

Cesium Lead Bromides—Structural, Electronic and Optical Properties



Aneer Lamichhane and N. M. Ravindra

Abstract In recent years, it has been found that lowering the dimensionality of halide perovskites leads to enhanced photoluminescence and stability than their three-dimensional counterparts. Further, the change in the dimensionality of an inorganic halide perovskite can evoke surprising ramifications to its intrinsic behavior. The dimensionality in perovskites is governed by its octahedral cages. In zero-dimensional perovskites, the octahedral cages are discrete, whereas in two-dimensional perovskites, they are connected with one another resulting in the formation of a layer. Likewise, in three-dimensional perovskites, the octahedral cages share the corner atoms with each other. This study describes the two-dimensional counterpart of cesium lead bromide perovskites. The structural, electronic and optical properties, in conjunction with their three-dimensional structure, are presented. The emergence of new physical phenomena with respect to the decreasing dimensionality of cesium lead bromide perovskites is analyzed.

Keywords $CsPbBr_3$, $CsPbBr_4$, $CsPb_2Br_5$ · Structural properties · Electronic properties · Optical properties

Introduction

Perovskite solar cells have gained notoriety in the last few years as their light-harvesting capacity has been augmented from 3.8% in 2009 to over 24.2% in 2019 [1]. Despite the demonstration of increasing efficiency of perovskite solar cells in a short time, there are several issues such as fabrication processes, stability, degradation, predictability in behavior, durability, toxicity that have hindered their use in the commercial realm [2–4]. Methylammonium lead halide has been extensively studied

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from the very beginning due to its potential as a perovskite solar cell material [5, 6]. However, this material deteriorates rapidly when exposed to light, heat, air or moisture [7, 8]. As an alternative to methylammonium lead halide, it has been found that lead-based and tin-based inorganic halide perovskites show better stability under external conditions and qualify as suitable materials for solar cells and other optoelectronic devices [9–12]. From the environmental perspective, tin-based perovskites are considered as a better choice than lead-based perovskites [13]. Nevertheless, the efficiency and stability of tin-based perovskites are inferior. Furthermore, it has been found that lead-based perovskite solar cells pose a minor environmental hazard [14–17]. Among several inorganic halide perovskites, cesium lead bromide, $CsPbBr_3$, shows promise as a candidate for the fabrication of solar cells and optoelectronic devices due to its stability, inherent direct band gap, broad absorption spectrum and good transport properties [18–20].

In recent years, it has been found that lowering the dimensionality of halide perovskites leads to enhanced photoluminescence and stability than their three-dimensional counterparts [21–24]. Further, the change in the dimensionality of an inorganic halide perovskite can evoke surprising ramifications to its intrinsic behavior. The dimensionality in perovskites is governed by their octahedral cages. In zero-dimensional perovskites, the octahedral cages are discrete, whereas they are connected with one another forming a layer in two-dimensional perovskites. Likewise, in three-dimensional perovskites, the octahedral cages share the corner atoms with each other. Generally, 2D perovskites are synthesized by inserting some suitable chemical that sits in the intercalated region and acts as a spacer between the layered structures. This technique is profound in 2D organic–inorganic hybrid perovskites. Moreover, the variation of such a spacer not only produces the desired stability to the structure but also yields different functionalities of significant interest to the required 2D system. This is illustrated in Fig. 1.

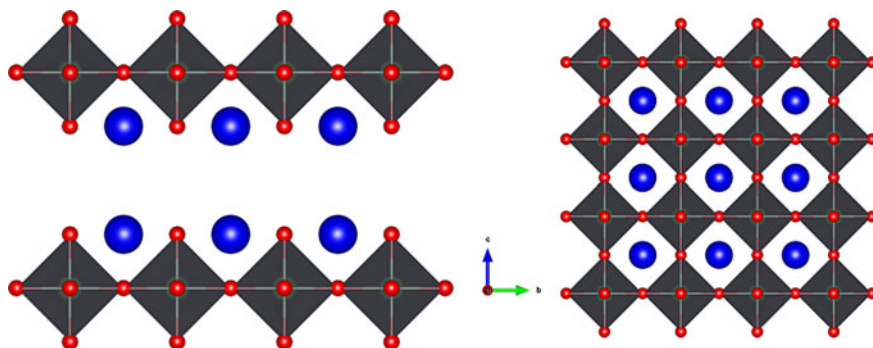


Fig. 1 Illustration showing the role of intercalated atoms as a spacer between the layer of octahedral cages in 2D (left) perovskites whereas the corner atoms link octahedral cages in all directions in 3D (right) perovskites. (Color figure online)

This paper describes the two-dimensional (2D) counterpart of $CsPbBr_3$. One plausible model for 2D- $CsPbBr_3$ would be Ruddlesden–Popper (RP) phase [25]— Cs_2PbBr_4 . Unfortunately, the RP phase is not so frequent in halide perovskites in contrast to oxide perovskites [26]. Nevertheless, it is of interest to proceed with the theoretical study of RP phase of $CsPbBr_3$. On the other hand, the most likely second model for 2D counterpart of $CsPbBr_3$ would be ternary halogen-plumbate $CsPb_2Br_5$. In contrast to Cs_2PbBr_4 , $CsPb_2Br_5$ can be synthesized at room temperature, different from $CsPbBr_3$ which requires a higher temperature. The first report on the synthesis of $CsPb_2Br_5$ was probably mentioned by Yu et al. [27] describing its efficient photoluminescence in the visible region (512 nm) with a quantum yield of 87%. In the paper of Sun et al. [28], the authors have reported that $CsPb_2Br_5$ results as a byproduct during the synthesis of $CsPbBr_3$, yielding higher photoluminescence by transitioning to $CsPb_2Br_5$. However, the work of Jiang et al. [29] has some contradiction by reporting $CsPb_2Br_5$ as an indirect band gap material with inactive photoluminescence. Further, Zhang et al. [30], in their paper, have clarified from the luminescence mechanism that $CsPb_2Br_5$ exhibits a band edge emission in the ultraviolet region and photoluminescence is associated with $CsPbBr_3$ byproduct in $CsPb_2Br_5$.

The need for a theoretical study of Cs-Pb-Br variants is significant due to the complexity in the synthesis and characterization of these materials. The aim of this paper is to study the structural, electronic and optical properties of 3D- $CsPbBr_3$ in conjunction with its 2D counterparts both RP phase Cs_2PbBr_4 and $CsPb_2Br_5$, utilizing the framework of density functional theory (DFT). The emergence of new physical phenomena with respect to the decreasing dimensionality of $CsPbBr_3$ is analyzed. It is anticipated that this work will be beneficial in the design and fabrication of solar cells and other potential optoelectronic devices.

Computational Details

This work utilizes first principles calculations based on DFT in which projector augmented wave (PAW) method was implemented using the Vienna Ab initio Simulation Package (VASP) [31, 32]. All the calculations were performed within the generalized gradient approximation (GGA) using Perdew, Burke and Ernzerhof (PBE) as exchange-correlation functional [33, 34]. The plane wave basis functions with large cut off energy—400 eV (greater than 1.3 times the maximum cut off energy) were used in all the three variants of Cs-Pb-Br along with a sufficiently large Monkhorst K-mesh for Brillouin zone integration. The lattice optimizations were performed with total energy convergence criteria of 10^{-6} eV and final force acting on each atom smaller than 0.02 eV/Å. The resultant optimized structures, along with the lattice parameters, are summarized in Table 1. For post processing, simulation tools such as Vesta [35], Vaspkit [36], Phonopy [37] and Sumo [38] are used in this study.

Table 1 Calculated structure of unit cell geometry with lattice parameters (a, b, c) along with their corresponding literature values in Angstrom

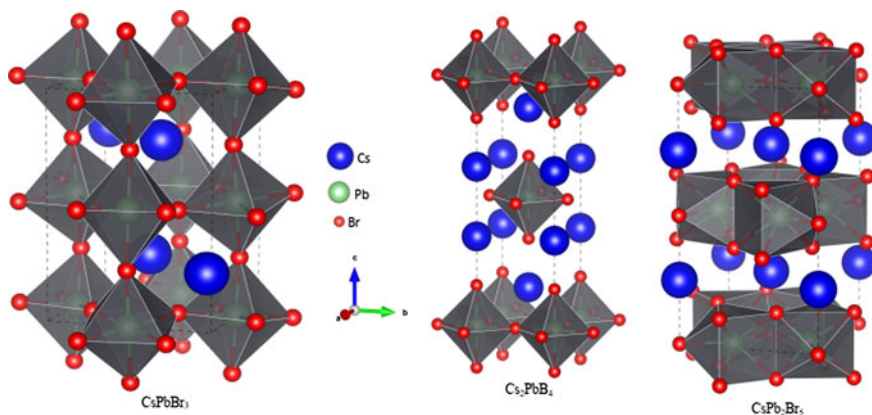
Material	Crystal system	Space group	Unit cell dimensions	Others work Experimental (Theoretical)
$CsPbBr_3$	Orthorhombic	Pnma (62)	$a = 8.34, b = 8.40$ $c = 11.97$	$a = 8.21, b = 8.25$ $c = 11.76$ [39]
Cs_2PbBr_4	Tetragonal	I4/mmm (139)	$a = b = 5.97$ $c = 18.33$	$(a = b = 5.95, c = 18.19)$ [40]
$CsPb_2Br_5$	Tetragonal	I4/mcm (140)	$a = b = 8.61$ $c = 15.47$	$a = b = 8.49$ $c = 15.19$ [41]

Results and Discussion

Structure and Stability

The computed structure of $CsPbBr_3$ crystallizes in the orthogonal space group of Pnma (62) and its 2D counterpart Cs_2PbBr_4 and $CsPb_2Br_5$ in the tetragonal space group of I4/mmm (139) and I4/mcm (140), respectively. Their structures are shown in Fig. 2.

$CsPbBr_3$ has interconnected or corner-sharing octahedron cage $[BX_6]^{-1}$ and Cs^{+1} residing at the center formed by eight such octahedral cages, whereas they are disjoint in Cs_2PbBr_4 . Similarly, the structure of $CsPb_2Br_5$ reveals that Cs^{+} resides in the intercalated region of $[Pb_2Br_5]^{-}$ layers. Further, the stability of these molecules can be verified by calculating their cohesive energy. The cohesive energy per atom (ΔE_c) for any molecule, say $A_aB_bX_x$, is quantified through the relation,

**Fig. 2** Structure of 3D and 2D variants of Cs-Pb-Br. (Color figure online)

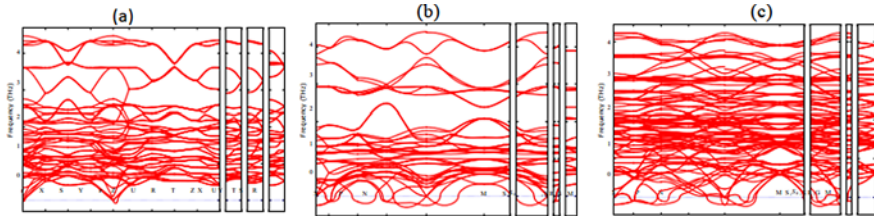


Fig. 3 Phonon dispersion diagram-**a** $CsPbBr_3$, **b** Cs_2PbBr_4 and **c** $CsPb_2Br_5$ under harmonic approximation. (Color figure online)

$$\Delta E_c(A_a B_b X_x) = \frac{aE(A) + bE(B) + xE(X) - E(A_a B_b X_x)}{a + b + x} \quad (1)$$

where $E(i)$, $i = A, B, X$ is the energy of an isolated atom i and $E(A_a B_b X_x)$ represents the total energy of $A_a B_b X_x$. The calculated values of cohesive energy per atom in eV for $CsPbBr_3$, Cs_2PbBr_4 and $CsPb_2Br_5$ are 2.89, 2.87 and 2.88, respectively. Therefore, it appears easy to dissociate Cs_2PbBr_4 among the three compounds of Cs - Pb - Br . Moreover, the orthorhombic phase is possibly the ground-state structures of $CsPbBr_3$, and tetragonal phases of their 2D counterparts are unstable at 0 K temperature. This can be seen by the presence of soft modes in their phonon dispersion diagram, as illustrated in Fig. 3. It has been reported that the stability can be affected by temperature as well as with the number of layers, in the case of 2D [42, 43]. Henceforth, one has enough room to suspect that these tetragonal structures might be stable at room temperature or higher, unless they have low phase transition temperature.

Electronic Properties

For simulating their electronic properties, we have computed the band structures, the total density of states (DOS) and partial density of states (PDOS); these are shown in Figs. 4(i) and (ii). One can notice that except $CsPb_2Br_5$, the other variants show direct band gap. Due to heavy atom Pb, the spin-orbit coupling (SOC) is included in all our calculations. In all three variants of Cs - Pb - Br , there is no significant change in the topography of the valence band with the inclusion of SOC. However, to demonstrate the conduction band degeneracy or split due to SOC, an illustration has been shown for $CsPb_2Br_5$. The calculated values of the band gap with SOC and without SOC, along with their literature values, are shown in Table 2.

It is well known that DFT calculations, using standard functional, severely underestimate the band gap, and due to the intrinsic error cancellation between SOC and neglect of quasi-particle corrections, the band gap computed without SOC has higher proximity to the correct value. Further, the orbital contribution of the valence band maximum (VBM) and the conduction band minimum (CBM) can be analyzed from

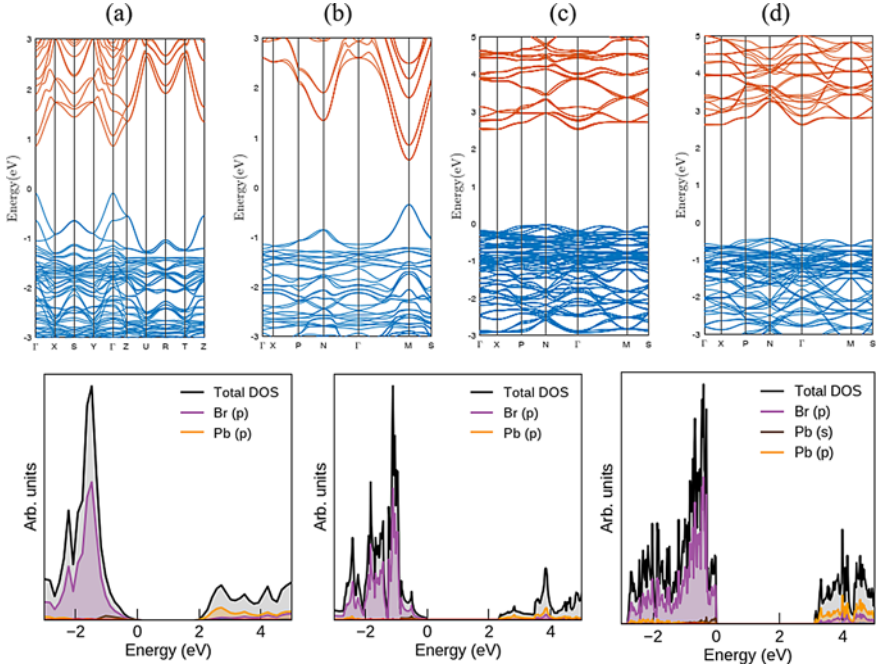


Fig. 4 (i). Top—Calculated band structure diagrams with SOC—**a** $CsPbBr_3$, **b** Cs_2PbBr_4 . **c** and **d** represent $CsPb_2Br_5$ with SOC and WSOC (without SOC), respectively. (ii). Bottom—Their corresponding DOS and PDOS are shown. (Color figure online)

Table 2 Calculated values of band gap (E_g) with SOC and WSOC along with their corresponding literature values in electron volts

Material	E_g SOC; WSOC	Experimental; Theoretical
$CsPbBr_3$	0.95; 2.0	2.25 [44]; 2.16 [45]
Cs_2PbBr_4	0.98; 2.26	—; 2.29 [46]
$CsPb_2Br_5$	2.55; 3.04	3.87 [47]; 3.08 [45]

their DOS and PDOS. In all three variants, the VBM is dominated by Br p state and the CBM mainly constitute Pb p state. It is interesting to note that Cs has no direct contribution to the band edge state.

The absorption of photons in halide perovskite solar cells leads to the generation of electrons and holes. These charge carriers are coupled with each other via Coulomb interaction to form quasiparticles in the form of excitons. The effective mass (m^*) is estimated by the parabolic fitting of energy (E) with momentum (k),

$$m^* = h^2 \left[\frac{\partial^2 E}{\partial k^2} \right]^{-1} \quad (2)$$

and the exciton binding energy (E_b) is calculated by utilizing the Wannier exciton model [48],

$$E_b = \frac{2\mu e^4}{(h8\pi\epsilon(\infty))^2} \quad (3)$$

where μ and \hbar are the reduced effective mass and Planck constant, respectively, e is the electronic charge and $\epsilon(\infty)$ is the permittivity at high wavelength limit (low frequency—static). In the case of $CsPbBr_3$, the computed values of m_e^* and m_h^* , under the effect of SOC, are $0.157 m_e$ and $0.124 m_e$, respectively, and E_b is $51.72 meV$. The literature values of m_e^*/m_h^* are slightly greater/less than $0.2 m_e$, indicating $m_h^* < m_e^*$ [49]. The calculated values of exciton binding energy for $CsPbBr_3$ are in the range of $27\text{--}63 meV$ [50]. Likewise, the values of m_e^* , m_h^* and E_b for Cs_2PbBr_4 are $0.17 m_e$, $0.19 m_e$ and $102.5 meV$, respectively. The values of m_e and m_h for Cs_2PbBr_4 are $0.194 m_e$ and $0.316 m_e$, respectively [51]. Similarly, for $CsPb_2Br_5$, the values of m_e^* , m_h^* and E_b are $0.52 m_e$, $2.41 m_e$ and $292.30 meV$, respectively. It should be noted that the high value of m_h^* can be justified by the flat valence band, as shown in Fig. 4c and d. The relative permittivity values, used in the calculations for $CsPbBr_3$, Cs_2PbBr_4 and $CsPb_2Br_5$, are 4.29, 3.45 and 4.47, respectively. These values are the geometric mean of their respective anisotropic values.

Optical Properties

For investigating the optical properties, the absorption coefficient spectra have been studied for all the three variants of $Cs\text{-}Pb\text{-}Br$. The absorption spectra of materials are of paramount significance as the first and foremost criterion for solar cells should exhibit very high values of the absorption coefficient in the visible range of the solar spectra. Secondly, they play a major role in determining the thickness of cells and therefore in influencing the aspects of cell design. For instance, materials having a higher absorption coefficient are not only suitable for solar cells, but also comparatively thin cells can be designed. The absorption coefficients of materials depend on the frequency of incident photons. The absorption spectral characteristics of $CsPbBr_3$, Cs_2PbBr_4 and $CsPb_2Br_5$ are shown in Fig. 5. The absorption coefficients were calculated from frequency (ω) dependent dielectric functions, $\epsilon(\omega) = \epsilon_1(\omega) + i\epsilon_2(\omega)$ according to the relation [52],

$$\alpha(\omega) = \frac{2\omega}{c} \left[\frac{(\epsilon_1^2(\omega) + \epsilon_2^2(\omega))^{\frac{1}{2}} - \epsilon_1(\omega)}{2} \right]^{\frac{1}{2}}, \quad (4)$$

The absorption edge values found in the literature for $CsPbBr_3$ and $CsPb_2Br_5$ are $2.4 eV$ [53] and $3.26 eV$ [54], respectively, which indeed agree well with our

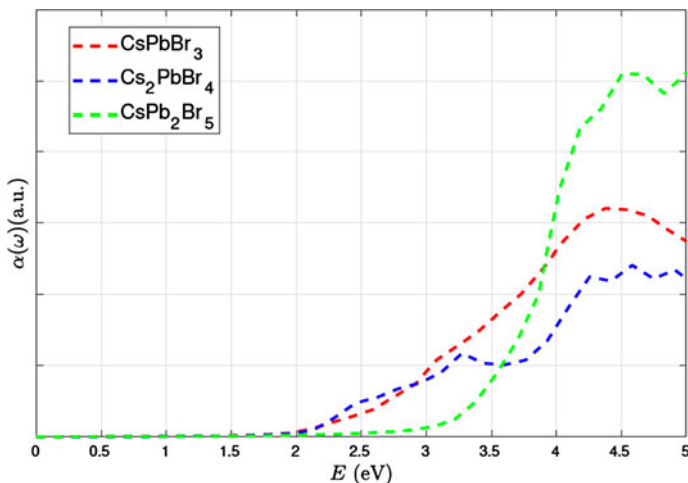


Fig. 5 Calculated absorption coefficients of *Cs-Pb-Br* variants in 100 directions. *CsPbBr₃* is anisotropic in all directions, whereas its counterparts *Cs₂PbBr₄* and *CsPb₂Br₅* are isotropic in 100 and 010 directions. (Color figure online)

computed values. It can be seen that there is no absorption below the band edge of these materials. *CsPbBr₃* and *Cs₂PbBr₄* show broad absorption ranging from the visible to UV region, whereas *CsPb₂Br₅* shows absorption prominent in the UV region. Therefore, *CsPbBr₃* and *Cs₂PbBr₄* are more suitable for photovoltaic applications.

Conclusions

In summary, we have studied the structural, electronic and optical properties of *CsPbBr₃* along with its 2D variants-*Cs₂PbBr₄*, *CsPb₂Br₅*. There is no significant difference in the cohesive energy of these compounds. All of them show anisotropy, and their band structures show noticeable variation in the conduction band region, due to spin-orbit coupling. Except *CsPb₂Br₅*, the other two variants possess intrinsic direct band gap. The optical properties reveal that *CsPbBr₃* and *Cs₂PbBr₄* have absorption edge in the wavelength range of visible to low UV, while *CsPb₂Br₅* shows dominant absorption in the UV region. The calculations show that the excitons are loosely bound in *CsPbBr₃* than its 2D counterparts.

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