

Chapter 3

Effects of Chain Extender on Dynamic Properties of PLLA/PBSL Blends

Masahiro Nishida, Yuma Takeuchi, Shun Furuya, Tetsuo Takayama, and Mitsugu Todo

Abstract The effects of the additive amount of a chain extender, with a weight ratio of 3–5 %, on stress-strain curves of poly(lactic acid) (PCL) and poly(butylene succinate-co-L-lactate) (PBSL) polymer blends were examined with a mixing ratio of PLA to PBSL of 70/30. The additive increased the elongation at break, and the strain rate decreased elongation at break.

Keywords Bioplastics • Strain rate • Tensile split Hopkinson pressure bar method • Polymer alloy • Crosslinking agent

3.1 Introduction

Biodegradable plastics and bioplastics are being investigated extensively, and new biodegradable plastics and bioplastics continue to be developed. Poly(lactic acid) (PLA) is a typical biodegradable bioplastic. Many studies have been conducted to develop various industrial products that can be manufactured using biodegradable plastics and bioplastics. However, such applications are currently limited to machine parts that are not subjected to high loading. Because PLA is brittle, its mechanical properties have been improved by polymer blends or natural-fiber reinforcement [1–3]. Because poly(butylene succinate-co-lactate) (PBSL) is a ductile and biodegradable polymer, PLA/PBSL polymer blends have great potential for high strength and toughness [4–7]. In order to increase their compatibility, a chain extender, Joncryl (ADR4370S, BASF), was added.

In this study, we examined the effects of different amounts of the chain extender (crosslinking agent) additive, 3–5 %, on dynamic and static tensile properties, with a mixing ratio of PLA to PBSL of 70/30, using a tensile split Hopkinson bar method and a universal testing machine. After dynamic and static tensile tests, the fracture surfaces of the specimens were observed.

3.2 Experimental Methods

3.2.1 Materials

Polymer blends of PLA and PBSL were prepared using PLA pellet from Nature Works Co. (Ingeo, 3100D) and PBSL pellet from Mitsubishi Chemical Co. (GS Pla[®] AZ91T). Because PBSL is ductile, there are many possibilities for high mechanical properties (for example, high impact strength) in polymer blends with PLA. Figure 3.1 illustrates the chemical structural formula of PLA and PBSL.

The mixing ratio (mass fraction) of PLA to PBSL was fixed at 70/30. During melt mixing, the chain extender (BASF, Joncryl ADR 4370-S) of the weight ratio of 3, 4, or 5 % was added. After melt mixing in a conventional melt mixer at 190 °C for 20 min, the blend mixtures were press processed using a conventional hot press at 190 °C and 30 MPa for 30 min.

M. Nishida (✉) • Y. Takeuchi • S. Furuya

Department of Mechanical Engineering, Nagoya Institute of Technology, Gokiso-cho, Showa-ku, Nagoya, Aichi 466-8555, Japan
e-mail: nishida.masahiro@nitech.ac.jp

T. Takayama

Graduate School of Science and Engineering, Yamagata University, 4-3-16 Jonan, Yonezawa, Yamagata 992-8510, Japan

M. Todo

Research Institute for Applied Mechanics, Kyushu University, 6-1 Kasuga-koen, Kasuga, Fukuoka 816-8580, Japan

Fig. 3.1 Chemical structural formula. (a) PLA. (b) PBSL

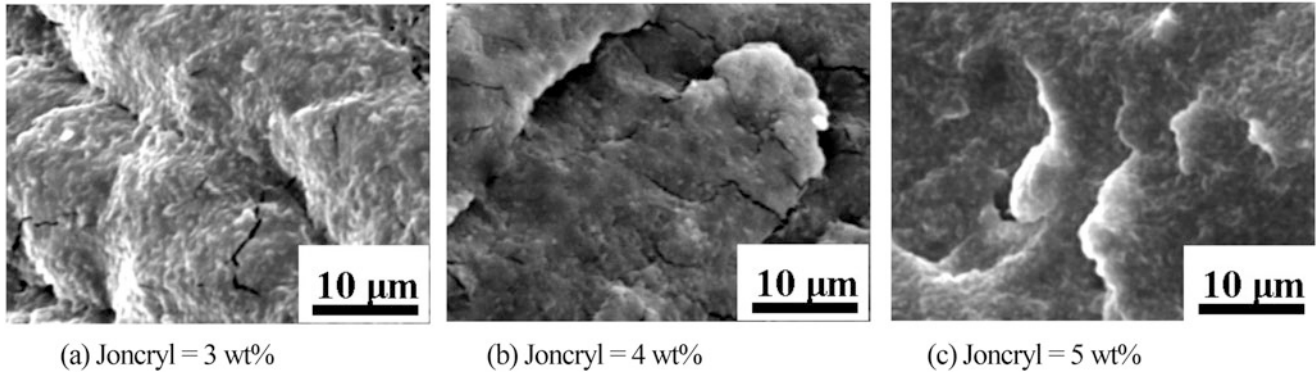
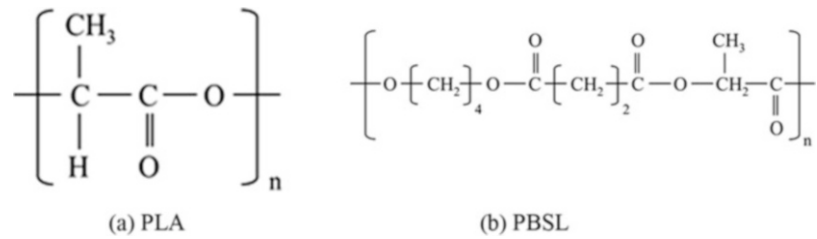
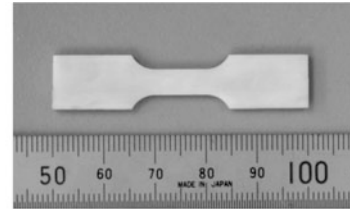
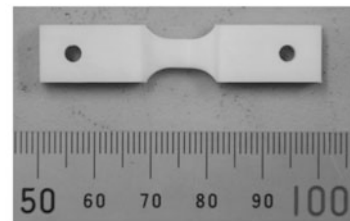


Fig. 3.2 SEM photographs before tensile tests. (a) Joncryl = 3 wt%. (b) Joncryl = 4 wt%. (c) Joncryl = 5 wt%

Fig. 3.3 Photograph of tensile test specimens. (a) Static tensile test (gauge area 2 mm by 5 mm, gauge length 10 mm). (b) Dynamic tensile test (gauge area 2 mm by 5 mm, gauge length 4 mm)



(a) Static tensile test (gauge area 2mm by 5mm, gauge length 10 mm)



(b) Dynamic tensile test (gauge area 2mm by 5mm, gauge length 4 mm)

Photographs of the cryo-fractured surfaces, taken using a scanning electron microscope (SEM), are shown in Fig. 3.2. Clear two-phase structures were not observed by a scanning electron microscope when the chain extender was added. The chain extender improved the compatibility of PLA and PBSL.

3.2.2 Static Tensile Tests

Quasi-static tensile tests were conducted at a crosshead speed of 0.2 mm/min (strain rate of 0.9×10^{-4} to $2.2 \times 10^{-4} \text{ s}^{-1}$) using a universal testing machine. The strain was calculated by images of the gauge area taken by digital camera. Dimensions of the specimens of static and dynamic tensile tests are shown in Fig. 3.3a.

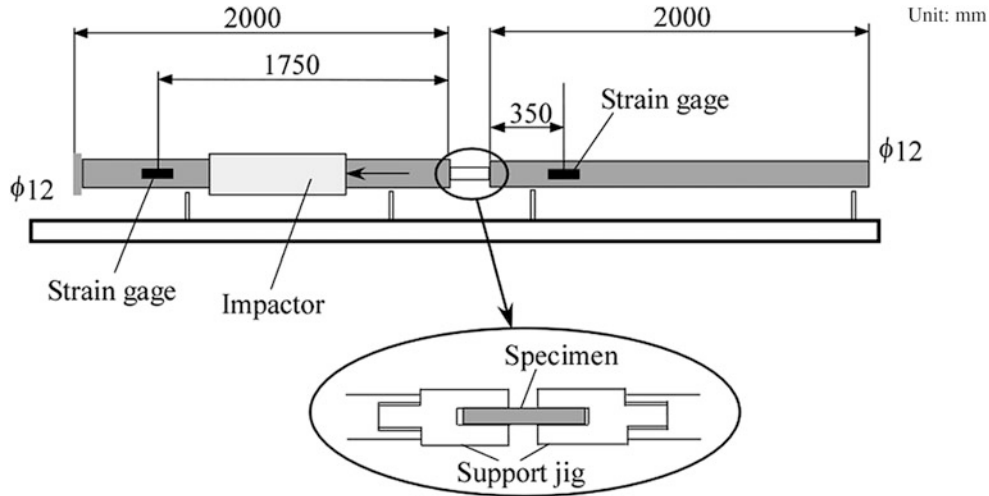


Fig. 3.4 Experimental setup for tensile Hopkinson bar (Kolsky bar) method

Table 3.1 Material constants of input and output bars used in calculations

	Density [kg/m ³]	Elastic wave velocity in bar, c_3 [m/s]	Young's modulus E [GPa]
Stainless steel, SUS304	8.0×10^3	4970	200

3.2.3 Dynamic Tensile Tests

At high strain rates, the dynamic properties of the specimens were examined by the tensile split Hopkinson bar test, as shown in Fig. 3.4. The diameters and lengths of input and output bars were 12 and 2000 mm respectively. Strain gauges were applied to both sides of the input and output bars at distances of 1750 and 350 mm from the specimen, respectively. The stresses and strains of the specimens [8, 9] were calculated from the strains of the bars measured by strain gauges, based on the following equations:

$$\sigma(t) = \frac{AE}{A_s} \varepsilon_T(t) \quad (3.1)$$

$$\varepsilon(t) = \frac{2c_3}{L} \int_0^t [\varepsilon_I(t) - \varepsilon_T(t)] dt \quad (3.2)$$

where ε_I and ε_T are the axial strains in the input bar induced by the incident wave and the axial strain in the output bar induced by the transmitted wave, respectively. E and c_3 are the Young's modulus and the elastic wave velocity of the input and output bars, respectively. L is the specimen's thickness. A and A_s are the cross-sectional areas of the input/output bars and the specimens. The input and output bars were made of stainless steel (SUS304), and their material constants used in the calculations are listed in Table 3.1. We preserved the specimens in a desiccator at a humidity of 30–40 % until just before use in order to avoid the effects of moisture absorption on the specimens. During the experiments, our laboratory was maintained at a temperature of 25 ± 2 °C and a humidity of 19–54 %. Dimensions of the specimens of dynamic tensile tests are shown in Fig. 3.3b.

3.3 Results and Discussion

The nominal stress-strain curves of the static tensile tests (Fig. 3.5) showed strain softening (work softening), which is defined as the reduction of stress with an increase in strain. When the additive was 5 %, the strain softening was not clear. The maximum stress with the 4 % additive was larger than those of 3 and 5 %. The elongation at break was almost the same among 3, 4 and 5 % additive ratios. At high strain rates, the stress-strain curves of the dynamic tensile tests (Fig. 3.6) did not show strain softening (work softening). The maximum stress and elongation at break were compared (Figs. 3.7 and 3.8);

Fig. 3.5 Stress-strain curves of static tensile test

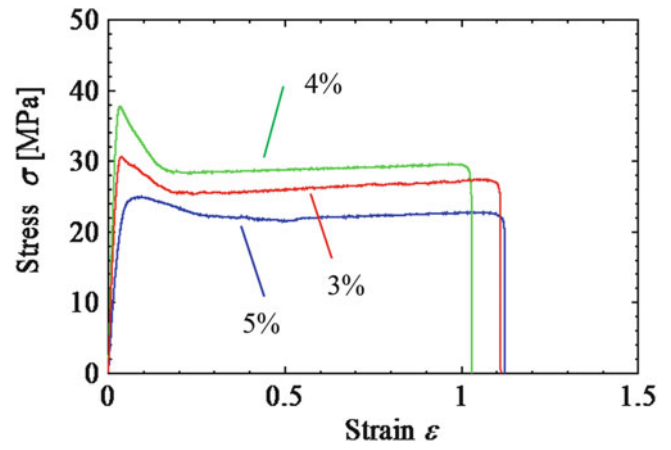


Fig. 3.6 Stress-strain curves of dynamic tensile test ($850\text{--}1400\text{ s}^{-1}$)

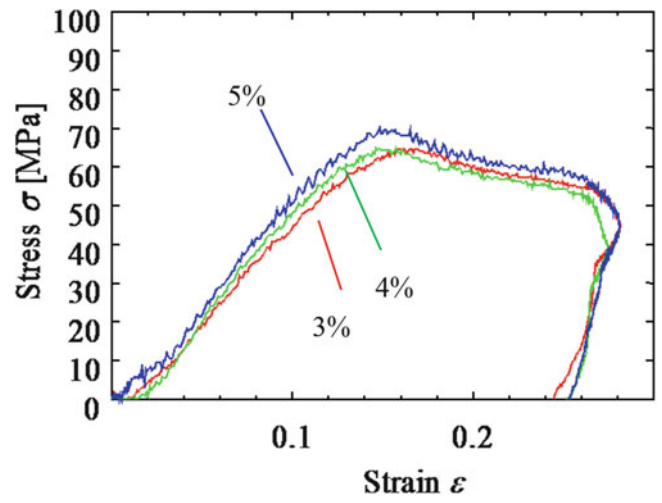


Fig. 3.7 Effects of mixing ratio on maximum stress

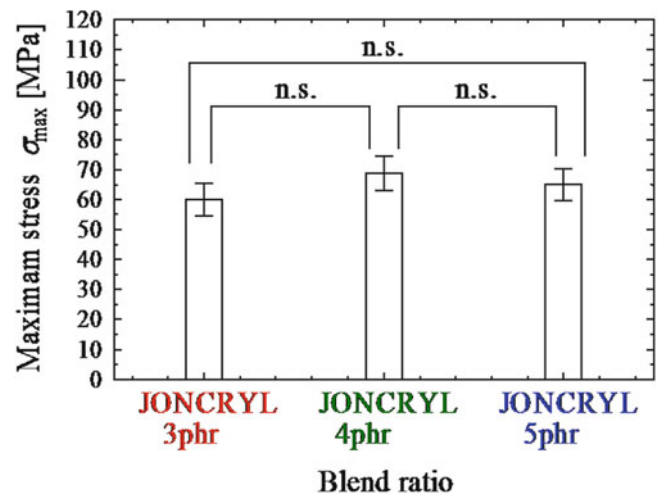


Fig. 3.8 Effects of mixing ratio on elongation at break

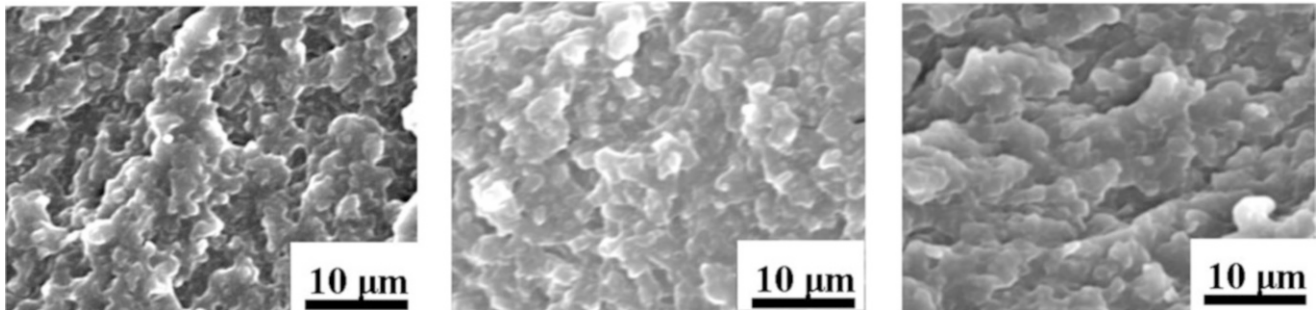
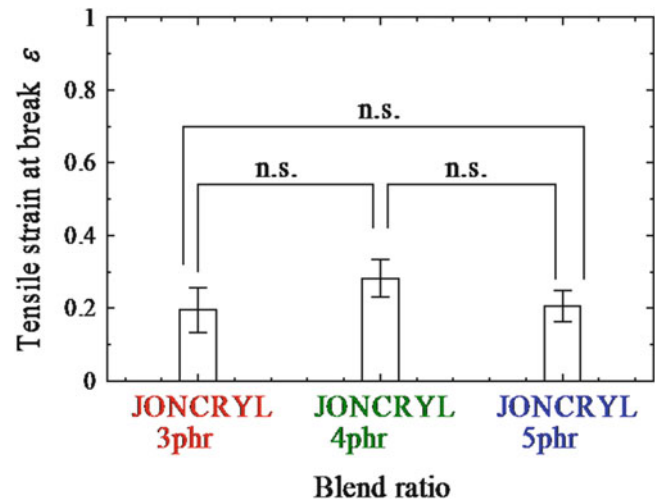


Fig. 3.9 Fracture surfaces of specimens at high strain rate

however, the comparisons showed no significant differences. Figure 3.9 shows SEM photographs of the fracture surface. The fracture surface was flat, and no clear elongation was observed. It seems that the elongation of materials decreased at high strain rate.

3.4 Conclusions

Dynamic tensile properties of a bioplastic PLA/PLBS blends were investigated experimentally using a tensile split Hopkinson bar and a universal testing machine. The additive clearly increased elongation at break at low strain rate. However, the additive increased slightly elongation at break. At low strain rate, stress-strain curves showed strain softening.

Acknowledgment Dynamic tensile properties of bioplastic PLA/PLBS blends were investigated experimentally using a tensile split Hopkinson bar and a universal testing machine. At low strain rate, the additive clearly but only slightly increased elongation at break. At low strain rate, stress-strain curves showed strain softening.

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