sulfonated hydrocarbon backbone membranes • Pendent sulfonated chitosan (PSC) • Polyaniline (PANI) • Sulfonated poly(amic acid) (SPAA) • Sulfonated poly(styrene-ran-ethylene) (SPSE) • Sulfonated polyimide (SPI)

1 Electrode Materials

Electrodes have a significant role in the electromechanical coupling of IPMC. Their physical properties such as electric conductivity, mechanical durability, and surface morphology can strongly affect the actuation performance and reliability of IPMC material. Proper electrodes for IPMC should be highly electrically conductive, mechanically compliant, durable against cyclic deformations, and electrochemically inert for operation in corrosive environment (e.g., water) in the presence of electric potential (~ 4 V). These design requirements limit considerably the choice of available materials. Various types of electrically conductive materials have been investigated for use in IPMC electrodes, including metals (primarily noble metals), transition metal oxides, and various carbon derivatives such as nanotubes, graphene, and nanoporous-activated and carbide-derived carbons. This chapter intends to provide an overview of these developments and current challenges.

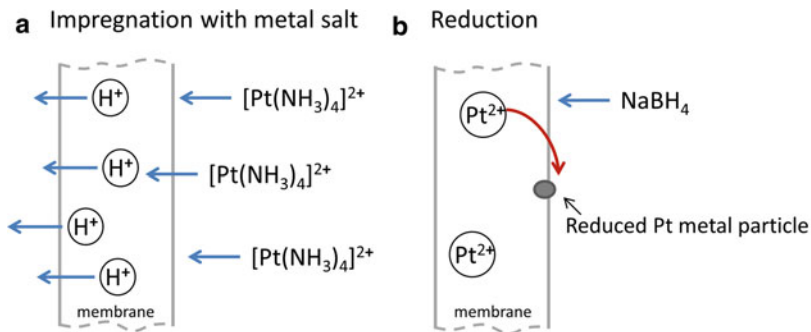
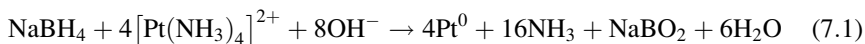


Fig. 1 Impregnation and reduction steps in the electroless plating process

1.1 Metallic Electrode Materials

Since the introduction of IPMC materials, the platinum and gold have been most widely used electrode materials for IPMC fabrication due to their high electrochemical stability and excellent electrical conductivity. These materials are fabricated into electrodes using chemical deposition method, also known as impregnation-reduction method or electroless plating (Shahinpoor and Kim 2001; Fujiwara et al. 2000). This method can be used with a wide range of ionic polymer membranes that have ability to selectively exchange cations or anions. The state-of-the-art manufacturing process of IPMC via electroless plating consists of three main steps. First, the ionic polymer membrane is impregnated with desired metal salt by soaking it in respective metal complex solution, such as tetraammineplatinum(II) chloride hydrate ($\text{Pt}(\text{NH}_3)_4\text{Cl}_2 \cdot \text{H}_2\text{O}$). The metal complex cations diffuse into the polymer through ion-exchange process as illustrated in Fig. 1. Secondly, the platinum complex cations in the polymer are reduced to metallic form at the membrane surface by a chemical reduction process in a solution containing sodium borohydride (NaBH_4). The reduction process occurs according the following reaction:



The described impregnation-reduction steps are typically repeated 1–3 times in order to increase the thickness and electric conductivity of the electrodes. The third step is the secondary plating (or surface electroding) process to deposit additional Pt at the outer surface of the electrode to further increase the electrode surface conductivity. The composite is immersed in the Pt complex solution, and by using reducing agents such as hydroxylamine hydrochloride and hydrazine monohydrate, the platinum is deposited on the top of the initial Pt electrode layer. A detailed description of the fabrication process can be found in (Kim and Shahinpoor 2003).

As can be seen in Fig. 2, the metallic platinum particles concentrate predominantly near the interface boundaries, typically within 5–20 μm range from the

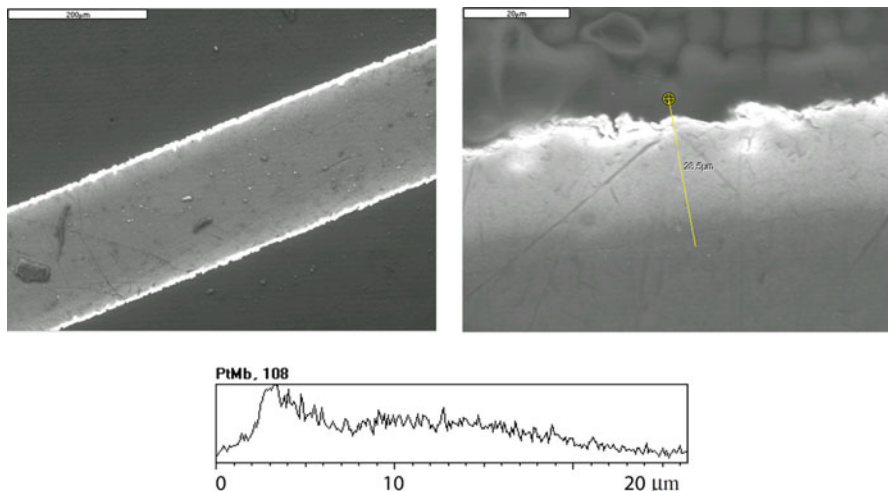


Fig. 2 Two SEM images (*top*) showing the cross section (*left*) and close-up (*right*) of a typical IPMC. The *bottom* graph shows an X-ray line scan of Pt. As can be seen, Pt is dense at the surface (Shahinpoor and Kim 2001. © IOP Publishing. Reproduced by permission of IOP Publishing. All rights reserved.)

surface. The plating conditions such as soaking time, concentrations, and temperatures are critical to the plating process and determine the penetration depth and conductivity of electrodes that affect the performance of IPMC (Kim and Shahinpoor 2003; Nemat-Nasser 2002). One of the concerns associated with Pt electrodes are developing microcracks under prolonged cyclic operation, leading to decrease in the electrode surface conductivity (Punning et al. 2007). Gold electrodes are more elastic and offer better mechanical stability and higher electrical conductivity, but on the other hand, their fabrication involves more complex processing (Fujiwara et al. 2000).

Several alternative materials and their combinations have been researched for IPMC electrodes. Using palladium as a supporting layer underneath the platinum layer has shown improved mechanical stability and higher transduction performance compared to the conventional platinum electrodes (Kim and Kim 2008; Palmre et al. 2014). In this electrode configuration, the Pd is deposited at the inner surface of the ionomer membrane prior to Pt surface layer using electroless plating method. With appropriate control of the plating conditions, highly dispersed Pd particles can be created not only at the ionomer surface, but deep in the polymer membrane (Fig. 3), thereby increasing notably the specific surface area of the electrodes (Palmre et al. 2014). This highly capacitive Pd electrode interface provides significant increase in the blocking force (Fig. 4) and mechano-electrical output of IPMC. However, the actuation response is slower compared to regular Pt electrodes.

IPMC electrodes composed of copper and platinum have been reported (Johanson et al. 2008), in which the reversible electrochemical processes – Cu dissolution and

Fig. 3 SEM micrograph of the cross sections of 1 mm thick Pd-Pt IPMC (*top*), and corresponding EDS line scan profile for elements of Pt and Pd (*bottom*) (Reproduced from Palmre et al. 2014)

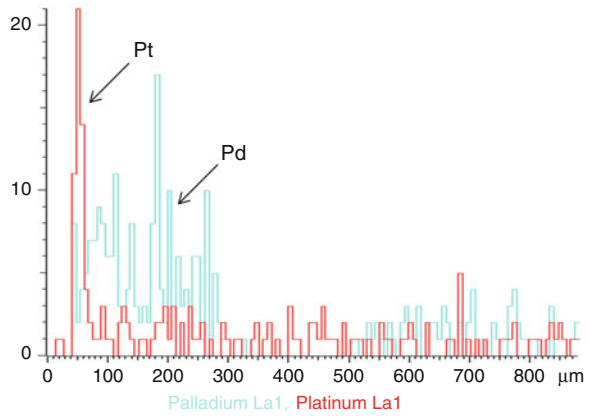
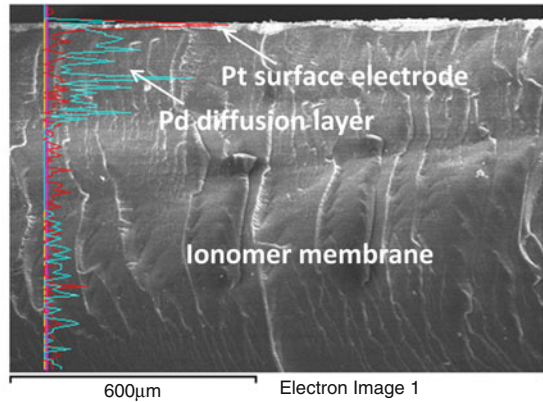
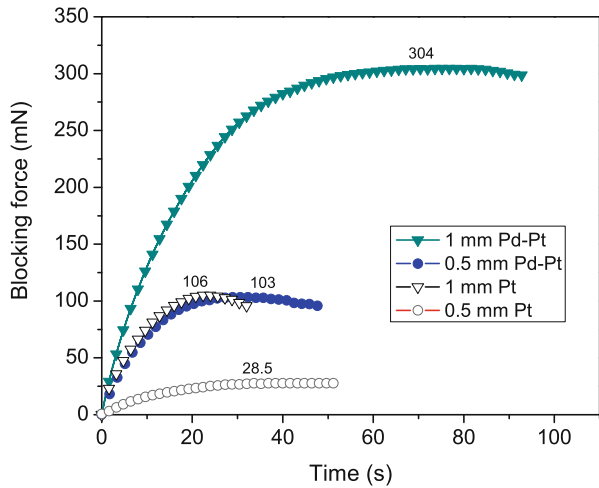


Fig. 4 Blocking force response in time for IPMCs with Pt and Pd-Pt electrodes at 4 V DC input (Reproduced from Palmre et al. 2014)



subsequent reduction of Cu^{2+} ions at the cathode upon actuation can maintain electrical conductivity between the platinum particles during the deformation. The drawbacks of this electrode configuration are the copper layer oxidation and growth of copper dendrites at the electrodes. Also, silver nanopowder and nickel have been used as cost-effective electrode materials (Chung et al. 2006; Siripong et al. 2006). However, the low electrochemical stability of these materials can limit the cycle life of IPMC.

From the aforementioned materials, the noble metals (Pt, Au, or Pd) are usually a preferred choice for IPMC electrodes due to their high electrical conductivity and electrochemical stability and availability in cation complex form that can be effectively used with electroless plating method.

1.2 Nonmetallic Electrode Materials

Recently, various nonmetallic conductive materials have gained interest as electrode materials for IPMCs. For instance, transition metal oxide powders (such as RuO_2) (Akle et al. 2007b), carbon nanotubes (Akle and Leo 2008; Lee et al. 2007), graphene (Kim et al. 2014b), highly porous-activated and carbide-derived carbons (Palmre et al. 2009), and carbon aerogels (Palmre et al. 2011) have been investigated for IPMC electrodes. The powder materials can be assembled into electrodes by physical heat-pressing using direct assembly method (Akle et al. 2007a) or casting technique (Fukushima et al. 2005). The direct assembly method consists of first mixing the conductive powder in Nafion-alcohol dispersion and then applying the conductor/ionomer mixture directly by painting on the surface of ionic polymer membrane (Fig. 5). Finally, the obtained composite is laminated between two gold foil layers by heat-pressing to conjugate the electrodes layers with the membrane (Fig. 6). Alternatively, the casting technique is based on casting the conductive powder/polymer mixture into individual electrode films and assembling them together by heat-pressing. Compared to the noble metals, the conductive powders are less expensive and have a large specific surface area, which is desired for creating high charge density at polymer-electrode interface. For instance, the specific surface area of carbon nanotubes can be up to $1000 \text{ m}^2/\text{g}$, while the highly porous carbide-derived carbons (CDCs) typically range from 1000 to $2000 \text{ m}^2/\text{g}$. CDCs are produced by extraction of metal ions from carbide precursor by chlorination at elevated temperatures (Presser et al. 2011). CDCs are also unique for their precisely controllable synthesis process that allows fine-tuning the pore dimensions according to electrolyte properties (Gogotsi et al. 2003). However, the mentioned powder materials have generally insufficient electric conductivity. Therefore, the electrode surface is covered additionally with a conductive gold foil layer (Akle et al. 2007a; Palmre et al. 2009). The added metallic layer makes the manufacturing process often more complicated and can cause problems due to the delamination under cyclic deformation (Akle et al. 2007b). The porous carbon materials, particularly activated carbons and carbide-derived carbons prepared at low chlorination temperatures, also tend to have poor electrical conductivity within the electrode cross section

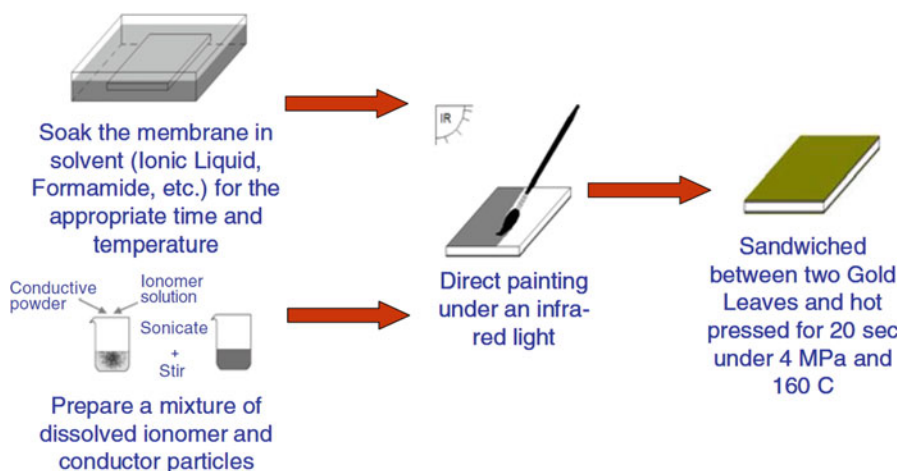


Fig. 5 Schematic showing the four steps of the direct assembly process for building IPMC materials (Reproduced from Akle et al. 2007a, with kind permission from Springer Science and Business Media.)

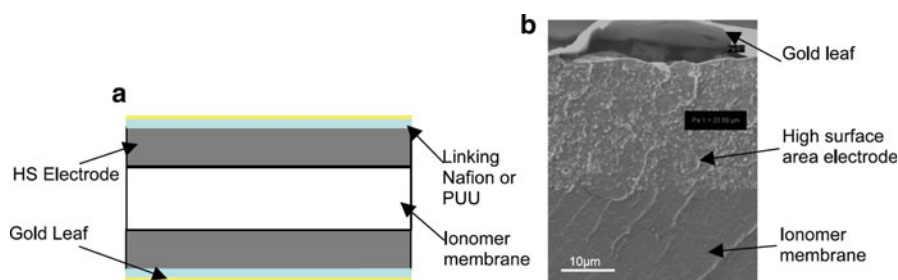


Fig. 6 (a) Schematic of IPMC assembled using direct assembly process. (b) SEM image of the upper high surface area RuO₂ electrode (Akle et al. 2007b. © IOP Publishing. Reproduced by permission of IOP Publishing. All rights reserved)

(Palmre et al. 2012). This results in a slow electrical double-layer charging and a slow electromechanical response of an actuator. Also, it has been noted that activated carbons, despite having a large specific surface area, exhibit a limited capacitance due to their low mesoporosity and resulting poor electrolyte accessibility (Lu et al. 2011). Therefore, conductive additives such as nanotubes, carbon black, and polyaniline (PANI) have been added in the carbon electrodes to increase the mesoporosity as well as electric conductivity, thereby improving considerably the performance of the electrodes (Lu et al. 2011; Sugino et al. 2011).

Several studies have indicated that the accumulation of electrolyte charges in the vicinity of electrodes is the key to producing a high strain and force output of IPMC (Akle et al. 2006; Wallmersperger et al. 2008; Pugal et al. 2011).

Table 1 Electromechanical properties of IPMCs with different conductive electrode materials (Palmre et al. 2009)

Electrode material	Maximum peak-to-peak strain (mε)	Maximum strain rate (me s ⁻¹)	Blocking force (mN)
Carbide-derived carbon	20.4	2.3	3.6
Coconut shell-based activated carbon	10.3	1.6	3.1
RuO ₂ (anhydrous)	9.4	0.93	3.3
RuO ₂ (hydrous)	6.9	1.7	3.5

Therefore, designing the electrodes with high specific surface area is gaining momentum in the development of IPMC materials (Torop et al. 2011; Palmre et al. 2011, 2014; Sugino et al. 2011). Promising electrode materials for this purpose are the mentioned highly porous carbons and carbon nanotubes. The CDC-based IPMC electrodes have shown one of the best performances among nonmetallic electrodes, exceeding the peak strain and strain rate of RuO₂ electrodes more than twice (Table 1) (Palmre et al. 2009). However, there are still challenges to overcome such as insufficient electric conductivity and limited electrolyte diffusion in the carbon electrode structure. Also, it is important to note that the nonmetallic powder materials require physical assembly by heat-pressing and therefore are mainly suited for fabricating “dry-type” actuators based on nonvolatile electrolytes such as ionic liquids (Akle et al. 2007b). One of the challenges with using the ionic liquids as electrolytes is the lower response speed compared to the water-solvated membranes (Bennett and Leo 2004). Therefore, water-based IPMCs manufactured by electroless plating of noble metals that offer one of the fastest strain responses and ability to be operated in aqueous environment are more suited for underwater robotic applications.

2 Ionic Polymer Membranes

Ionic polymer membranes play an important role in determining actuator performances. Especially, their electro-chemo-mechanical properties, such as ion transport properties (ionic-exchange capacity (IEC), liquid electrolyte uptake and ionic conductivity, etc.) and the mechanical properties (tensile modulus, strength, and elongation, etc.), must be evaluated carefully to enhance the actuator performances. As the most popular materials used for IPMC membranes, perfluorinated polymers, such as Nafion and Flemion with ionic sulfonate or carboxylate groups, respectively, are widely investigated. Nafion is a perfluorinated sulfonic acid ionomer membrane consisting of a Teflon-like backbone and short side chains terminated by hydrophilic sulfonic acid groups. However, these perfluorinated ionomer membranes suffer from significant drawbacks, such as low actuation bandwidth, low blocking force and durability, environmentally unfriendliness of fluorinated polymer, and high cost of fabrication, which curtail their practical applications. To overcome these problems, many synthetic ionic polymers have been proposed as leading candidates. As one of

alternative ionic polymers, sulfonated hydrocarbon polymers have received significant amount of attention due to their cost-effectiveness, easy fabrication, tunable stiffness, and high ion transport properties, resulting from their controllable monomer composition, especially via manipulation of block copolymers. Another solution is to utilize naturally abundant functional biopolymers, such as cellulose-derivatives and chitosan with high ionic conductivity, environmental friendliness, low cost, and uniform film formation. Yet another solution is to reinforce functional nanoparticles in a polymer matrix, resulting in a high-performance nanocomposite membrane. This section intends to provide an overview of the recent research developments in ionic polymer membranes.

2.1 Sulfonated Hydrocarbon Backbone Membranes

A great deal of effort has been invested in the development of sulfonated aromatic or aliphatic hydrocarbon backbone polymers by adopting the post sulfonation process, blending method, and cross-linking procedure to control hydrophilic-hydrophobic morphology and electro-chemo-mechanical properties. For instance, sulfonated poly(styrene-ran-ethylene) (SPSE, Wang et al. 2010a) with aliphatic hydrocarbon backbone, cross-linked SPSE (XSPSE, Wang et al. 2010c) with better microphase-separated morphology by UV irradiation, and sulfonated aromatic PEI (polyetherimide) (SPEI, Rajagopalan et al. 2010) with controllable stiffness have been investigated as alternative IPMC membranes. Also, several ionic networking membranes have been developed through blending and cross-linking methods. Poly(styrene-alt-maleimide) (PSMI)-incorporated poly(vinylidene fluoride) (PVDF) (PSMI/PVDF, Lu et al. 2008a, b) exhibited several times larger bending performance than the Nafion counterpart due to unique hydrophilic nanochannels. Sulfonated poly(ether ether ketone)-incorporated PVDF (SPEEK/PVDF, Jeon et al. 2009) showed excellent electromechanical responses due to the tailored stiffness and nanochannels inside the ionic networking matrix. Cross-linked PVA/SPTES (Wang et al. 2010b) was developed by physical cross-linking between SPTES (sulfonated poly(arylenethioethersulfone)) copolymer and PVA (polyvinyl alcohol), resulting in an absence of back-relaxation and a dramatic increase in bending deformation. Different types of hydrocarbon backbone membranes have also been presented and compared (Jo et al. 2013). However, these hydrocarbon series still have limitations that do not satisfy the desired electro-chemo-mechanical properties and fail to deliver better actuation performances beyond those of Nafion (Fig. 7).

2.2 Sulfonated Block Copolymer Membrane Materials

Several sulfonated block copolymers such as SSEBS (sulfonated poly(styrene-*b*-ethylene-co-butylene-*b*-styrene)) block copolymer (Wang et al. 2007), ABA-Triblock copolymer (Imaizumi et al. 2012), sulfonated pentablock ABCBA copolymers (Gao et al. 2012), and sulfonated pentablock ionomer (PBI)-based

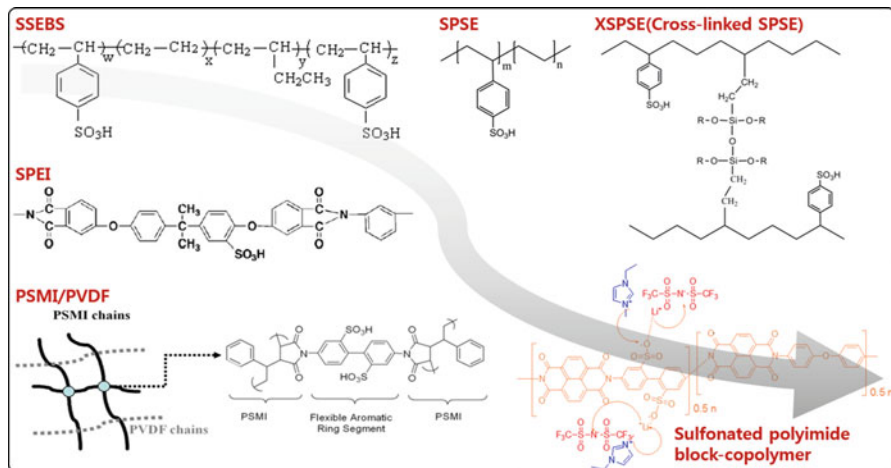


Fig. 7 Several hydrocarbon backbone membranes for preparing IPMC membrane materials (SSEBS, Wang et al. 2007. © 2007 Elsevier Ltd. Reproduced by permission of Elsevier BV.; SPSE, Wang et al. 2010a. © 2010 Society of Chemical Industry. Reproduced by permission of John Wiley & Sons, Inc.; SSPSE, Wang et al. 2010c. © 2010 Elsevier Ltd. Reproduced by permission of Elsevier BV.; SPEI, Rajagopalan et al. 2010. © 2010 Elsevier Ltd. Reproduced by permission of Elsevier BV.; PSMI/PVDF, Lu et al. 2008a. © 2008 WILEY-VCH Verlag GmbH & Co. Reproduced by permission of John Wiley & Sons, Inc.; SPI, Cheedarala et al. 2014. © 2014 WILEY-VCH Verlag GmbH & Co. Reproduced by permission of John Wiley & Sons, Inc. All rights reserved.)

copolymers (Vargantwar et al. 2012), which contain hydrophilic nanochannels and well-organized nanostructure networks with micro-/nanomorphology, have been investigated to enhance actuator performance. Unfortunately, these sulfonated block copolymer membranes have difficulties in fabrication, which include a complex synthesis procedure and low mass production, and deficiencies in their electrochemo-mechanical properties. Therefore, an alternative synthetic approach is needed to develop novel and simple block copolymers with intriguing nanostructures in a polymer matrix. As one of promising block copolymers, sulfonated polyimide (SPI) block copolymers contain alternate hydrophobic and hydrophilic multiblocks having aliphatic and aromatic segments and display high thermal stability, high ionic conductivity, reliable mechanical properties, and low price. Recently SPI-based actuator with well-defined silver electrodes was developed via an in situ self-metallization (Song et al. 2011). The total fabrication procedure involves the synthesis of a sulfonated poly(amic acid) (SPAA) membrane precursor for SPI derived from BTDA, ODA, and lithium-containing BDSA as a dianhydride and diamine, respectively. After the ionic-exchange process with the Ag salt, the SPAA-Ag⁺ membrane was subjected to thermal treatment, resulting in imidization of SPAA and simultaneous reduction of silver cations to silver metal layer on both sides of the membrane as shown in Fig. 8a, b.

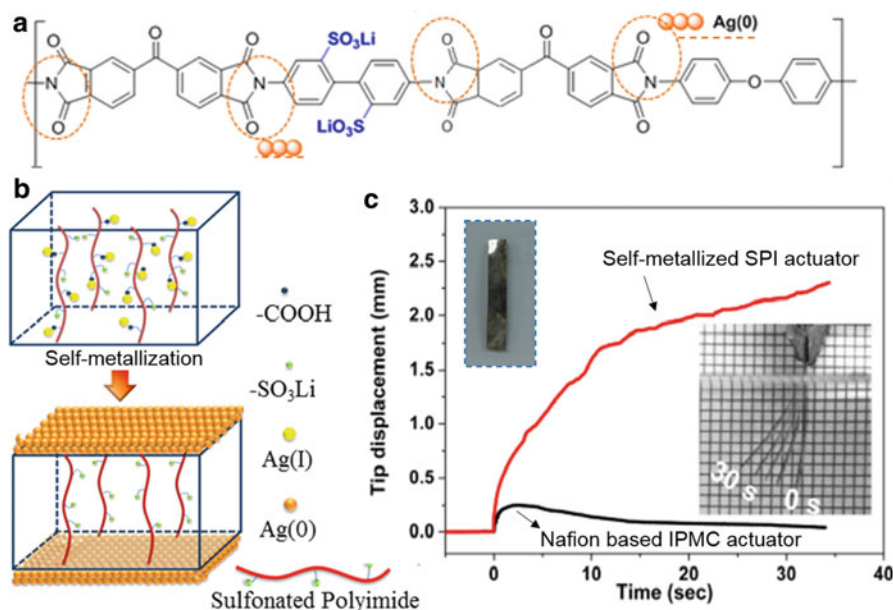


Fig. 8 (a) Chemical structure of SPI with self-metallized silver electrodes, (b) schematic illustration of self-metallization process, (c) step responses of self-metallized SPI actuator and Nafion-based IPMC actuator under 0.5 V DC voltage (Song et al. 2011. © 2011 WILEY-VCH Verlag GmbH & Co. Reproduced by permission of John Wiley & Sons, Inc. All rights reserved.)

Compared with a Nafion-based actuator, the self-metallized SPI actuator with highly conductive silver electrodes showed a much larger tip displacement without the back-relaxation phenomenon under low 0.5 V DC voltage (Fig. 8c). But further investigation for high-performance actuator is needed due to its moderate electroactive performance and poor durability, resulting from the oxidation of silver layers under higher voltage stimulation. Furthermore, SPI-based actuators from a combination of ionic liquid, polyimide, and carbon electrode materials were reported (Imaizumi et al. 2013). But, until now, SPI-based polymer actuators, which have relatively high mechanical properties, low actuation performance, and inferior durability due to densely packed polymer matrix and low compatibility with metallic electrodes fabricated by electroless plating, have not been improved.

Very recently a self-assembled 3D ionic networked SPI polymer actuator with π - π stacked layers and alternate hydrophilic nanochannels have been investigated by introducing simple and strong atom-level regio-specific interaction of hydrophilic and hydrophobic SPI coblocks with anions and cations in the ionic liquid (EMI. Tf₂N) (Cheedarala et al. 2014). Moreover, facile and ultrafast all-solution process involving solvent blending, dry casting, and drop casting was developed to make all-organic soft actuators with highly conductive and flexible PEDOT:PSS electrodes (Fig. 9a). This higher bending performance of the newly developed Ntda-SPI-SO₃Li-EMI.Tf₂N is induced by the high ionic conductivity and tuned

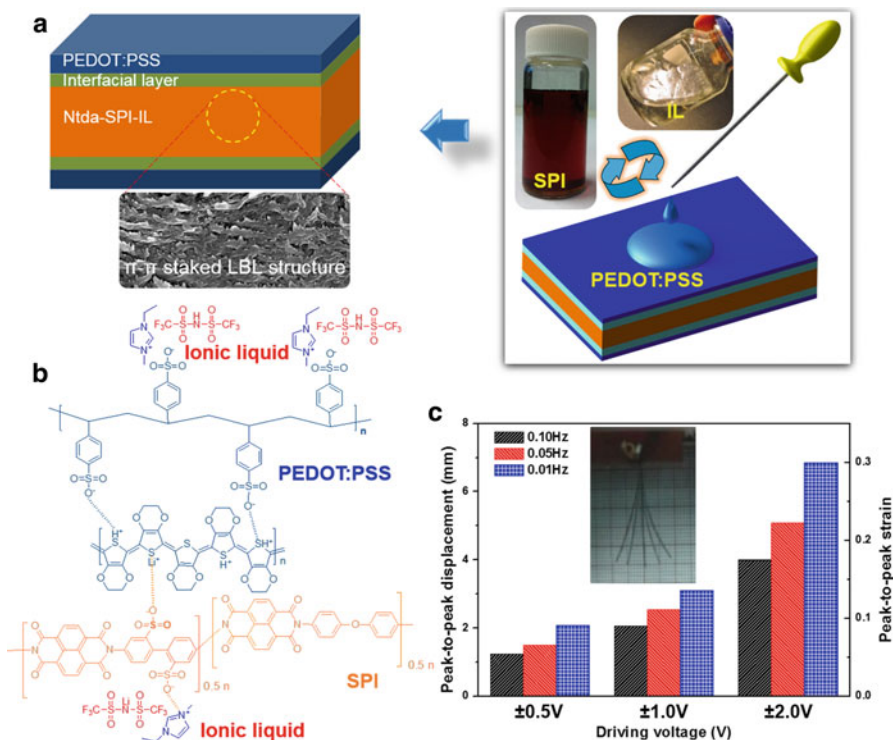


Fig. 9 (a) Schematic illustration of SPI actuator based on a π - π stacked 3D ionic network membrane and ultrafast solution processing, (b) chemical interaction among SPI, ionic liquid, and PEDOT:PSS, (c) peak-to-peak strains at different voltages and frequencies (Cheedarala et al. 2014). © 2014 WILEY-VCH Verlag GmbH & Co. Reproduced by permission of John Wiley & Sons, Inc. All rights reserved.)

mechanical properties, resulting from strong ionic interactions among the SO_3Li , ionic liquid and PEDOT:PSS, and π - π stacked 3D-networked polymer matrix with continuous and interconnected ion transport nanochannels (Fig. 9b). Therefore, large peak-to-peak strains of over 0.05 were acquired in all frequency ranges over ± 0.5 V as shown in (Fig. 9c).

2.3 Biopolymer Membrane Materials

Development of high-performance biopolymer membranes has been attempted to solve the problems of conventional synthetic ionic polymers. As one of efforts, well-defined and regular electrospun PANI/CA biopolymer actuators with a good dispersion of chopped PANI (polyaniline) nanoparticles inside the CA (cellulose acetate) nanofibers were reported (Hong et al. 2013). Only a small amount of chopped PANI nanoparticles (0.1 % and 0.5wt%), inducing well-dispersed nanoporous structures

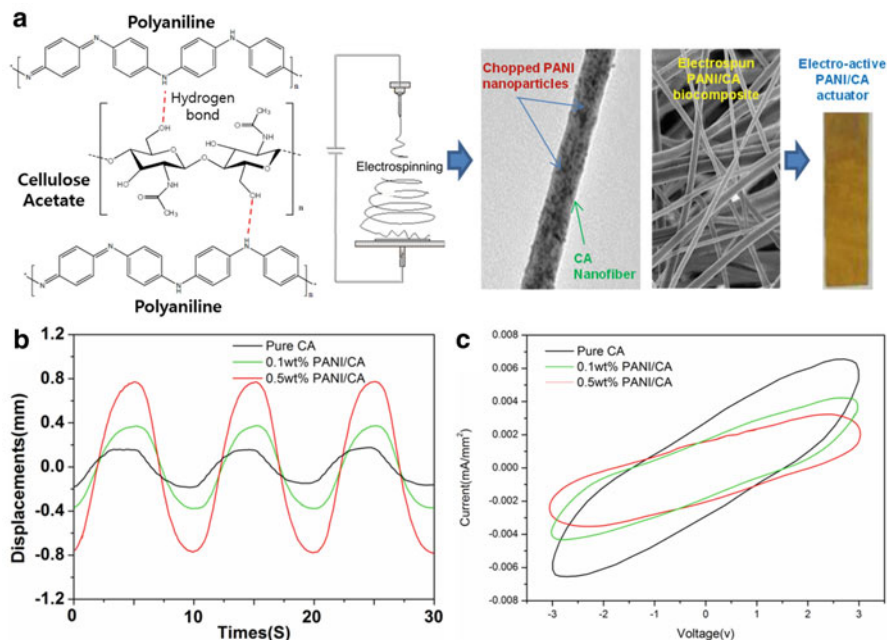


Fig. 10 (a) Schematic diagram of fabrication process for electrospun PANI/CA actuators, (b) harmonic responses, (c) hysteresis responses of electrospun actuators under sinusoidal electrical inputs with peak voltage of 3 V and excitation frequency of 0.1 Hz (Hong et al. 2013). © 2013 Elsevier Ltd. Reproduced by permission of Elsevier BV. All rights reserved.)

(Fig. 10a), can improve the actuation performance of cellulose-based actuators due to the enhanced material properties as presented in Fig. 10b. Moreover, the consumed power of 0.5wt% PANI/CA actuator is lower than that of others because of the effect of highly conductive PANI nanoparticles and their good dispersion in CA matrix (Fig. 10c). Therefore, this investigation suggests that the electrospinning is an extremely practical and effective technique for constructing a series of well-dispersed nanoporous membranes with controllable and repeatable products, unlike conventional casting method.

To make high-performance air-working actuators, ionic liquids as mobile salts should be embedded in the cellulose membrane. But pure cellulose is not familiar with absorbing much ionic liquid because of its packed structure and high crystallinity. Therefore, freeze-dried bacterial cellulose (FDBC) with sponge-like porous network structure through a simple freeze-dry method at -50°C for 24 h can be used to solve these problems. Because FDBC with high porosity and large surface area can absorb a large quantity of ionic liquid and exhibit enhanced electrochemical properties. A novel high-performance electro-active biopolymer actuator with a tri-layered sandwich structure that consists of FDBC and poly(3,4-ethylenedioxythiophene) poly(styrenesulfonate) (PEDOT:PSS) face sheet layers was developed as presented in Fig. 11a (Kim et al. 2013). The actuation

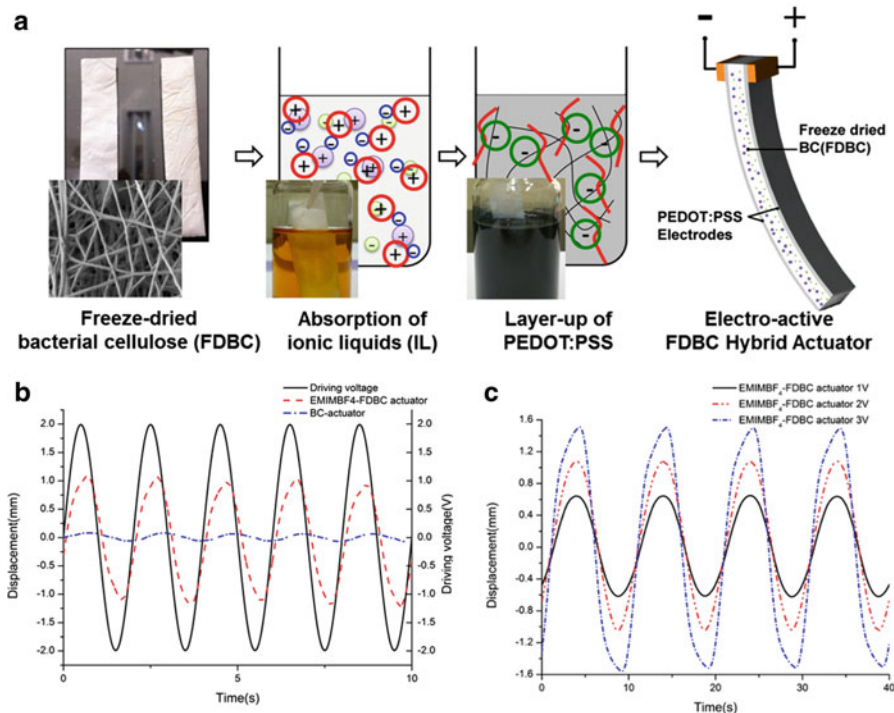


Fig. 11 (a) Fabrication steps of the FDBC actuator, (b) actuation performance of a FDBC actuator compared with a pure BC actuator, (c) harmonic response of FDBC actuators using EMIM-BF₄ (Kim et al. 2013. © 2013 IOP Publishing Ltd. Reproduced by permission of IOP Publishing. All rights reserved.)

performance of the developed actuator was significantly improved through the synergistic effects of the electrochemical doping processes of PEDOT:PSS electrode layers and sufficient ion migration of the dissociated ionic liquids inside the FDBC with a sponge form. So the FDBC actuator exhibited a much larger bending deformation than pure bacterial cellulose actuator as shown in Fig. 11b. Also, under sinusoidal excitation with an excitation frequency of 0.1 Hz and peak voltage of 3 V, the EMIM-BF₄ absorbed FDBC actuator, inducing much larger bending deformations, reached a peak tip displacement of ± 1.511 mm (Fig. 11c).

2.4 Nanocarbon-Composite Membrane Materials

Reinforcement by the incorporation of electrically conductive 0D/1D/2D/3D carbon nanoparticles into polymer matrix is a promising approach to achieve improved electro-chemo-mechanical properties including enhanced ionic transport properties and tunable mechanical stiffness over pristine polymers. Therefore, conductive

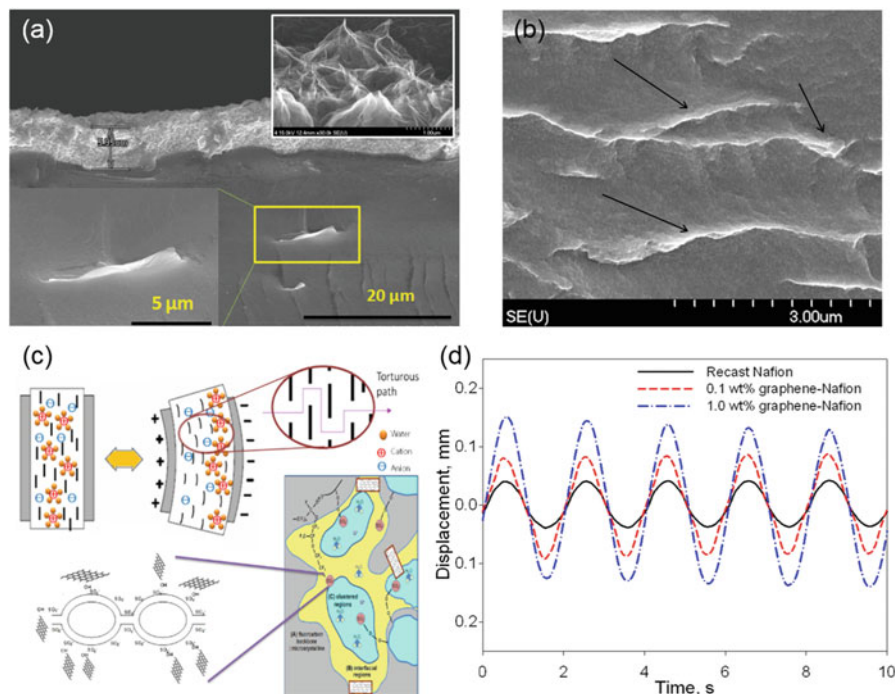


Fig. 12 Graphene-Nafion polymer actuator; (a) cross-sectional SEM image, (b) magnified SEM image, (c) schematic representation of ionic conductivity in hydrated graphene-Nafion composites, (d) harmonic response (Jung et al. 2011). © 2010 Elsevier Ltd. Reproduced by permission of Elsevier BV. All rights reserved.)

carbon nanoparticles such as 0D (fullerene, polyhydroxylated fullerene (PHF)), 1D (SWNT, MWNT, CNF), 2D (Graphene), and 3D (hierarchical G-CNT-Fe/G-CNT-Pd carbon nanostructure) have been reinforced into ionic polymer membranes to enhance their physicochemical properties and actuation performance. Several nanocomposite actuators have been investigated, i.e., (i) 0D carbon nanostructures: C₆₀-reinforced Nafion (Jung et al. 2010; Oh et al. 2010), fullerenol-cellulose (Li et al. 2011), and fullerenol-SPEI (Rajagopalan and Oh 2011), (ii) 1D carbon nanostructures : CNF-reinforced SSEBS (Wang et al. 2009), (iii) 2D carbon nanostructures : Graphene-Nafion (Jung et al. 2011), Graphene oxide-chitosan (Jeon et al. 2013), and Ionic polymer-graphene composite (IPGC, Kim et al. 2014b), (iv) 3D carbon nanostructures : 3D G-Fe-reinforced polyurethane (Lee et al. 2014), and 3D G-Pd-reinforced conducting polymer (Kim et al. 2014a).

Especially graphene is an intriguing 2D flat material of monolayer carbon atoms whose distinct properties make it very promising in actuator applications. Recently, a graphene-reinforced Nafion nanocomposite actuator was developed as shown in Fig. 12a (Jung et al. 2011). The layered structure of graphene in the graphene-Nafion composites showed that graphene layers are aligned parallel to the surface of

membrane and can provide tortuous pathway for ion transport (Fig. 12b, c). Although higher graphene concentration showed a decrease in water adsorption caused by the reduction of degree of ion clustering due to a decrease in the cluster size and the exchange sites per cluster, a marginal increase in IEC with increasing graphene was observed, thereby leading to enhanced actuation performance. The tip displacement of the 1.0 wt% graphene-reinforced actuator is almost two times that of the recast Nafion-based IPMC actuator (Fig. 12d). These results demonstrate that the electro-chemo-mechanical properties and actuation performances were significantly improved due to a minute loading of graphene, resulting from the great interaction between graphene and Nafion.

In order to develop an economically viable, highly durable, and bio-friendly ionic polymer actuator with superior electro-chemo-mechanical properties, a simple route for a high-performance ionic bio-nanocomposite based on pendent sulfonated chitosan (PSC) and functionalized graphene oxide (GO) have been investigated (Jeon et al. 2013). The amine groups in the chitosan were actively used to tune the degree of sulfonation (DOS) of PSC by controlled reaction with 1,3-propyl sultone and with GO. Thus, PSC can act as a higher ionic-exchangeable membrane due to the availability of propyl sulfonic acid groups that can strongly binding with free amines and with ionic liquid. Furthermore, electro-chemo-mechanical activities of PSC can be maximized by simply reinforcing with GO, resulting in improving in-plane mechanical stiffness and electro-chemo-mechanical properties through strong ionic interactions and bonding with free amines and sulfonic acid groups (Fig. 13a). Minute loading of graphene oxide (1.0 wt%) to PSC greatly improves step responses without the back-relaxation phenomenon (Fig. 13b, c). This improvement can be attributed to relatively higher ionic conductivity and large capacitance, and effective delay of ionic liquid migration in the reduced ionic pores homogeneously distributed in GO(1.0 wt%)-PSC-IL matrix.

Recent attention to the correlation of morphological structure (microphase-separated morphology, well-dispersed porous structures and self-assembled nanostructures, etc.) with ion migration mechanisms to enable more efficient ion transport has led to new ways to make high-performance IPMC actuators. There are several research approaches to easily controlling hydrophobic and hydrophilic composition, size of ionic nanochannels, and their stiffness of ionic polymer membranes as well as modulating significant factors such as their ionic-exchange capacity, degree of sulfonation, capacitance, ionic conductivity, liquid electrolyte uptake, and mechanical properties. As one of the recent advances in ionic polymer membranes, herein we briefly introduced promising alternative ionic polymer membranes such as self-assembled sulfonated polyimide block copolymers, functional cellulose-based biopolymers, and graphene-reinforced nanocomposites, and technical considerations including freeze drying method, all-solution process, electrospinning technique, and reinforcement of conductive nanoparticles. Especially, the sulfonated 3D-networked porous structure membranes based on sulfonated polyimide block copolymer and pendent sulfonated chitosan biopolymers can be promising candidates for high-performance ionic polymer membrane materials. Furthermore, electrically conductive and hierarchical nanoparticle additives like 3D carbon

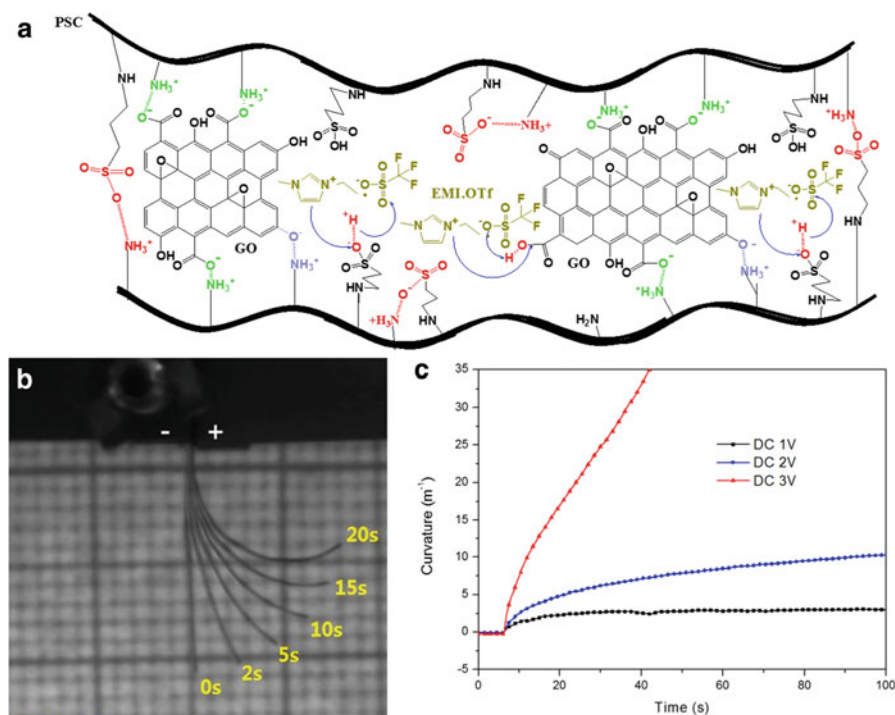


Fig. 13 (a) Synthesis of GO-PSC-IL bio-nanocomposite membrane, (b) electro-mechanically deformed shapes of GO(1.0 wt%)-PSC-IL actuator under a step input, (c) step responses of GO(1.0 wt%)-PSC-IL actuator according to driving voltages (Jeon et al. 2013. © 2013 WILEY-VCH Verlag GmbH & Co. Reproduced by permission of John Wiley & Sons, Inc. All rights reserved.)

nanostructures can greatly increase the output tip displacement, generated blocking force and energy efficiency even more. However, there is still need for more challenges and research studies to realize full potential of the ionic polymer-based high-performance IPMCs and overcome their insufficient controllable structure, material properties, and low repeatability and robustness of the actuation performance.

3 Conclusions

In this chapter, recent developments in electrode materials and ionic polymer membranes used for manufacturing IPMCs were reviewed. Although noble metals such as platinum and gold are commonly used for electrodes in water-based systems and applications for their excellent electrochemical properties, also various nonmetallic conductive carbon derivatives are considered as promising alternatives for fabricating “dry-type” IPMC actuators. These carbon derivatives include nanotubes and nanoporous-activated and carbide-derived carbons. While their electric conductivity

is generally less than that of noble metals, the mentioned carbon materials offer some important qualities such as high specific surface area and lower cost.

In spite of wide popularity of Nafion membranes, other less expensive and better-performing ionomers have been explored for IPMCs. One of the alternatives is sulfonated hydrocarbon polymers that offer cost-effectiveness, easy fabrication, tunable stiffness, and high ion transport properties, resulting from their controllable monomer composition. Also, naturally abundant functional biopolymers, such as cellulose-derivatives and chitosan with high ionic conductivity, environmental friendliness, and low cost, have been used for IPMC membrane. A promising approach is also reinforcement by incorporating functional carbon nanoparticles, such as fullerenes, nanotubes, and graphene, in a polymer matrix, resulting in a high-performance nanocomposite membrane.

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