# **Optical Parameters of ZnO Thin Films**



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**Abstract** Zinc oxide (ZnO) is a very important material with diverse uses in solar cells, optoelectronic devices, and other applications. Optimizing the optical properties is required in such applications. In this work the influence of annealing in nitrogen atmosphere and aluminum doping on the optical properties was investigated. Undoped (ZnO) and aluminum doped (ZnO:Al) thin films were deposited using spray pyrolysis (SP) method on glass substrates, and some of these films were annealed in nitrogen atmosphere. The microstructure of the films was explored using X-ray diffraction, and it is found that the films are polycrystalline. Transmittance of the films was recorded as a function of wavelength in the range of 300–1100 nm. Transmittance was used to calculate the absorption coefficient, extinction coefficient, reflectance, refractive index, real and imaginary parts of the dielectric constant, and optical conductivity. All these parameters are found to be affected by Al-doping and annealing. It is found that optical parameters were influenced by Al-doping and annealing.

**Keywords** Characterization • Electronic materials • Energy conversion and storage • Transparent conducting oxides • Zinc oxide • Spray pyrolysis

## Introduction

Zinc oxide (ZnO) is a wide band gap, II-VI compound semiconductor that has attracted the attention of the scientific community. It has three phases; the cubic (zinc blend) phase of space group  $F\overline{4}3m$  or  $T_d^2$ , which is a metastable phase that can be stabilized when it is grown on cubic substrates, rocksalt which can be obtained at relatively high pressures, and hexagonal (wurtzite) phase of space group P63mc or C6v4, which is the thermodynamically stable phase under ambient conditions. The lattice constants of the hexagonal phase are c = 5.025 Å and a = 3.249 Å

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[1, 2]. ZnO has several applications including solar cells, optoelectronic devices, gas sensors, piezo-electric transducers, and phosphors. It is a transparent conducting oxide (TCO) of transparency comparable to that of tin oxide (SnO<sub>2</sub>) and indium tin oxide (ITO), so it is an alternative to both of them. Naturally, ZnO is an n-type semiconductor due to intrinsic defects such as oxygen vacancies and/or zinc interstitials. The n-conductivity can be enhanced by intentional doping. Aluminum doping was found to be effective in improving the electrical and optical properties of ZnO, and aluminuim doped zinc oxide (ZnO:Al) is commonly used as a front contact in thin film solar cells such as CdTe-based and Cu(In,Ga)Se<sub>2</sub> (CIGS)-based solar cells.

There are different physical and chemical methods to prepare ZnO thin films such as chemical solution deposition (CSD) [3], DC magnetron sputtering [4], spin coating [5], and spray pyrolysis technique (SP) [6–10]. The spray pyrolysis (SP) method was chosen because it is cost effective and simple method. The film's microstructure was checked using X-ray diffraction, and the transmittance was recorded at room temperature. The transmittance was used to deduce the reflectance, absorption coefficient, refractive index, extinction coefficient real and imaginary parts of the dielectric function, and optical conductivity of the films. The results were analyzed and discussed in the light of using the films in solar cells' applications.

#### **Experimental Procedure**

ZnO thin films were deposited on ultrasonically cleaned glass substrates using the spray pyrolysis technique. A precursor solution was prepared from  $7.61 \times 10^{-3}$  mol of zinc chloride (ZnCl<sub>2</sub>), and  $1.33 \times 10^{-5}$  mol of Al<sub>2</sub>O<sub>3</sub>, in 300 ml of distilled water, and about 1 ml of hydrochloric acid (HCl). The solution was intermittently sprayed on the hot substrates of temperature  $T_s = 450$  °C. Nitrogen (N<sub>2</sub>) was used as the carrier gas, where the optimum carrier gas pressure was around 5 kg/cm<sup>3</sup>. The microstructure of the films was explored by X-ray diffraction (XRD) using XRD-7000 X-Ray diffractometer with Cu K<sub>a</sub> radiation ( $\lambda = 1.54$  A°). Transmittance was measured at room temperature using a double beam Shimadzu UV 1601 (PC) spectrophotometer in the wavelength range  $\lambda = 290-1100nm$  with respect to a piece of glass similar to the substrates. Annealing was performed in nitrogen atmosphere at 250 °C for 45 min.

#### **Results and Discussion**

Figure 1 displays the X-ray diffractograms of two films of the set under study. The films are polycrystalline with a hexagonal (wurtzite) structure. The peaks were assigned to their corresponding Miller indices (hkl). The (100), (101), and (002) diffraction peaks are observed, with the highest intensity is that of the (002) peak. So, the preferential orientation is the one perpendicular to the surface (002), or in



Fig. 1 X-Ray diffractograms of ZnO thin films. a As deposited. b Annealed at 250°C for 45 min

other words along the optical c-axis, which is perpendicular to the substrate [11]. This is because the c-axis has the lowest energy of the hexagonal crystalline structure [12].

Figure 2 shows the transmittance T% of as-deposited, annealed, undoped, and Al-doped ZnO thin films plotted against the wavelength  $\lambda$ . All the films showed high transmittance (60%-92%) in the visible and near infrared spectral regions. The figure shows that the maximum transmittance of the films reaches as high as more than 91% for the ZnO:Al as-deposited and annealed film in the red and near infrared regions, which means the films can be used as forecontacts, windows, or antireflection coatings in solar cells. It is observed that the transmittance increased when the films were doped with Al, and when the films were annealed. The highest transmittance after annealing is due to the increase of films crystallinity, and decrease of film defects especially surface defects, in addition to the decrease of surface roughness. The increase of transmittance with Al-doping is due to the absorption coefficient in the visible region [13]. A slight shift of the band edge towards lower wavelengths was observed, where the annealed ZnO:Al has the shortest cut-off wavelength, while as-deposited ZnO has the longest one.

Figure 3 displays the reflectance (R = 1 - T) of the same films deduced from the transmittance and plotted against wavelength  $\lambda$ . As the figure shows, the reflectance decreases rapidly near the absorption edge of ZnO, then it decreases slowly and becomes approximately constant in the visible and near infrared region, where the films become highly transparent. The reflectance is smallest for the Al-doped and annealed film due to the increase in the density of free charge carriers, which are the electrons. The greatest reflectance is that for the as-deposited ZnO in most of the wavelength range.



ZnO has a direct band gap, and for a direct transition, the optical absorption edge is analyzed by the following equation [14],

$$(\alpha h\nu)^2 = C(h\nu - E_g) \tag{1}$$

where  $\alpha$  is the absorption coefficient,  $h\nu$  is the photon's energy, C is a constant, and  $E_g$  is the direct band gap energy. The absorption coefficient  $\alpha$  was deduced from the transmittance T using the following formula

$$\alpha = \frac{\ln(1/T)}{t} \tag{2}$$

where T is the transmittance and t is the film thickness. To estimate the band gap energy of the films, a plot of  $(\alpha h \nu)^2$  versus  $(E = h\nu)$  was performed and displayed

in Fig. 4. A linear fit was performed in the linear parts of the curves as shown in the figure, and the direct band gap energy was estimated and displayed in the figures. The obtained values of the annealed and doped films are all the same (3.28 eV), while that of the as-deposited undoped film is 3.27 eV. The reason why, the films have approximately the same value of band gap energy is that, they were deposited at high temperature (450 °C), which means the crystal growth was approximately complete, the grains have a large size as can be deduced from the sharp (002) peak, hence doping and annealing have unobservable influence on the band gap energy.

The extinction coefficient k was calculated from the following formula

$$k = \frac{\alpha\lambda}{4\pi} \tag{3}$$

Figure 5a shows the extinction coefficient, and Fig. 5b shows the refractive index for as-deposited, and annealed ZnO and ZnO:Al thin films, both as functions of wavelength in the range  $\lambda = 300 - 1100$  nm. As Fig. 5a shows the extinction coefficient *k* decreases abruptly at the absorption edge, then it becomes approximately



**Fig. 4** The plots of  $(\alpha hv)^2$  versus (E = hv), and linear fits for: **a** as deposited ZnO thin films. **b** Annealed ZnO thin films. **c** As deposited ZnO:Al thin films. **d** Annealed ZnO:Al thin films



Fig. 5 The extinction k coefficient (a), and refractive index n (b) for as-deposited, and annealed ZnO and ZnO:Al thin films as a function of wavelength  $\lambda$ 

constant. Its value for all films at  $\lambda = 550$  nm is in the range 0.022–0.034, where the smallest value is for the annealed ZnO:Al films and the greatest value is for the as-deposited ZnO thin films. The small value of *k* is preferred for use of the thin film in solar cells.

Reflectance is related with the refractive index n and the extinction coefficient k by the following relationship [15]

$$R = \frac{(n^* - 1)^2}{(n^* + 1)^2} = \frac{(n - 1)^2 + k^2}{(n + 1)^2 + k^2}$$
(4)

and the complex refractive index of the film is given by

$$n^* = n + ik \tag{5}$$

Solving Eq. (4) for the refractive index *n* gives

$$n = \frac{(1+R) + \left[(1+R)^2 - (1-R)^2 (1+k^2)\right]^{1/2}}{1-R}$$
(6)

The refractive index decreases with wavelength  $\lambda$  as shown in Fig. 5b, where this decrease is strong in the region of the band edge, but becomes slower after that. Then the refractive index becomes constant in the visible and near infrared regions (1.35–3.5), and  $n_{\infty}$  can be estimated as 1.33 for annealed ZnO:Al, 1.41 for both annealed ZnO and as-deposited ZnO:Al, and 1.57 for as-deposited ZnO. These results are consistent with those obtained by Tüzemen et al. [16] for ZnO thin films prepared at room temperature using pulsed filtered cathodic vacuum arc deposition (PFCVAD), where the value of refractive index in the visible and near infrared spectral regions is in the range 1.5–2.4. Also from Fig. 5b it can be noticed that at  $\lambda = 550$  nm, *n* is restricted in the range in the range 1.75- 2.17 for all films, where the smallest value is

for the annealed ZnO:Al films, and the greatest value is for the as-deposited ZnO thin films. It is well known that the refractive index is mainly determined by the crystal quality, and it is correlated with crystallinity, surface roughness, and resistivity of the films.

The complex dielectric function is given as

$$\varepsilon^* = \varepsilon_1 + i\varepsilon_2 \tag{7}$$

where  $\varepsilon_1$  is the real part which is associated to the refractive index, and the imaginary part  $\varepsilon_2$  which is related to the absorption coefficient or extinction coefficient. The real and imaginary parts of the dielectric function are given by the following equations [15]

$$\varepsilon_1 = n^2 - k^2 \tag{8}$$

$$\varepsilon_2 = 2nk \tag{9}$$

Figure 6 displays the real and imaginary parts of the dielectric function  $\varepsilon_1$ , and  $\varepsilon_2$  respectively as functions of wavelength of light  $\lambda$ . Both of  $\varepsilon_1$ , and  $\varepsilon_2$  decrease with  $\lambda$ , but the decrease of  $\varepsilon_2$  is faster, and in both cases the decrease is fast around the band edge, after which both of them become approximately constant, where  $\varepsilon_1$  for long wavelengths denoted as  $\varepsilon_{\infty}$  is 1.68 for annealed ZnO:Al, 2.00 for both annealed ZnO and as-deposited ZnO:Al, and 2.43 for as-deposited ZnO. These values are consistent with the relationship  $\varepsilon_{\infty} = n_{\infty}^2$ . Figure 6 shows that at  $\lambda = 550nm$  the real dielectric function is in the range of 3.06–4.76, and  $\varepsilon_2$  is in the range of 0.083–0.161 for all films, where the smallest values are for annealed ZnO:Al and largest ones are for as-deposited ZnO. The values of  $\varepsilon_1$  are always larger than those of  $\varepsilon_2$ . That is  $\varepsilon_1$  is related to the refractive index, while  $\varepsilon_2$  is related to the extinction coefficient.



Fig. 6 The dielectric function against wavelength  $\lambda$ . a Real part  $\varepsilon_1$ . b Imaginary part  $\varepsilon_2$ 



The optical conductivity of the ZnO thin films can be determined by the formula [17];

$$\sigma = \frac{\alpha nc}{4\pi} \tag{10}$$

where *c* is the speed of light. Figure 7 depicts the optical conductivity  $\sigma$  against wavelength of incident radiation  $\lambda$ . It is observed that  $\sigma$  shows the same behavior as the other optical constants discussed before. That is it decreases with wavelength  $\lambda$ , and the decrease is fast in the region of the absorption edge, then it becomes slow, and then constant in the visible and near infrared regions. The smallest  $\sigma$  is that of annealed ZnO:Al film and the largest one is for the as-deposited ZnO film.

### Conclusions

Undoped and aluminum doped ZnO thin films were deposited on glass substrates by the spray pyrolysis method at substrate temperature  $T_s = 450$  °C. X-ray diffraction was used to explore the microstructure of the films, where all films showed hexagonal (wurtzite) structure, with preferential orientation along the c-optical axis perpendicular to the substrate. UV–VIS spectroscopy was used to investigate the optical properties of the films. The transmittance was measured at room temperature and used to deduce reflectance, band gap energy, extinction coefficient, refractive index, real and imaginary parts of the dielectric function, and optical conductivity.

The maximum transmittance exceeds 91%, and transmittance for all films is in the range of 60–91% in the visible and infrared regions. The optical energy band gap  $E_g$  is 3.27 for as-deposited ZnO thin films, but 3.28 for annealed ZnO thin film, annealed, and as-deposited ZnO:Al. The extinction coefficient *k* and refractive index

Fig. 7 Optical conductivity  $\sigma$  of ZnO films under study against wavelength of radiation  $\lambda$ 

*n* decrease with wavelength, and the values of *k* are small which reflects the high quality of the films. Real and imaginary parts of the complex dielectric function decrease with radiation wavelength and go to a constant value. The behavior of  $\varepsilon_1$  follows the refractive index, while the behavior of  $\varepsilon_2$  follows that of the extinction coefficient. These results demonstrate that good quality of the prepared ZnO thin films and their potential use in solar cells and the optoelectronic industry.

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